PARTIAL MELTING EXPERIMENTS ON CHONDRITIC PRECURSORS:

A POSSIBLE ORIGIN FOR EUCRITES

AND THE BASALTIC ACHONDRITE PLANETOID

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ABSTRACT OF THE THESIS

Partial Melting Experiments on Chondritic Precursors:

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Partial melting experiments on synthetic analogues and natural samples of Murchison and a mixture of 70% H-chondrite - 30% CM-chondrite were conducted to investigate the origins of eucrites and the basaltic achondrite planetoid (BAP). The experiments show that if eucrites were produced by partial equilibrium melting of a chondritic precursor, fractionation of Fe metal and olivine is required on BAP to produce appropriate Fe-Mn-Mg systematics. The eucrites can also be produced by equilibrium crystallization, as well as by fractional melting of the modified H-CM precursor. A comparison between the experiments indicates that the H-CM composition not only matches the oxygen isotope constraint, but is a more appropriate precursor than Murchison since less extensive fractionation is required to produce eucritic melts. The only problem with H-CM composition is a massive overabundance of alkali elements. Production of an H-CM precursor is possible through the collision of two asteroids, and the subsequent melting of that newly formed parent planetoid.

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Introduction

Meteorites can be classified into three major families: the iron meteorites, the stony meteorites, and the stony-iron meteorites (GLASS, 1982). The iron meteorites consist almost exclusively of iron and nickel metal, while the stony meteorites consist of mainly silicate material, and the stony-irons consist of both silicate and metal. Of these three groups, the stony meteorites are by far the most abundant and can be further subdivided into two groups: the chondrites or those meteorites which contain small silicate spheres called chondrules, and the achondrites, those meteorites which do not have chondrules (GLASS, 1982). Eucrites, which are the meteorites of interest in this study, belong to a subgroup of the achondritic meteorites known as the basaltic achondrites. The basaltic achondrites constitute the largest group of achondritic meteorites (JUREWICZ et al., 1993; MASON et al. 1979), with the eucrites being their most numerous member.

The mineralogy of eucrites consists predominantly of pigeonite, plagioclase, the silica polymorphs of tridymite and quartz, with small additions of accessory phases such as augite, ilmenite, chromite, phosphates, troilite, and metal.

Texturally, most eucrites are breccias, containing lithic clasts with basaltic textures. However, some eucrites are interpreted to have cumulate textures (CONSOLMAGNO and DRAKE, 1977).

The two other members of the basaltic achondrites, that are often associated with the eucrites, are the diogenites and the howardites. The diogenites consist almost entirely of orthopyroxene with minor amounts of the accessory phases of augite, pigeonite, plagioclase, olivine, chromite, troilite, tridymite, quartz, and metal, while the howardites are breccias that have a mineralogy similar, but not identical, to a combination of diogenites and eucrites (McCARTHY et al, 1973; MASON et al, 1979). Based on overlapping oxygen isotope signatures, the eucrites, diogenites and howardites appear to originate from the same parent body (CLAYTON, 1993). Since the eucrites and diogenites both appear to have crystallized in relatively low pressure environments, probably at or near the surface, they may represent

different lithologic provinces of the same parent body that are dominated by different rock types or they may be two different lithologies in a layered crust on the parent body (TAKEDA, 1979; DELANEY 1986b, 1986c). The howardites, in either case, probably resulted from impact events that mixed and further processed pre-existing eucrite and diogenite lithologies (WARREN, 1985).

At present, the best candidate for the parent body of the eucrites, as well as the diogenites and howardites, is the asteroid 4 Vesta. Vesta has an optical reflectance spectral pattern that is similar to the spectral data obtained from actual samples, such as the eucrites Nuevo Laredo (McCORD *et al.*, 1970), Nobleborough (FEIERBERG and DRAKE, 1980), Béréba, and Pasamonte (BINZEL and XU, 1993); the diogenite, Johnstown (BINZEL and XU, 1993), and the howardite, Kapoeta (CHAPMAN and SALISBURY, 1973). The samples that match Vesta best are the eucrite, Béréba, and the howardite, Kapoeta. A recent discovery by BINZEL and XU (1993) shows that 20 main belt asteroids (less than 10km diameter) that are in orbits associated with Vesta all have spectral patterns that are consistent with the basaltic achondrites. The asteroidal spectra vary from asteroid to asteroid, with some being nearly identical to Vesta, while others match spectra taken from diogenites, like Johnstown (BINZEL and XU, 1993). Therefore, sufficient variability appears to exist to indicate that the eucrites, diogenites, and howardites (HED) were all derived from the same parent body.

The probability that Vesta is a source of the basaltic achondrites is further strengthened by the BINZEL and XU (1993) investigation of the orbits of their 20 newly discovered asteroids. Twelve asteroids were found to have orbits similar to that of Vesta, while the remaining eight were found to have orbits between Vesta and the Jovian 3:1 orbital resonance. Assuming that Vesta was the parent body of these smaller asteroids, the ejection velocities necessary to place the asteroids into their present orbits was determined. According to BINZEL and XU (1993), the positions of the asteroids span the distance between Vesta and the 3:1 resonance and suggest that other asteroids may have entered the resonance. Since the sizes of ejected fragments from an impact are inversely proportional to their ejection

velocities (BINZEL and XU, 1993), the ejection velocities of small (0.1 to 1.0 km) fragments should be sufficient to reach the resonance, and thereby allow fragments of eucrite, diogenite, and howardite lithologies from these asteroids to be delivered into Earth-crossing orbits. A study by FARINELLA *et al* (1993) argues that fragments from asteroids like Vesta, can be injected into the resonance in approximately 10⁶ years. Subsequent ejection of the fragments from the resonance into an Earth-crossing orbit, based on numerical calculations of fictitious bodies near 2.4 AU (Vesta's location) by FARINELLA *et al* (1992), indicate that the fragments could reach Earth in an additional 10⁶ years. Considering the cosmic-ray exposure ages of eucrites, diogenites, and howardites vary between 5-40 x 10⁶ years (HERZOG and CRESSY, 1977; SCHULTZ, 1987), there appears to be agreement between the existing data and the model, facilitating the possibility that Vesta is the parent asteroid to the HED meteorites. Associating Vesta and its related asteroids with the basaltic achondrites, however, does not explain how the parent body or the eucrites, diogenites, and howardites formed.

There are two general types of models that try to explain the formation of the eucrites. The first model, a fractional crystallization model (MASON, 1962; McCARTHY et al., 1973), suggests that eucrites crystallized from a residual liquid. This residual liquid resulted when a presumably chondritic precursor (olivine-rich) melt experienced olivine fractionation at depth inside its parent body. Fractionated olivine was left behind to form the mantle, while the melt continued to rise towards the surface. Subsequent fractionation of mafic mineral cumulates, near the surface, produced cumulate orthopyroxenites, which are represented by the diogenites, leaving a residual liquid behind to crystallize into the eucrites. A mass balance constraint by WARREN (1985) on the howardites supports this model, arguing that if the eucrites are identical to the basaltic component and the diogenites are identical to the pyroxenite component in the howardites, then more than half of the eucrites probably formed as residual liquids of melts that had previously fractionated from diogenites. The second type of model (STOLPER 1975, 1977; HEWINS and NEWSOM, 1988) suggests that the noncumulate eucrites formed as partial melts of an olivine rich source region and

experienced little or no fractional crystallization during their formation, while the diogenites formed from separate melts.

However, since the achondrite models are attempting to describe the petrogenetic history of a planetoid, it strains credibility to assume that a single style of magmatism was responsible for all of the lithologies sampled by the basaltic achondrites. (The older terminology of planetoid is used here instead of parent body or asteroid to avoid any preconceived ideas about such bodies. Planetoid simply implies a relatively small planetary body which experienced geological processes that are usually associated with planets, such as magmatism or impacts.) More likely, both partial melting and fractional crystallization mechanisms acted to varying extents in different geographical areas of the source planetoid (DELANEY, 1986a, b, c; DELANEY et al., 1981). The distribution of the igneous lithologies on the Earth and Moon are quite variable in composition and regionally are dominated by different lithologies that formed under different stratigraphic, lateral, or temporal conditions on the planetary body. The Earth and Moon therefore provide precedents for the basaltic achondrite planetoid (BAP) having a provincial distribution of igneous lithologies, with some areas of crust being dominated by eucrites, while others contain more abundant diogenite or ultramafic lithologies. The cosmic ray exposure ages for basaltic achondrites (AYLMER et al., 1986; HERZOG and CRESSY, 1977; HERZOG et al., 1978; SCHULTZ, 1987) indicate that approximately three to seven cratering events (depending on the interpretation of the data) within the last 40 million years were responsible for the ejection of samples from their parent body. A single igneous province on BAP, therefore, is unlikely to have been sampled.

STOLPER (1977) demonstrated that partial melting experiments on eucrites at low temperatures (1160 to 1180°C) and at oxygen fugacity conditions of one log unit below the iron-wüstite or Fe-FeO buffer yielded a glass composition that was multiply-saturated with olivine, pigeonite, plagioclase, spinel, and metal. The multiple saturation indicates that the melt probably formed at or near an invariant point and is likely to be a significant source of magmas of constant composition. Since this mineralogy is compatible with the partial melting

of a chondritic meteorite, an implication of STOLPER'S results is that eucrites might be formed by the fairly simple process of partial melting of a chondritic precursor.

Since STOLPER'S (1977) study, there have been several studies concerning the melting of chondritic meteorites. Some experiments, such as those done by TAKAHASHI (1983) and WALKER and AGEE (1988) concentrated on aspects of melting that are not directly applicable to the eucrite problem, such as very high pressure 30 kilobars (TAKAHASHI, 1983) or thermal gradients (WALKER and AGEE, 1988). However, others did partial melting experiments that are very relevant to achondrite genesis. KUSHIRO and MYSEN (1979) did partial melting experiments on the L6 chondrite, Y74354, and showed that their 1200°C glasses were within the compositional range of the eucrites and howardites and therefore suggested chondrites may have a genetic relationship with the basaltic achondrites. BARTELS and GROVE (1991) experimented with partially melting magnesian eucrite clasts from howardites, at one atmosphere and one kilobar pressure, to determine constraints on the composition of eucritic precursor material, as well as, the melting and crystallization processes on the parent body of the eucrites. JUREWICZ et al (1991, 1993) did partial melting experiments on Allende (CV3) and Murchison (CM2) and produced melts that are remarkably similar to the true eucrites and to eucritic melts produced by STOLPER (1977). However, they noted that the MnO content of their resulting glasses was systematically lower than in natural eucrites.

These melting experiments provide evidence that partial melting of a chondritic precursor produces melts that resemble the major and minor element chemistry of the eucrites and therefore suggest the possibility that a chondritic precursor to the basaltic achondrites may have existed. However, two major problems become apparent if any single chondritic class of meteorite is the eucrite precursor. The first problem is that the oxygen isotopic signature of the eucrites does not match any known class of chondritic meteorite. The second problem, as highlighted by JUREWICZ et al. (1993) and recognizable in the experiments of KUSHIRO

and MYSEN (1979), is that melting experiments on typical chondrites produce melts with unrealistically high molar Fe/Mn ratios.

In an attempt to reconcile these problems, DELANEY (1993) calculated a eucritic precursor composition constrained by the oxygen isotopic ratios of basaltic achondrite samples. The model (Table 1 and Figure 1) involves mixing material from two different types of chondrite meteorite in proportions that give the oxygen isotopic signature of the eucrites in the mixture. As both mixing components are chondritic, the major element chemistry of the mixture remains chondritic. The only mixing constraint on the model (in a two component mixture) is that the two components must be on opposite sides of the BAP oxygen mass fractionation line, so that the resulting mixture has the isotopic signature of the eucrites. The two major types of meteorites that were involved in the mixing models were carbonaceous chondrites and ordinary chondrites. They were used because the closest chondritic meteorites to the BAP fractionation line are the H chondrites (a type of ordinary chondrite) on the δ^{17} O enriched side and the CM or CV chondrites (two types of carbonaceous chondrites) on the $\boldsymbol{\delta}$ ¹⁷O depleted side. However, other members of the two types of meteorites were also explored as possible mixing members. The model subsequently uses some of the carbon present within the precursor to reduce sufficient iron metal from the silicate FeO component so that appropriate silicate molar Fe/Mn ratios for the eucritic precursor are reproduced. The results of DELANEY'S calculations yielded several chondritic mixtures [nicknamed "cocktails"] (shown in Table 1) that might be used as a eucritic precursor. One cocktail, produced by mixing 70% H chondrite with 30% CM chondrite, has a remarkable compositional similarity to the 'eucritic parent body' [precursor] calculated by DREIBUS and WÄNKE (1980) and appears to be promising as a model precursor.

This study reports on three series of experiments that used both synthetic and natural starting compositions of the chondritic cocktail (70% H - 30% CM) described above and the CM chondritic meteorite, Murchison. The experiments were designed to: 1) investigate the Fe-Mn-Mg systematics resulting from the removal of iron from a partially melted chondritic

precursor of the eucrites; 2) determine and compare the phase relationships resulting from the partial melting of different chondritic starting compositions; 3) test the DELANEY (1993) model to determine whether an appropriate precursor to the eucrites could be derived from the cocktail (70% H - 30% CM); and 4) allow direct reproduction and confirmation of the experiments done by JUREWICZ et al. (1993).

Experimental and Analytical Techniques

Before any experiments could be run, strict procedures had to be developed so that the experiments were tightly constrained compositionally and to ensure precise and reproducible run temperatures and oxygen fugacities. Without precisely defined constraints a comparison between these experiments and the experiments of other researchers would be impossible. The procedures involved 1) the preparation of the synthetic and natural starting compositions, 2) the calibration of the temperature and oxygen fugacity, or the activity of oxygen present, within the furnace, and 3) the calibration of the electron microprobe analyses, the means by which the experiments were characterized.

Sample Preparation

Three series of experiments were run for this research. The first series of experiments were essentially the test experiments. They were designed to explore whether synthetic compositions were a feasible alternative to using natural materials and to understand the effects of Fe loss on the Fe-Mn-Mg systematics. The understanding of the Fe loss in the experiments is critical, since it can occur due to several factors, both wanted and unwanted. The oxygen fugacity, which controls the amount of iron being reduced and removed from the silicates in the experiment, is the major "wanted" factor since it can be calibrated precisely. A second factor, which can at least be accounted for, is the amount of carbon present within the sample. Carbon reacts with FeO to produce Fe-metal and carbon dioxide. The amount of Fe-metal produced by this reaction can be calculated and subtracted out of the total amount of

metal produced during the experiment to determine the effect of altering the oxygen fugacity settings. The third major factor, which is the most difficult to account for, is the amount of iron lost to the platinum wire that holds the experiment inside the furnace. Since iron readily diffuses into platinum, either the iron that diffuses into the platinum wire from the experiment must be accounted for, or an equilibrium must be attained between the experimental charge and the platinum wire, allowing the experiment to neither gain nor lose iron to the wire. This equilibrium is accomplished by pre-saturating the platinum wire with iron and will be discussed in greater detail later.

The Series I experiments were simplified synthetic equivalents of the CM chondritic meteorite, Murchison (Table 2A) (based on the analysis of JAROSEWICH, 1990). Four analogue compositions (Table 2B) were made and contained seven oxides: SiO2, Al2O3, Cr₂O₃, FeO, MnO, MgO, CaO. The oxides of SiO₂, Al₂O₃, Cr₂O₃, MnO, and MgO were added as pure oxide powders, the "FeO" was from Fe2O3 powder, and the "CaO" was taken from CaCO₃ powder (Table 3). The compositions were prepared by heating up the individual oxides and carbonates to 250°C to drive off any adsorbed water that was present. The oxide powders were then weighed and assumed to be anhydrous. The compositions were finally assembled and ground into a fine powder (to about 75 microns) using a SPEX grinder. The four analogue compositions had four different bulk molar Fe/Mn ratios. Despite the attempt to remove the adsorbed water from the compositions, the effects of the added water in the oxides resulted in the incorrect Fe/Mn ratios in the four analogue compositions. Although the nominal Murchison Fe/Mn ratio of 140 was initially intended to be used (along with Fe/Mn ratios of 100, 60 and 20), microprobe analyses of glasses made from the four compositions showed that the errors resulted in the highest bulk Fe/Mn ratio being 85. The three other compositional ratios turned out to be 65, 45, and 20. Although the ratios were not those planned, the analogues were used anyway and served their planned purposes. Table 2B shows the nominal Murchison composition (Fe/Mn 140) and the averaged microprobe analyses of the four analogue compositions used.

The Series II experiments were used to investigate eucritic origins. These experiments were run with two compositions that were prepared from <u>synthetic</u> materials (Table 2A and 2C). The compositions consisted of a Murchison analogue and a modified version of the 70% H chondrite - 30% CM chondrite mixture proposed by DELANEY (1993). This series of compositions consisted of the same oxides as in the first series, but also included the oxides TiO₂, Na₂O, and NiO (for the Murchison composition only). The TiO₂ and NiO were pure oxides, but the Na₂O was taken from NaHCO₃ (Table 3). These compositions were heated, weighed, assembled, and ground during their preparation in the same manner as the compositions of the first series. Table 2C shows that the Series II synthetic compositions also did not yield the desired abundances of the oxides. The experimental compositions (especially the H-CM analogue) for these experiments however actually proved quite useful and will be discussed later.

The Series III experiments were run using <u>natural</u> samples of Murchison and a mixture of 70% of the H chondrite, Allegan, and 30% Murchison (Table 2A). Allegan and Murchison were initially received as chips and were ground into fine powders (to about 75 microns) before they were used. The powders were then individually baked at 400°C for two hours, and then weighed out. For the (70%) Allegan - (30%) Murchison mixture, the samples were then ground a second time.

To insure that starting compositions for the synthetic (Series I and II) experiments were homogeneous, high temperature glasses were made from each of the starting compositions. For the Series I compositions, two glasses were made for each of the four starting compositions, while three glasses were made for each of the two Series II starting compositions. After making the glasses, the glasses were analyzed by electron microprobe for the oxide abundances contained within them. The two (Series I) or three (Series II) glass compositions for each starting composition were than averaged and the deviation of each glass composition from the average was determined. Since only two or three glasses were made for each starting composition, the standard deviations were very large, being either

50% (2 glasses) or 33% (3 glasses). The tests, therefore, only prove that there were no large scale heterogeneities present, not that the compositions were homogeneous. Ultimately, to insure homogeneity, a greater number of sample glasses should have been taken. Fortunately, the differences between the glasses for each starting composition were extremely small (less than 2% from the average glass composition for the major elements, and less than 5% from the average for the minor elements for all of the compositions) and the absence of large-scale heterogeneity could be assured.

Since there was limited amount of actual meteoritic material available for the Series III experiments, high temperature glasses were not made to test for homogeneity. Instead, modal analysis was used in conjunction with the compositions of the silicate phases within each experiment to calculate the starting bulk composition. The calculated bulk composition in each experiment was then compared to the nominal bulk composition of either natural Murchison or natural Allegan-Murchison. In general, there was excellent agreement between the calculated and nominal bulk compositions, with only one experiment not yielding good results (Natural Murchison, 1180°C, IW-0.5). This experiment experienced olivine fractionation during quenching, which made the calculation extremely difficult.

Furnace Temperature Calibration

Furnace Description

The furnace that was used for the experiments was a Deltech 31 VTOS/C vertical muffle tube gas mixing furnace (Figure 2). The furnace design was fairly simple, containing a 4 centimeter diameter, 60 centimeter long, alumina tube (called the muffle tube), surrounded by six equally spaced U-shaped Kanthall molybdenum disilicide heating elements, which are capable of sustained operation in air at temperatures up to 1800°C. Encompassing the heating elements and muffle tube, is a steel box which contains 12 cm of insulation. Brass cooling heads attach to and seal against both ends of the muffle tube causing an isolated chamber to be made. The internal atmosphere of the muffle tube can subsequently be

monitored and regulated for the ambient oxygen (or other gas) fugacity. An access port within each cooling head allows the regulated gases to pass through the muffle tube. For the Deltech furnace used, the gases entered at the upper cooling head and exited at the lower cooling head, however, the reverse setup could have just as easily been used. As long as the position of the oxygen fugacity calibration, temperature calibration, and experimental charges are in the same spot, there is no advantage to the direction of the flow of the gases. The brass cooling heads' primary purpose, in addition to providing an isolated chamber, is to water cool the ends of the muffle tube, allowing a stable thermal gradient to exist inside the furnace. This stable gradient prevents erratic temperature fluctuations caused by air drafts. The cooling heads also prevent the instrumentation ports and the access ports from becoming dangerously hot (TURRIN, 1984).

The furnace temperature was regulated by a Eurotherm 211 programmable temperature control unit that is connected to an internal, type B 94% platinum - 6% rhodium versus 70% platinum - 30% rhodium wire thermocouple (KINZIE, 1973; POWELL, 1972). The thermocouple is situated next to the heating elements and enters through an access port in the back wall of the furnace (TURRIN, 1984). This programmable temperature control unit allows the user to implement programmed heating and cooling paths.

Sample Rod and Thermocouple Assembly

To measure the temperature within the furnace, a thermocouple was needed that could function over the temperature range of the experiments (1100°C to 1600°). After looking at the various types of thermocouples, a type S thermocouple (KINZIE, 1973; POWELL, 1972) was decided upon. The thermocouple consists of two wires, one pure platinum wire and one 90% platinum-10% rhodium wire and is included within the sample rod assembly. To make the thermocouple (and sample rod), a 40 cm long, four bore hole alumina rod is attached to a brass cap (Figure 2). The Pt and Pt-Rh wires are each inserted into one of the holes in the alumina rod. The ends of the wires are then butt welded together at one end (the end that is

inserted into the furnace), using an oxygen-acetylene torch, to form a small uniformly melted sphere. When the welded end of the wires is at a different temperature than the unwelded ends, a thermoelectric current is produced. Since the wires are made of different materials, the resistance of each wire is different and a potential is created between them. When a potentiometer is attached to the unwelded ends, the potential can be measured between the two wires. The output voltage can than be monitored and calibrated to give the temperature at the welded end.

Two pieces of platinum wire are then inserted into the remaining two holes in the alumina rod and allow the experimental charges to be suspended slightly below the thermocouple (Figure 3).

Hotspot Location

Calibration of the furnace began by first determining where the hottest point in the furnace was. This point was found by carefully sliding the sample rod up and down within the muffle tube of the furnace, until the maximum possible temperature was read by the thermocouple. The hotspot was located 33 cm from the base of the brass head on the sample rod and remained fixed throughout the experiments. The size of the hotspot was found to be about three cm high and the temperature variation within the hotspot was 1-2°C.

Calibration of the Furnace and Sample Rod Thermocouples

In order to calibrate temperature correctly, the nominal temperature at which some process or event occurs must be known. One of the simplest and most precise methods for calibrating is to compare the measured and nominal melting point of an element or compound. Since the experiments were to be run over a temperature range of 1100°C to 1600°C and calibration temperatures comparable to the experimental range would be preferred, the melting points of gold (1064°C) and palladium (1554°C) were chosen to calibrate against.

The procedure for calibrating takes advantage of the fact that both gold and palladium are excellent conductors, by creating a circuit within the furnace and waiting for the circuit to break when the melting point of Au (or Pd) is reached. To set up the circuit, a piece of thin Pt wire is attached to each of the sample support wires. The thin Pt wires are attached to a piece of Au or Pd wire, by making tiny loops in the thin Pt wires and slipping the Au (or Pd) wire through each loop (Figure 4). The tips of the Au (or Pd) wire are then slightly bent so it does not slide off from the thin Pt wires. Next, a voltmeter is attached to the support wires at the top of the sample rod. If everything is set correctly, when the sample rod is placed in the furnace, there should be little or no resistance [in ohms (Ω)] through the Au (or Pd) and Pt wire circuit just made. The next step is to bring the furnace temperature, if possible, to within 20°C of the Au (or Pd) melting point and allow the furnace to equilibrate. Depending on the difference between the initial and final temperatures, the furnace may take up to 45 minutes to thermally equilibrate, however it is extremely important that the furnace be at thermal equilibrium. After the temperature appears to be stable (at equilibrium), the sample rod is carefully inserted into the furnace, while monitoring the voltmeter and insuring the circuit remains intact. The temperature is increased very slowly (1-2°C every 5-10 minutes) towards the melting point of Au (or Pd), while monitoring the thermocouple until the Au (or Pd) wire melts, breaks the circuit, and makes the voltmeter register infinite resistance. It is important to wait at least 5-10 minutes between each increase in temperature to allow the furnace to equilibrate. After the circuit is broken, the sample rod and furnace display temperatures are then noted for the calibration. If the difference between the sample rod thermocouple calibration temperature and the melting point is more than 3-5°C, a new thermocouple welding should be made to insure that corrosion or recrystallization of the wires does not further hamper the ability to measure the temperature precisely. [It is important for the reader to realize that the displayed furnace readings are not true temperatures, although the values would tend to imply this. In fact, the displayed furnace readings only estimate the desired temperature, but are actually used to control the amount of electrical current being supplied to the heating elements. Since conductivity increases with decreasing temperature, more heat per ampere is produced at lower temperatures than at higher temperatures. The calibrated temperature path shown in Figure 5 demonstrates this effect by NOT being parallel to the nominal one to one line. At low temperatures, it is skewed to the right of the nominal line (current is more efficiently transformed into heat), but passes over to the left of the nominal line at high temperature (current is less efficiently transformed into heat. Table 4 lists the furnace display readings (Dfurn) and the corresponding nominal temperatures taken with sample rod (Tsr) thermocouple in ten degree increments shown in Figure 5.)]

To make a new welding, the lower 3-5 centimeters from both the Pt and Pt-Rh wires should be removed. The two cut ends then are butt welded into a small uniformly melted sphere at the base of the thermocouple. If the subsequent calibration is still off, replacement of the thermocouple wires is probably necessary. A good calibration should give consistent and extremely small temperature differences (1-2°C) for both the Au and Pd calibrations.

Table 5 shows the results of the thermocouple calibration experiments using the furnace display readings and sample rod thermocouple. The first calibration yielded consistent results for both Pd and Au, but needed a 20°C correction. The lower portion of the wire had undergone extensive recrystallization from its previous use, so a new welding was made. Calibration #2 gave inconsistent results, suggesting a completely new thermocouple was needed. Calibration #3 was run with the new thermocouple. The correction factor for the sample rod thermocouple is 0°C, since the thermocouple readings match the melting points. This thermocouple calibration procedure was carried out approximately every 100-200 hours of experimental time, depending on the running temperatures of the experiments.

Furnace Oxygen Fugacity Calibration

Fugacity is a thermodynamic function that is equivalent to activity in non-ideal gases, but can be used in place of partial pressure in reactions that involve real gases and mixtures. At low pressures (~1 atmosphere), such as those experienced within the furnace (TURRIN,

1984), fugacity is essentially equivalent to partial pressure. Inside the muffle tube of the furnace, carbon monoxide and carbon dioxide are used to control the reaction: 2CO₂ → 2CO + O₂. Thus, the oxygen fugacity is proportional to the fugacity of CO and inversely proportional to the fugacity of CO₂, assuming the temperature remains constant, since the sum of the partial pressures of the three gas species must equal one atmosphere (assuming ideal conditions). The oxygen fugacity is also controlled as a function of temperature, assuming the ratio of CO:CO₂ remains constant. An increase in temperature will result in a larger proportion of CO₂ dissociating and causing a proportional increase in the oxygen fugacity. A consequence of this reaction is that the CO-CO₂ system is buffered with respect to oxygen, and will dissociate readily to maintain the total pressure should oxygen be removed from the system. Maintaining a continuous replenishment of gas so that neither gas is ever completely consumed or dissociated allows control of the buffering process and the oxygen fugacity (TURRIN, 1984). Since gas flow rates are proportional to the partial pressure of a gas species, oxygen fugacity control can be attained by regulating the gas flow rates, which was accomplished for these experiments, using precision flowmeters.

The amount of each gas used to control the oxygen fugacity varied depending on the run temperature of the experiment and the specific desired oxygen fugacity value. Since one of the aims of the experiments was to study the effects of iron loss on the silicate portions of the chondritic compositions, all of the experiments were run below the iron-wüstite (Fe-FeO) buffer, where FeO dissociates to form iron metal. The ratio of the volumes of CO:CO₂ ranged generally between 8:1 and 50:1 for temperatures between 1180°C and 1400°C and oxygen fugacities of -0.5 to -1.5 log units below the iron-wüstite buffer.

The calibration of the oxygen fugacity was achieved by two different methods to insure its reliability and accuracy, and will be discussed in detail below. The first method involved the calculation of the electromotive force of a fugacity cell at the IW buffer and subsequently determining the flowmeter settings, by trial and error, that would result in that electromotive force. The second method involved using the flowmeters to determine the IW buffer. This

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method took advantage of the reduction-oxidation reactions of iron at this buffer, by using iron foil as an indicator of the position of the buffer. Assuming both the first and second methods were correctly done, the flowmeter settings from each method should match.

Method I: Calculation and Measurement of the Fe-FeO Buffer

The first method used for determining the iron-wüstite (IW) buffer was to first calculate the electromotive force (e.m.f.) at the IW buffer using the equations from JUREWICZ (1986) and CHOU (1987):

General equation: 1) e.m.f. = $(-\Delta V/2.303 \text{ R}) \times T \times \log (fO_2)$

IW buffer equation 2) e.m.f. = $(0.0496) \times T \times \log (fO_2)$

where the e.m.f is measured in volts, ($-\Delta V/2.303$ R) is the molar volume and R is the gas constant that has a value of 8.314 J/K mol, T is the temperature in Kelvin, and (fO_2) is the oxygen fugacity in atmospheres. Next, the correct settings for the flowmeters were found, using an oxygen fugacity sensor and an electrometer, so that the calculated e.m.f. of the IW buffer appeared on the readout display of the electrometer.

NOTE: If the $log(fO_2)$ was needed the more familiar equations:

General equation: 3) $\log (fO_2)P,T = A/T + B + C(P-1)/T$

IW buffer equation: 4) $\log (fO_2) = (-27215 / (273.15 + T)) + 6.57$

from CHOU (1987) could be used, where A, B, and C are constants relating to the specific buffer, in this case the IW buffer, where T is temperature measured in degrees Kelvin, (fO₂) is oxygen fugacity measured in atmospheres, and P is pressure measured in bars.

Oxygen Fugacity Sensor

The oxygen fugacity sensor used (KINZIE, 1973; SATO, 1971; HUEBNER, 1987; NAFZIGER *et al*, 1971) was a yttrium-zirconia cell that contains an internal four bore alumina tube. The alumina tube accommodates a Pt and 90%Pt-10%Rh thermocouple identical to the thermocouple on the sample rod. On the inside and outside of the closed end of the yttrium-zirconia cell is a piece of tightly attached Pt foil. Two pieces of platinum wire are attached to the Pt foil, one wire for the exterior, one for the interior (Figure 6).

The oxygen fugacity sensor works by having molecular oxygen, which is delivered to the interior of the sensor through one of the bore holes in the alumina tube, ionized inside the sensor. The oxygen becomes ionized when electrons are made available to the oxygen by the platinum electrode, which is in contact with the yttrium-zirconia cell wall. The oxygen ions then pass through the cell wall using the available vacancies in the crystalline structure of the wall. Once the ionized oxygen exits the cell wall, its electrons are stripped from it by the outer platinum electrode, which transports the electrons to the electrometer, which registers them as a current or e.m.f. Meanwhile, the de-ionized oxygen reacts with carbon monoxide to form carbon dioxide. This tendency for oxygen to travel out the cell wall will continue as long as there is a greater oxygen fugacity on the interior of the cell than on the exterior.

Results of Method I

The initial attempts to acquire flowmeter readings, based on the calculated e.m.f. values proved inconsistent, since the furnace did not equilibrate over a reasonable amount of time (1-2 hours). The e.m.f. readings varied widely with the same flowmeter settings and the flowmeter settings did not agree (even closely) with the values sent by the manufacturer. Also, the flowmeters measuring the CO and the exhaust gas output were erratic and did not work properly because the flowmeter floats had a tendency to stick. Since the furnace was previously being run with a mixture of hydrogen and carbon monoxide, the sticking was

probably due to water buildup inside the flowmeters as well as the fact that the flowmeters were not standing vertically, but were inclined.

Under normal conditions the flowmeters should have a high degree of precision. They should allow consistent oxygen fugacity results to within ± 0.05 log units or ± 5 millivolts on the e.m.f. electrometer. The accuracy of the flowmeters should also have been quite good. Estimates of the accuracy up to that point however were about 60% different from the specifications sent with the flowmeters.

To correct the problems with the exhaust flowmeter, it was simply bypassed. Due to its removal, there was no regulation of the exhaust gases. Its removal resulted in the $CO_2/(CO+CO_2)$ ratio coming to within 5-15% of the calculated values for the e.m.f. of the IW buffer. The e.m.f. readings also stabilized. The CO flowmeter was placed under vacuum in an attempt to rid it of H_2O and stop the float from sticking. Both the CO and CO_2 flowmeters were squared and leveled. To further correct the problems, the gas mixing connections, which were suspected of causing back pressure on the outlet side of the flowmeters, due to an inadequate initial setup, were redesigned and rebuilt to insure that pressure from the high volume gas (CO) line would not hinder the flow of the low volume (CO₂) gas line (Figure 7A). The restructured gas mixer is shown in Figure 7B.

After rebuilding the gas equipment, another problem was soon discovered. The available oxygen fugacity sensor was nine centimeters too short to reach the hotspot in the furnace by inserting it through the upper cooling head (the same way the sample rod is inserted). Unfortunately, it also could not be inserted through the basal cooling head due to its design. To further complicate the calibration procedure, the supplies were not available to make a new sensor at the time. Therefore, the initial calibration in Table 6 was taken by inserting the sensor through the upper cooling head, nine centimeters from the hotspot and in a temperature gradient.

Table 6 shows the flowmeter settings that are needed to acquire the e.m.f. values that were calculated using equations 2 and 4 discussed above in the beginning of this section. The

reader should note that there is a temperature difference between Table 6 and Table 4. This difference results because of the use of the thermocouple inside the oxygen fugacity sensor, instead of the sample rod thermocouple. Since the sensor is nine centimeters shorter than the sample rod, the temperature could not be taken in the hotspot. Also, as a result of the oxygen fugacity cell thermocouple being inside the oxygen fugacity sensor, it is surrounded and thermally shielded by the yttrium-zirconia walls and platinum foil.

Comparisons between calculated plots and observed plots of the $f(O_2)$ versus $CO_2/(CO+CO_2)$ and temperature versus $CO_2/(CO+CO_2)$, shown in Figures 8A, 8B, 9A and 9B, indicated that the atmosphere within the furnace was about 0.4 - 0.6 log units more oxidized than it should have been at the given $CO_2/(CO+CO_2)$ ratios (the positions of the observed fugacities are shifted to lower $CO_2/(CO+CO_2)$ ratios than calculated). Since there was a fairly consistent shift of $CO_2/(CO+CO_2)$ across the entire range of $f(O_2)$ and temperature, the shift suggested that a leak existed within the furnace. A test was therefore performed to determine if this was true.

The ratio of CO₂/(CO+CO₂) and the temperature were held constant, while the total pressure (flow rates) of the gases was varied. If there was a leak, the e.m.f. would become more reduced (lower values) with increasing pressure, while the lower pressures would be more oxidized. If there was not a leak, there should be no change in the e.m.f. since the ratio and hence the partial pressure of each gas would remain constant and only the total pressure would change. The two tests that were run and are shown in Table 7, indicated that a leak did exist.

Shortly after determining the leak, it was discovered that the existing oxygen fugacity sensor could be altered and reassembled to allow it to fit into the basal cooling head and reach the hotspot. Due to this fortunate occurrence, it was possible to calibrate the leak. As long as the sensor could calibrate in the hotspot, the leak would be accounted for by the sensor. The decision was made to seal the leak only if the problems persisted, which they did not.

The new oxygen fugacity sensor was inserted into the bottom of the furnace, while the sample rod was also inserted from the top. This allowed a temperature reading from both thermocouples. The oxygen fugacity sensor thermocouple was found to read consistently 6°C lower than the sample rod thermocouple. This difference was caused by the shielding from the yttrium-zirconia cell and platinum foil which surrounds the thermocouple inside the oxygen fugacity sensor. The oxygen fugacity sensor thermocouple was not directly used for temperature calibration purposes, but was used to compare to the sample rod thermocouple readings when needed. Table 8 shows the results of the final oxygen fugacity calibration. Please note that the calculated e.m.f. readings in Table 8 need to have a fudge factor added to them after the iron foil calibration is run, to correct for the difference between the calculated e.m.f. values listed in Table 8 and the observed e.m.f. values determined by Method II with the iron foil.

Method II: Testing for the Iron-Wüstite Buffer Using Iron Foil

The second method for calibrating the oxygen fugacity was to determine the flowmeter settings where Fe $^{\circ}$ foil is first oxidized (2Fe $^{\circ}$ + O $_2$ \rightarrow 2FeO) at a given temperature (i.e. find the IW buffer). The procedure was to first set the temperature to the required value and set the flowmeters to an obvious reducing atmosphere within the furnace (high CO, low CO $_2$) and allow the furnace to equilibrate (about 15-20 minutes). Next, a piece of iron foil was placed onto the sample rod; a note of the CO $_2$ setting was made; the CO $_2$ flowmeter was turned down to limit the amount of oxygen mixing with the furnace atmosphere when opening the top of the muffle tube; and the sample rod was placed into the furnace. The CO $_2$ flowmeter was then reset to its original value. After 15 minutes, the sample rod was removed from the furnace and the iron foil was water quenched. If the foil was oxidized the entire surface of the foil would be cracked and brittle, while if it was still reduced, the surface would remain smooth and glossy. If the foil was still reduced, resetting the flowmeters for more oxidizing conditions and repeating the procedure would be necessary. If it was oxidized, a new piece of foil would

be needed and the flowmeters would need to be set to more extreme values than the initial ones.

The most efficient way to find the buffer, assuming the experimenter is not sure what the flowmeters should approximately read, is to use a binary search procedure. This is accomplished by taking one known oxidizing and one known reducing condition, and subsequently taking a measurement halfway between these two measurements, noting the result after each test. By using this procedure, the range needed to be covered can be quickly constricted, until the experimenter can determine the settings for the buffer. If, however, the flowmeters settings for the buffer are generally known, another method is to creep up on the buffer from the reducing side, and thereby not utilize as much iron foil. Table 9 shows the results of the calibration. These observed e.m.f. readings are the three values determined using the iron foil test. The observed e.m.f. readings differ by 4 millivolts from the calculated e.m.f. values listed in Table 8. Therefore, the list of calculated e.m.f. values shown in Table 8 need to have 4 millivolts added to their values in order to be consistent with this calibration (one millivolt is approximately equal to a change in $f(O_2)$ of 0.01 log units).

The final CO₂/(CO+CO₂) ratios were 3-5% off from the calculated values, substantially better than 60% off when the calibration began. Figure 10 shows a graph derived from Tables 8 and 9 that indicates what CO₂/(CO+CO₂) gas ratio is required at a given temperature to achieve the correct fugacity. The gas ratios on the graph are based on the flowmeter (ml/min) readings, not the arbitrary scale readings also printed on the flowmeters.

Experimental Methodology

The oxygen fugacity and sample rod thermocouple calibrations were performed regularly in between experimental runs to insure that the oxygen fugacity and temperature of the furnace did not deviate with time. The oxygen fugacities, which will be quoted in log units relative to the iron-wüstite buffer reaction, have a confidence interval within \pm 0.1 log units for each experimental run. Most of the experiments were conducted at IW -0.5 log units or IW -

1.5 log units, in order to bracket the work of JUREWICZ et al. (1993), and to assess the influence of different imposed oxygen fugacities on the starting compositions. A few experiments were run at IW-1.0 so that our results could be correlated with those of JUREWICZ et al (1993).

Most of the experiments were run in pairs, however, a few of the Series I experiments were run with up to four charges simultaneously in the furnace, while a few of the Series II experiments were run individually. In each of the Series II and III experiments, the synthetic and natural mixtures respectively, the Murchison and cocktail mixture were run simultaneously so that the charges would be assured of experiencing identical experimental conditions, and permit one-on-one comparisons of the run products for each composition. The charges were either both synthetic compositions or both natural compositions when they were run together. Synthetic and natural compositions were not run simultaneously as these were believed to be sufficiently different that a simple comparison would not be easily made.

Aliquots of approximately 100 mg of powder were pressed into 6mm x 4mm pellets in all of the experiments. For the Series I experiments pure Pt wire was used to hold the experimental charges. Since the effect of Fe loss was to be studied, no attempt was made to impregnate the Pt wire with Fe to minimize Fe loss. The Pt wire was used as a proxy for a reducing atmosphere.

For the Series II experiments, Fe loss was minimized by use of pure Fe wire to hold the charges. However, for experiments above 1200°C, the Fe wire became too weak to support the charges for the duration of the run. The solution which was also employed for all of the experiments in the last series, was to impregnate Pt wire with Fe prior to its use with the experimental charges. Charges of powdered magnetite (100 mg) were heated on the Pt wire loops in the furnace at reducing conditions (1400°C and fO_2 at IW-1.5) for periods of about 48 hours. This resulted in diffusion of iron into the Pt wire. The Fe impregnated Pt wire was then etched in hydrochloric acid to remove the excess iron oxide from the surface. Pellets of the relevant experimental charges were then placed on these Pt-Fe wire loops and placed in

the furnace. The Pt-Fe wire loops were reused as often as possible and were placed in hydrofluoric acid after each run to remove any excess silicate material. Typically, the wire loops lasted for three to five experiments before they became so brittle that they disintegrated.

Microprobe analysis of the iron impregnated platinum wire showed that a decreasing gradient of iron existed through the cross section of wire. The edge of the wire that was in direct contact with the iron pellet contained approximately 40 wt% iron, while on the opposite side of the same piece of wire there was approximately 15 wt% iron. Since it was not practical to analyze each impregnated wire, there was always cause for skepticism during the experiments concerning whether the experimental charge was in equilibrium with the iron inside the wire. To identify the effects of either the loss or gain of iron between the experimental charge and the Pt-Fe wire, two methods were employed. For the synthetic experiments, reversal runs [Cooling/crystallization experiments instead of heating/melting experiments] were used. As long as both experiments yielded nearly the same results, there was a fair degree of confidence that little or no loss or gain of iron to the experiment occurred. For the natural experiments, however, reversal runs were not employed due to the limited amount of natural materials available. To insure these experiments remained in equilibrium with the iron impregnated wire, modal recombination of the bulk composition, using the abundances and microprobe analyses of the phases present within the experiment, was employed. Although a few experiments resulted in iron loss, as will be pointed out later, in general, the results of the experiments were extremely good.

Although platinum was used as the support wire for the charges, several other metals could have been used, with molybdenum being the most likely candidate. The reason platinum was chosen in favor of the others is mainly because 1) platinum does not oxidize and enter the silicate compositions at the range of conditions the experiments were being conducted at; 2) it is very malleable and therefore can be bent to properly support the charges; and 3) it has an extremely high melting point, an absolute necessity to support the

charge at higher temperatures. Molybdenum has certain advantages over platinum because it has a much lower affinity for iron and has an even higher melting point. Unfortunately, molybdenum oxidizes at high temperature, (even in iron reducing atmospheres) where it tends to start entering the structure of some minerals. As an example, STOLPER (1977) used open molybdenum foil capsules to support many of his experimental charges and noted that molybdenum was found to be present as a minor element in many of his experimental glasses, while it was present as a major element in his experimentally produced spinels. Since molybdenum was not part of his starting composition, its presence tends to further complicate any interpretation concerning the phase relationships. It was this last consideration that made platinum the prime choice for the support wire.

Once the charges were placed on the wire loops, they were inserted into the furnace for a period of time thought to be sufficient to allow equilibration for those running conditions. When this time period was up, the charges were quenched. The time necessary to reach equilibrium was initially estimated by looking at the work of other experimenters, such as JUREWICZ *et al* (1993) and STOLPER (1977). The duration for the experiments was then modified depending on the planned running conditions. If running conditions were either at a lower temperature or lower oxygen fugacity than previously published experiments, the duration was lengthened. The duration was lengthened because both lower $f(O_2)$ and lower temperature result in smaller amounts of melt being produced within the experimental system, and the amount of melt generally controls the time to equilibrate. The duration was shortened if higher $f(O_2)$ or a higher temperature were run. Since reversal experiments were run for about the same duration as the original, the experiments could be evaluated to determine if sufficient time was allowed for equilibration.

Because of mechanical limitations on the furnace used, the experimental charges had to be quenched by removing the sample rod from the furnace and placing the charges quickly into water, with the entire quench process taking three to five seconds. Drop quenching through the bottom of the furnace was not possible for these experiments. However, the

results suggest that problems associated with quenching were minor and do not influence the results significantly.

Analysis Procedure

Preparation of Experimental Charges for Analysis

Before the experimental charges from the furnace could be analyzed, they needed to be prepared. First, the charges were removed from the support Fe or Pt-Fe wire by either simply picking them up with clean tweezers or by breaking them off using a pair of pliers. The samples were then placed into short quarter-inch diameter brass tubes and impregnated with epoxy. Once the epoxy hardened, the samples were cut in half and were polished using ten micron, six micron, and one micron diamond laps on polishing wheels. Both halves of each sample were often polished to allow a greater analysis area.

After polishing the samples, they were coated with a layer of carbon, approximately 20 nanometers thick, using a high vacuum thermal evaporation unit. The carbon layer is placed over the sample to avoid an electrostatic build-up on the sample when the electron beam of the microprobe strikes the sample. The electrostatic build-up acts like an umbrella over the sample and deflects the electrons away from the sample, causing analytical errors. Since the carbon layer conducts, it causes any electrostatic charge to dissipate and be grounded. An estimate of the carbon layer thickness is made by inserting a piece of polished brass in with the samples. As the carbon is deposited onto the surface of the brass, the surface changes color. Small amounts of carbon produce a dull orange interference color, however a carbon layer about 20 nanometers thick on brass gives a brilliant purple interference color, which is easily recognizable and reproducible.

Electron Microprobe Analysis

The experimental charges were analyzed on the JEOL JXA-8600 Superprobe in the Department of Geological Sciences at Rutgers University. The charges were characterized

first using backscattered electron imaging and energy dispersive spectrometer (EDS) analysis. The EDS and backscattered imaging provided extremely fast, semi-quantitative and qualitative information concerning the chemistry, texture, mineralogy, and surface features of the experimental charges. However, the analyses were fully quantified for their chemical composition using wavelength dispersive spectrometer (WDS) analysis. An accelerating voltage of 15 keV and a beam current of 10nA, measured in a Faraday cup, were used to allow areas within the charges approximately one to two microns in diameter to be analyzed for their major (5 wt% or greater) and minor (0.1 to 5 wt%) element chemistry. High degrees of accuracy and precision were desired from the microprobe analyses. To ensure high accuracy, previously well-characterized, homogenous, natural and synthetic mineral standards were used (Table 10 and 11 and Appendix I) (JAROSEWICH et al, 1980; TAKEI, 1976, 1978; TAKEI and KOBAYASHI, 1974). Since the silicate phases within the experimental charges were the primary concern of the research, the selection of the standards was controlled by their mineral composition, while the metal and spinel phases, although of interest during the research, were of secondary concerns. Standards with similar elemental abundances and mineral characteristics as the phases within the experimental charges were chosen for the 10 elements contained within the starting compositions (Table 2). Thus, the standards used were olivines, pyroxenes, or feldspars, with chromite being the only non-silicate. To ensure reproducibility, a suite of additional standards were routinely analyzed to test the microprobe calibration. Only after the analyses of all standards were checked against their nominal compositions, and any consistent errors corrected, were any samples analyzed.

Because of the constraints of being able to analyze small volumes and still achieve high precision, the time taken for each analysis needed to be fairly long. The long analysis time needed derives from the fact that the detection limit of the microprobe varies inversely with the square root of time. Thus, for elements that are present within a given phase at trace levels (0.1 wt% or less), the time required to analyze them increases quadratically as the

detection limit is lowered. (e.g. increasing the analyzing time by a factor of 10, improves the precision by only about 3x). For the major elements, analyzing times were set for 40 seconds (of detector live time, which is slightly longer than real time), while the minor elements were analyzed for 100 seconds (GOLDSTEIN et al, 1992). In general, these analysis times resulted in a standard deviation of 3-7% for the minor elements, while the major elements had a standard deviation of 1%.

The compositions for all of the analyses were corrected using the ZAF model, which is a generalized procedure for transforming the measured x-ray intensities into estimates of the mass fraction (e.g. weight percent) of the elements present. The ZAF procedure assumes that there is a linear relationship between the observed intensity of the x-rays and the concentration of the element present. However, the effects of atomic number (Z), absorption (A), and fluorescence (F) cause the data to deviate from this linear relationship, so multiplication factors are used to correct the data, before the final mass fraction of each element present is calculated (HEINRICH, 1991). The problem with calculating the final mass fraction of each element however, is that Z, A, and F are all functions of the composition of the unknown. Since they are related by the equation: $k = C_n(ZAF)$, where $k=I/I_0$ is the intensity ratio for the unknown and standard, and C_{n} is the composition of the unknown, the equation cannot be solved analytically. An iterative procedure is used instead. In order to solve the equation, an estimate of the unknown composition must be made. Based on this estimated composition, the ZAF factors can be calculated. Once calculated, the ZAF factors are substituted into the above equation and the equation is solved for the composition, C_{n+1} . This new composition, C_{n+1} , is then compared to the original, C_n . If C_{n+1} - C_n is less than some predetermined value [usually less than 0.00001, when C is expressed as a mass fraction (0.01%)], the composition is considered to have converged on the true composition. If the difference is greater than the predetermined value, than the new composition's ZAF factors are substituted into the equation above and another composition is calculated. This

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process of iterating is continued until the difference between two consecutively calculated compositions is within the predetermined value limit.

Table 10 lists the analysis details used in calibrating the microprobe. The first column lists the elements that were analyzed, while second column shows the spectrometer diffraction crystal used for each element. Different crystals can be used for many elements depending on which x-ray line is being measured. In the third and fourth columns of Table 10, the background positions are given. These are positions on either side of a given x-ray peak, where x-rays are detected and used to calculate the background intensity. The relationship between the spectrometer position (measured in mm) and x-ray wavelength (measured in mm) is:

$$L = 2R\lambda / 2d$$

where:

L = spectrometer position (measured in millimeters)

2d = d - spacing of the diffracting crystal (measured in nanometers)

R = radius of the spectrometer Rowland circle (measured in millimeters)

The wavelengths can subsequently be converted into x-ray energies (measured in eV) by use of the equation: $E = hc / \lambda$

where: E = energy of the x-ray

h = Planck's constant (= $4.1357 \times 10^{-15} \text{ eV sec}$)

c = velocity of light in a vacuum (= 3 x 10⁸ m/sec)

The background intensity is then compared to the peak intensity to calculate the abundance of the element. The fifth column is the ZAF correction factor used in calculating the mass fraction of the element present. A correction factor must be calculated for each element in the standard that is used in the calibration (e.g. Si and Na both use Plagioclase from Lake County, and have different correction factors). The correction factor is calculated on the basis the bulk composition of the standard. The sixth, seventh, and eighth columns show the

baseline, window, and bias, which are three factors that deal with the detector electronics within the microprobe. The detector electronics, which in this case was a pulse height analyzer (PHA), integrates the total charge produced by an x-ray photon and converts the charge into a single voltage pulse that can be further processed for counting or display purposes. Since the PHA can only select and transmit pulses in a predetermined voltage range, a bias voltage is needed to adjust the incoming charge so the PHA can process it. The bias voltage required to adjust the incoming charge is inversely related to the atomic number of the element producing the photons. The baseline and window control the position and the range over which the pulse height intensity can be read (measured in volts) (GOLDSTEIN et al. 1992). Finally, the last column lists the description of the standard used.

Appendix II contains the list of analyses from all three series of experiments. The appendix lists all of the analyses in oxide weight percent and shows the same data converted into mineral formulas.

Approach to Equilibrium

Confidence that the charges were at, or close to, equilibrium was achieved in three ways. First, microprobe analysis allowed the author to insure that all of the minerals and glasses present within each charge were homogeneous. The second test was made by calculating partition coefficients for Fe, Mg, and Mn in the coexisting olivines and glasses, and comparing them to published results (JONES, 1984; TAKAHASHI, 1978; LONGHI *et al*, 1975). These two tests, while not providing absolute evidence that equilibrium was approached, allowed disequilibria data to be rejected. The final and most reliable method to test for experimental equilibrium was to bracket the equilibrium conditions by running reversal experiments when possible. The normal experimental routine involved placing the charge(s) in the furnace at the target experimental temperature and at the target $f(O_2)$ after allowing the temperature and gas mixture to stabilize. The experiment was then continued for a duration deemed appropriate for the attainment of equilibrium. These experiments could be

considered to be heating experiments. Reversal experiments were performed in three steps. First, the charge(s) was placed in the furnace for two hours at a higher temperature (about 200°C higher) than the target temperature and in oxidizing conditions. After the two hours, the temperature of the furnace was reduced to the target temperature. Once the target temperature was achieved, the oxidizing gas mixture was adjusted to the required (reducing) oxygen fugacity and the charge was allowed to reequilibrate for a duration equal to the normal routine experiment. Thus the reversals were cooling experiments rather than heating experiments. The early stages of the reversals were performed in oxidizing conditions, instead of reducing ones, to minimize the amount iron metal that was lost to the platinum wire while at the higher temperature.

Although complete reversals (both fugacity and temperature) would have been preferred to only temperature reversals, there was no reliable means of assuring limited or no Fe loss to the Pt-Fe support wires. A complete reversal would involve going from high temperature, low $f(O_2)$, to the normal experimental conditions at a lower temperature, higher $f(O_2)$. This reversal will nearly always suffer from Fe loss, because a much greater proportion of Fe diffuses into Pt at higher temperatures and/or lower oxygen fugacities than Fe and FeO diffusing out of the Pt into the charge at a lower temperature and/or higher oxygen fugacities. Although ideally there should be an equilibrium eventually established with no net Fe loss to the charge, in practice some of the Fe will permanently be lost to the Pt during the higher temperature, lower oxygen fugacity portion of the reversal.

Results of Experiments

Series I Experiments

The first series of experiments used four analogues of Murchison, which were supposed to differ only, by their starting bulk molar Fe/Mn ratios (85, 65, 45, and 20). However, the starting compositions seen in Table 2B, show that relative to nominal Murchison, the compositions are MgO depleted. The MgO depletion results from adsorbed water in the MgO

powder. Even though the powder was dried at 250°C for two hours, as mentioned earlier, the attempt was obviously ineffective. The Fe/Mn ratios (140, 100, 60, 20) originally wanted were also not achieved and appear to result from either varied abundances of adsorbed water in the MnO and Fe₂O₃ oxide powders, or perhaps the incorrect weighing of the oxide powders. Although these four compositions bear little resemblance to nominal Murchison, they served their intended purposes.

A list of the experimental parameters and resulting phase compositions is shown in Table 12A, while Appendix IIA contains the microprobe analyses of all the Series I experiments.

This series of experiments, as mentioned earlier, were designed to explore the effects of iron loss on the Fe-Mn-Mg systematics of a partially melted chondritic meteorite, and to determine if synthetic compositions was a feasible alternative to using natural compositions. Since each of the experiments were analyzed by microprobe and the experimental conditions they were run at recorded, the experiments could be used to test whether any consistent problems existed with either the temperature or oxygen fugacity calibration.

The experiments were run at a number of temperatures, but oxygen fugacity was not tightly controlled. Instead, the oxygen fugacity was set for IW-0.5 at 1600°C and left untouched (constant CO₂/CO+CO₂ ratio). The effect on the experiments is that as temperature decreases, the oxygen fugacity decreases, and increases the amount of Femetal produced in the charge. However, since platinum was used to support the experiments and it is a siderophile element, bulk iron loss will occur in the charge, as some portion of the Fe-metal in the charge diffuses into the Pt wire, thus becoming fractionated and isolated from the charge. Essentially, two reactions are trying to reach an equilibrium state simultaneously. First, the Fe in the charge is trying to reach equilibrium with the Fe-metal in the Pt wire. This Fe-metal becomes fractionated from the charge. Secondly, the FeO in the charge is trying to reach equilibrium with the Fe-metal in the Charge. This second equilibrium is caused by the presence of a reducing atmosphere (e.g. IW-0.5 or lower). This Fe-metal should ideally remain within the reacting system of the charge. If an inert support wire were used, the Fe-

metal should not diffuse into the support wire. As a result of these two reactions, a greater total amount of FeO is converted to Fe-metal, than normally would be converted. An additional result of these reactions is the silicates of the charge contain less FeO than would normally be expected given the same oxygen fugacity and supported by an inert wire. This diffusion effect simulates the production and fractionation of Fe-metal into the core during the formation of a planetoid.

An important point concerning the first series of experiments is that many of the charges did not reach equilibrium as planned. Many of the experimental glasses show a fairly wide range of compositions, while the coexisting olivines and pyroxenes show more uniform compositions (Appendix IIA). Many of the lower temperature experiments also contained many relic silica grains, causing certain areas of the charges to contain silica enriched or silica depleted glass pockets. The disequilibrium between the silica enriched or depleted pockets of glass and the remaining charge was caused by insufficient grinding of the starting materials (but mainly the silica), and insufficient time in allowing the charges to equilibrate in the furnace. The disequilibrium condition was recognized when investigating these charges, so the starting materials were more finely ground and the times were substantially extended (nearly four times longer in some cases) for the subsequent Series II and III experiments.

These experiments demonstrated that the bulk Fe/Mn ratio of a CM chondritic composition can be altered (DELANEY and BOESENBERG, 1993) to varying degrees, while leaving the remaining elements in the composition largely unaffected (Appendix IIA). Figure 11 demonstrates that the bulk Fe/Mn ratios from the Murchison Fe/Mn 85 analogues could produce compositions that contained Fe/Mn ratios of 5 to 50 in the glasses, depending on the conditions set. High temperature (1580°C) experiments (Exp. M19), where 90% of the iron in the charge diffused into the Pt wire, could produce similar compositions to very low oxygen fugacity conditions. At lower temperature (1250-1300°C) experiments (Exp. M3 and M4), oxygen fugacities are simulated closer to the range of conditions in which eucrites appear to crystallize.

Along with the reduction of the Fe/Mn ratio, an increase in the normative pyroxene content of the bulk chondritic composition occurs. This is important because eucrites lie essentially on the peritectic point in the olivine-plagioclase-silica pseudoternary (LONGHI and PAN, 1988a, b). A bulk CM chondritic composition, like Murchison, is initially too Fe-rich and olivine normative to produce a glass by partial melting that resembles a eucritic composition. However, with removal of Fe, the bulk CM composition moves directly towards the silica component of the pseudoternary, and towards the pyroxene field (Figure 12). As a result, the bulk composition moves into a position on the phase diagram that produces eucritic glasses when partially melted. The part of the phase diagram, where the most favorable eucritic precursor compositions, such as DREIBUS and WÄNKE (1980) and DELANEY (1993), is the area between where bulk Murchison and the "Fe-free" Murchison compositions plot.

The experiments also showed that a synthetic composition could be used as proxy to a natural composition, but extremely careful procedures must be followed if the starting composition is made from oxide powders. The experiments further indicated that the temperature calibration appeared to be correct, since the petrology of the resulting phases of each experiment was consistent with the settings for the temperature and the amount of Fe loss to the Pt wire.

Series II Experiments: Synthetic Compositions

The synthetic compositions of the Series II experiments did not duplicate the bulk compositions of the chondritic starting materials as planned either (Table 2C). The synthetic Murchison composition was calculated on the basis of the bulk meteorite and assumed <u>all</u> of the iron present within Murchison (FeO + Fe-metal + FeS) was present as FeO. The synthetic H-CM mixture was based on only the silicate portion of the precursor, thus the Fe-metal, FeS, and Ni-metal contents are not included in the composition. However, a problem was discovered with both synthetic compositions. Microprobe measurements of glasses quenched from high temperature revealed that the actual compositions of the synthetic compositions

deviated from those calculated. As with the previous series, the oxide powders retained an abundant amount of adsorbed water, despite the attempts to dehydrate them. The modification is minor for the synthetic Murchison, however the synthetic H-CM composition was affected much more extensively and turned out to be substantially depleted in MgO. However, both compositions were used. Both, the planned nominal compositions and the actual experimental compositions used are shown in Table 2C. Table 13 summarizes the Series II experimental results by showing the average experimental silicate compositions. The complete microprobe analyses of all the Series II experiments are shown in Appendix IIB.

Murchison Analogue Results

Fe-Mn-Mg Systematics of the Synthetic Murchison Analogue

Figure 13A demonstrates the effect of Fe removal during the experiments on the starting synthetic Murchison composition. Plotted in the diagram is the bulk starting composition, the coexisting glasses, and the coexisting olivines after reaction at a variety of temperatures and oxygen fugacities. The curved dashed line, going from the starting composition to the origin, is the path the bulk composition would take as iron is removed, while the curved dashed line passing through the glasses is a representative path for the calculated equilibrium glasses produced in the experiments as iron is removed from the bulk composition. The experiments show the influence on a fairly high Fe/Mn ratio in the starting composition. To reach typical eucritic levels for Fe/Mn within these glasses, approximately 50% iron fractionation is required at an oxygen fugacity between IW-0.5 and IW-1.0. It should be noted that the Fe loss path for the glasses does not pass through the eucrite compositional area on the diagram. In order for a glass to be produced with eucritic ratios for both Fe/Mn and Fe/(Fe+Mg), a starting composition with an initially higher Fe/(Fe+Mg) ratio is required. One way of increasing the Fe/(Fe+Mg) ratio, while maintaining the other elemental ratios within the glasses, is to fractionate out olivine. This is possible because many elements, such as Ti,

Al, Ca, and Na are incompatible to the olivine crystal structure, so their elemental abundances remain undisturbed in the coexisting phases.

Phase Equilibria of the Synthetic Murchison Analogue

Figure 14 shows a close-up portion of the olivine-plagioclase-silica pseudoternary calculated from the work of LONGHI and PAN (1988b) and projected from wollastonite. In the figure, bulk eucrites can be seen clustering near the peritectic point. This point is where the first melt will be produced from those bulk compositions projected into the Opx-OI-Plag pseudoternary (the minimum melting point within the Opx-OI-Plag pseudoternary), which includes all of the synthetic and natural compositions used during this research. The phase boundaries (shown as bands) are plotted to match the Fe/(Fe+Mg)x100 ratio of the glasses, which is about 60. Two of the more "eucritic" synthetic Murchison glasses are shown along the olivine-pyroxene peritectic and demonstrate that there is an increase in temperature away from the peritectic point along the olivine-pyroxene boundary. Because temperature effects the formation at which a given phase can crystallize, with decreased temperature, the synthetic Murchison composition (and other compositions similar to Murchison), causes a more pyroxene and plagioclase normative (more eucritic), and less olivine glass to form (Table 14).

Many of the data points shown in Figure 14 are shifted away from the peritectic point and plot inside the pyroxene field. This shift is caused by varying abundances of sodium within each experiment, because the phase boundaries shift as a function of the Na/(Na+Ca) ratio within the glass. The shift is similar to that caused by varying the Fe/(Fe+Mg) ratio. The phase boundaries shown [and based on the work of LONGHI AND PAN (1988b)], assume that the anorthite content of the glasses is An₉₀, when in fact, many of the glasses have An contents of about An₉₅₋₉₉.

Figures 15 A and B show the olivine-pyroxene phase boundary plotted twice, once for an Fe/(Fe+Mg)x100 ratio of 30 (Figure 15A) and once for an Fe/(Fe+Mg)x100 ratio of 55 (Figure

15B). The figures illustrate that as a composition, becomes more magnesian, the olivine field expands and the olivine-pyroxene phase boundary moves away from olivine. This compositional transformation is well demonstrated by the synthetic Murchison 1300°C, IW-0.5 glass in Figure 15B, and the more magnesian 1300°C IW-1.5 glasses plotted in Figure 15A. The near overlapping of the 1300°C IW-0.5 and 1400° IW-0.5 glasses (Figure 15 B) on the olivine-pyroxene boundary suggests that this point is very likely the point where the synthetic Murchison bulk composition first encounters this phase boundary while crystallizing olivine.

The Effects of Temperature and Oxygen Fugacity on the Synthetic Murchison Experiments

The compositions of the silicate phases in the synthetic Murchison experiments indicate a variety of effects that occur with a decreasing temperature. First, the glass compositions become more plagioclase and pyroxene normative (or more eucritic) (Table 14), while coexisting pyroxenes become more calcic and coexisting olivines become more iron rich (Table 13A). Secondly, there is a increasing fraction of sodium retained in the melts (Table 13A). Despite sodium being a highly volatile element, the amount of sodium retained within the glasses is three to four times larger than in the experimental glasses of JUREWICZ *et al.* (1993). The sodium retention appears to be directly correlated to increases in the normative plagioclase content of the glasses (Table 14). The exception to this sodium trend is the 1180°C IW-1.5 glass, which appears to have suffered from incomplete equilibration and localized relict silica enrichment. There is also an increase in the Fe/(Fe+Mg) and Fe/Mn ratios of all silicate phases present (Table 13A).

The effects of lowering the oxygen fugacity are also indicated in the resulting phase compositions. The first is an obvious decrease in the Fe/(Fe+Mg) and Fe/Mn in the silicates (Table 13A). The decrease in the iron causes a mass balance effect increasing the relative amount of calcium and aluminum present within the glasses and pyroxenes. Higher Cr₂O₃ contents (Table 13A) result in the glasses as fugacity is lowered due to the increased stability of Cr²⁺ over Cr³⁺. Reduction of the bulk charge occurs faster than melting at low to medium

temperatures (1180-1400°C)(based on the iron foil calibrations of the oxygen fugacity). Since reduction causes an increase in the normative silica content (and plagioclase content) of the charge (Figure 12) a corresponding increase in the viscosity of the melt will occur. This increased viscosity will slow the diffusion rate and will extend the time required to attain equilibrium in the experimental charge.

Comparison of Glasses Produced from the Synthetic Murchison Analogue with Eucrites

The synthetic Murchison glasses in general compare favorably with average eucrites (Figure 16A), but contain non-eucritic abundances of certain oxides. The TiO₂ contents are consistently higher than eucrites, while MnO is consistently lower. The latter of these trends can also be recognized in data of JUREWICZ et al (1993) and KUSHIRO and MYSEN (1979).

H-CM Chondritic Mixture Analogue Results

Synthetic H-CM and Natural 70% Allegan-30% Murchison Relationship

The synthetic H-CM composition was initially planned to be a simplified version of the natural 70% Allegan - 30% Murchison mixture, containing only the silicate portion of the natural 70% Allegan - 30% Murchison composition (the metallic iron and sulfide portions being intentionally left out). However, due to the problems caused by the adsorbed water as discussed earlier, the synthetic H-CM composition was modified. This modification increased the Fe/(Fe+Mg) ratio of the planned starting synthetic composition to a ratio of 30.5, which should have been 26, the same ratio as the natural 70% Allegan - 30% Murchison mixture. The difference caused by the modification, is equivalent to approximately 20% olivine being subtracted from the silicate portion of the natural composition. Therefore:

(Natural Allegan-Murchison) = (Synthetic H-CM) + (Metal) + (Sulfide) + (~20% Olivine).

Fe-Mn-Mg Systematics of the Synthetic H-CM Mixture

The synthetic H-CM mixture (Figure 13B) requires very little iron loss (at IW-0.5) from the bulk composition to produce typical eucritic Fe/Mn ratios of 30 to 40 within the glasses at temperatures of 1180-1200°C. This composition, unlike synthetic Murchison, does form eucritic Fe/(Fe+Mg) ratios (1180°C IW-0.5). The coexisting equilibrium silicates to this eucritic glass consist of Fo₇₁ olivine and En₇₁Wo₅ pyroxene (Appendix II, Synthetic H-CM, 1180°C IW-0.5).

The 1400°C IW-0.5 glass (Figure 13B) also seems to represent the best candidate for a diogenitic precursor. This melt, if separated from the existing equilibrium solid phases and allowed to fractionate orthopyroxene and a small amount of olivine, should create a cumulate orthopyroxenite. Unfortunately, the glass plots substantially to the iron enriched side of the equilibrium glass iron loss curve and appears to indicate that the experiment was subject to some form of solid/melt fractionation.

Phase Equilibria of the Synthetic H-CM Mixture

The H-CM glasses plot at the lower end of the pyroxene-plagioclase boundary (Figure 14), which is different from the results of synthetic Murchison. The apparent shift is caused by the higher starting sodium content in the H-CM composition and the constraints of the projection scheme. Higher sodium causes an increase in silica in the resulting melts and makes them more plagioclase normative (Table 14) at low temperature (1180-1200°C). However, several eucrite data also fall in this area and overlap the 1180°C and 1200°C IW-0.5 glasses in the pseudoternary.

In Figure 15B, the effect of fractionation of quenched olivine on the H-CM glass compositions, which plot along the pyroxene-silica boundary (Fe/(Fe+Mg)=55) in the 1300°C IW-0.5 experiments, can be seen. If these experiments had quenched properly, the glasses should have plotted on the olivine-pyroxene peritectic, in essentially the same position as the synthetic Murchison 1300°C IW-0.5 glass. However, since olivine fractionation occurred, the

H-CM glass composition became so silica saturated that the composition moved directly across the pyroxene field to the pyroxene-silica boundary. This glass actually has a much larger range of compositions than shown, starting from its present position and trending directly away from the olivine. The other two experiments (Figure 15A) (1300°C IW-1.5 and 1400°C IW-0.5) however, plot on the olivine-pyroxene peritectic (Fe/(Fe+Mg)=30) and contain no fractionation effect.

The Effects of Temperature and Oxygen Fugacity on the Synthetic H-CM Mixture Experiments

One important difference between synthetic H-CM composition and synthetic Murchison is the formation of pigeonite at the conditions of 1300°C and IW-0.5 (Table 13B) in the H-CM composition instead of only olivine, reflecting the more pyroxene normative starting bulk composition of the synthetic H-CM mixture.

Comparison of Glasses Produced from the Synthetic H-CM Mixture with Eucrites

The glasses produced by experiments on the synthetic H-CM mixture provide the closest compositional match to the mean eucrites yet seen (Figure 16B). The best match to a eucrite was produced at 1180°C and IW-0.5. The major discrepancy was Na₂O, which is very overabundant by a factor of 1.5.

Compared to the synthetic Murchison results, the synthetic H-CM results are less variable, being no more than 20% off on any oxide, except Na₂O, at temperatures of 1180°C and 1200°C and at IW-0.5.

Series III Experiments: Natural Compositions

Natural Murchison Results

Fe-Mn-Mg Systematics of Natural Murchison

The effects of iron removal from natural Murchison on the Fe/Mn and Fe/(Fe+Mg) ratios

of coexisting olivines and glass are seen in Figure 17A. Oxygen fugacities of IW-0.5 remove

about 10-15% of the bulk iron within natural Murchison, but much lower oxygen fugacity

conditions (IW-1.5) are needed to remove approximately 50% of the bulk iron from the

starting composition and reduce Fe/Mn to eucritic values of ~50. Eucritic Fe/Mn ratios (30-45)

can be produced in the glasses from natural Murchison staring material, but the Fe/(Fe+Mg)

ratio is too low (45 instead of 65), so the iron loss curve for the natural Murchison glasses

misses the eucritic compositional field and passes through more Mg-rich compositions

(JUREWICZ et al, 1993).

Phase Equilibria of Natural Murchison

Compositions of the natural Murchison glass (1180°C IW-1.0) plot in LONGHI'S

pseudoternary (Figure 14) close to the peritectic point. They are displaced to the plagioclase

side of the peritectic point because of their high sodium contents (Table 13C).

The 1400°C IW-0.5 glasses fall on the olivine-pyroxene boundary (Fe/(Fe+Mg)x100=30)

(Figure 15A) and appear to be consistent with the synthetic Murchison experiments seen in

Figure 15B. Both sets plot at approximately the position where olivine fractionation path from

the Murchison bulk composition first intersects the olivine-pyroxene peritectic line.

The Effects of Temperature and Oxygen Fugacity on the Natural Murchison Experiments

Modal analysis of the Series III experiments (Table 15) was used to determine the

abundances of the phases present. With increasing temperature, there is an increase in the

modal abundance of glass and Fe-metal present, that corresponds to a decrease in the

abundance of olivine (and possibly chromite). Lowering the oxygen fugacity however, causes increased pyroxene and metal abundances.

Comparison of the natural Murchison experiments to synthetic Murchison experiments shows that the natural compositions appear to retain more sodium than the synthetics (Table 13A and C). This is probably caused by the initial state of the sodium. In the natural compositions, the sodium is chemically bonded within the mineral sites of some silicates in Murchison. In order to volatilize the sodium these minerals must be broken down and the bonds broken first. However, in the synthetic compositions sodium is present as a relatively unstable oxide phase (NaHCO₃ · xH₂O), that may decompose easily with increasing temperature. Because only weak bonds of Na to other elements need to be broken, it volatilizes immediately upon being heated.

Comparison of Natural Murchison to Literature Data

The natural Murchison 1180°C IW-1.0 experiment duplicates the conditions of JUREWICZ *et al.* (1993), and is almost identical to their result (Figures 18 and 19A). The cause of the difference between this natural Murchison experiment and the JUREWICZ *et al.* (1993) probably reflects a small difference in the oxygen fugacity calibration (~0.2-0.3 log units) between the two laboratories.

Comparison of Glasses Produced from Natural Murchison with Eucrites

The glasses from experiments on the natural Murchison composition do not match eucrites very well except for the 1180°C IW-1 run (Figure 19A). The glass from the 1180°C IW-0.5 run is depleted in silica (Table 13C) relative to the other Natural Murchison experiments, containing only ~42-44 wt% SiO₂, when the expected values are 47-49 wt%. A possible explanation for the anomalous results is that a heterogeneous sampling of the starting composition was taken, which may have been enriched in a CAI (calcium aluminum

inclusion) component. A high CAI component could account for the higher than expected abundances of TiO₂, CaO, and Al₂O₃, and the lower MgO content.

A substantial overabundance of sodium can be seen in the glasses of all three Natural Murchison experiments shown on the diagram.

Allegan-Murchison Mixture Results

Fe-Mn-Mg Systematics of the Allegan-Murchison Mixture

Three runs were carried out on the Allegan-Murchison mixture at a temperature of 1180°C, but were run at three different oxygen fugacities (IW-0.5, IW-1.0, & IW-1.5). A fourth experiment was carried out at a temperature of 1400°C and at an oxygen fugacity of IW-0.5 to investigate the possibility of producing a glass from which diogenites could subsequently be derived.

The equilibrium glass compositions in these charges are similar to those in the synthetic H-CM runs, but are more magnesian (Figure 17B and 13B), because the starting composition of the natural Allegan-Murchison mixture has a lower Fe/(Fe+Mg) ratio than that of the synthetic H-CM mixture. At 1180°C and oxygen fugacities of IW-0.5, the Allegan-Murchison glass and olivine show 37% reduction (Figure 17B). As a result, glass Fe/Mn ratios (55-70) substantially higher than those of eucrites are produced when eucritic Fe/(Fe+Mg) ratios are attained. At lower fugacity conditions (IW-1.5), the reverse problem occurs, with inappropriately low Fe/(Fe+Mg) ratios forming with eucritic Fe/Mn ratios.

Phase Equilibria of the Allegan-Murchison Mixture

The two Allegan-Murchison glasses (1180°C IW-0.5 & 1180°C IW-1.0) plot close to many of the synthetic H-CM glasses in the silica-olivine-plagioclase pseudoternary of Figure 14 demonstrating the relationship by fractionation (of metal, sulfide, and olivine) between the two starting compositions. Three analyses from 1180°C IW-0.5 glass seem to trend into the pyroxene field. This trend, however, appears to be an artifact of the projection caused by the

shift of the phase boundaries in the pseudoternary from the increasing Na/(Ca+Na) ratio contained within the glasses. Since the three glass analyses plotting in the pyroxene field have Na₂O contents that are 0.3 - 0.5 wt% lower than the glasses at the peritectic, a shift in the position of the peritectic point (and the phase boundaries) related to the Na/(Ca+Na) ratio is probably occurring. The composition of the 1180°C IW-1.0 glass is the best match to eucrite compositions that were made in the natural experiments. This glass plots slightly above the eucrites (probably because of the overabundance of sodium).

The 1400°C IW-0.5 glass (Figure 15) is displaced from the olivine-pyroxene peritectic line toward SiO₂. This probably indicates that quenched olivine fractionation occurred, causing the glass composition to migrate off the peritectic boundary directly away from olivine toward the silica-pyroxene cotectic line.

The Effects of Temperature and Oxygen Fugacity on the Allegan-Murchison Mixture Experiments

The compositional and modal trends for the natural Allegan-Murchison glasses are extremely similar to those in the natural Murchison experiments. The glasses from the Allegan-Murchison experiments contained the highest abundances of sodium (Table 13D) and therefore had a high albite component within the mineral normative calculations (Table 14). The Allegan-Murchison glasses tended to be more plagioclase and slightly less pyroxene normative than the natural Murchison glasses.

The modal analysis (Table 15) showed that with increasing temperature, an increase in the modal abundances of glass and Fe-metal occurs and corresponds to a decrease in the abundances of olivine (and possibly chromite). The effect of lowering the oxygen fugacity however, causes increased pyroxene and metal abundances.

Comparison of Glasses Produced by the Allegan-Murchison Mixture with Eucrites

The natural Allegan-Murchison mixture does not provide as good a match to the eucrites (Figure 19B) as the synthetic H-CM composition and has deviations from the mean eucrites. The composition however, does yield promising results for producing eucrites from chondritic sources.

Discussion

The eucrites have unique characteristics that distinguish them from other types of meteorites. Among these characteristics are the bulk composition, modal abundances of mineral phases, and the chemical composition of those phases. From these characteristics, the history of the eucrites can begin to be ascertained, from its original formation on its parent planetoid to its final arrival and deposition on Earth. Further information from the apparently related meteorites, the diogenites and howardites, can than be used to interpret a more detailed petrogenetic history of the basaltic achondrite planetoid (BAP).

Igneous rocks can be formed under a variety of circumstances. For basalts, such as the eucrites, there are essentially four dominant processes that can play a role in their formation:

1) equilibrium crystallization, 2) equilibrium melting, 3) fractional crystallization, and 4) fractional melting.

The major control of any igneous rock's formation, however, is the available source region from which the igneous composition derives. For the eucrites and diogenites, the obvious source region is material from the asteroid belt, which is where nearly all of the meteorites come from. Since the majority of meteorites are chondrites and many researchers (STOLPER, 1977; MORGAN *et al*, 1978; DREIBUS and WÄNKE, 1980; JUREWICZ *et al*, 1993) have determined BAP is likely to resemble a chondritic meteorite, chondrites of some type appear to be the preferred starting material. However, a mixture of different chondritic materials is required, if the oxygen isotopes are to be satisfied, such as a composition combining an H chondrite and a CM chondrite (DELANEY, 1993). The formation of an H-CM

planetoid appears plausible considering BUCHANAN *et al* (1993) found carbonaceous clasts in the Bholghati and EET87513 howardites, while ZOLENSKY *et al* (1992) found carbonaceous clasts in the LEW 85300 polymict eucrite, as well as in the G'Day, Kapoeta, Jodzie, Y793497 howardites and LEW 87295 polymict eucrite (ZOLENSKY *et al*, 1995). Even if these particular clasts (most of which were identified to be CM2 clasts) are not remnants of the impact that produced the planetoid but were instead deposited on BAP after its formation (which seems likely), they demonstrate that the necessary chondritic material is available to form the source region for the HED meteorites.

Equilibrium Partial Melting and Crystallization and the Basaltic Achondrite Planetoid

If we assume that equilibrium conditions existed on BAP at the time the eucrites and its associated meteorites formed, a trace of the paths of equilibrium crystallization and equilibrium melting on the olivine-plagioclase-silica pseudoternary should show what results from each process. Investigation reveals that the equilibrium melting and crystallization paths trace over one another, but in the reverse direction, and the final products formed using either melting or crystallization are identical. Therefore, equilibrium crystallization and equilibrium melting will be discussed together.

From the results of the Series II and III equilibrium partial melting experiments, a broadly eucritic melt can be produced from a chondritic source. However, the melt created will not produce appropriate Fe/Mn ratios with eucritic Fe/(Fe+Mg) ratios, and alkalis are overabundant. Ignoring the alkali problem for now (this will be discussed later in describing the formation of the basaltic achondrite planetoid), and concentrating on the synthetic H-CM and natural Allegan-Murchison experiments, since they showed more promising results than the Murchison composition, the main problem is modifying the Fe/(Fe+Mg) ratio of the bulk composition, while keeping its Fe/Mn ratio constant (or vice verse), thereby creating bulk ratios that, when partially melted (or crystallized), can produce eucritic glasses. Figure 20 demonstrates graphically what is required to achieve a satisfactory eucrite precursor

composition once a mixture like Allegan-Murchison has formed. The silicate portion of the Allegan-Murchison composition (labeled nominal H-CM in the diagram) is reduced until approximately 12 wt% FeO is converted to Fe-metal. This reduction moves the bulk composition down the iron loss path from Fe/Mn=54 to Fe/Mn=38. This composition. however, has an Fe/(Fe+Mg) of 19. Based on the natural Murchison and Allegan-Murchison experiments, this Fe/(Fe+Mg) is too low to be a eucritic precursor assuming equilibrium conditions existed, and would require an Fe/(Fe+Mg) ratio of ~30 to produce a glass with a eucritic Fe/(Fe+Mg). In order to modify the composition to a higher Fe/(Fe+Mg) ratio, approximately 20 wt% olivine must be fractionated. Extraction of the olivine moves the composition parallel to the Fe/(Fe+Mg) axis (shown by the hollow arrow) toward the synthetic H-CM bulk composition point. If olivine is the only fractionating phase, only Fe and Mg abundances are substantially modified, leaving the remaining elements essentially unaffected. When partial melting of this olivine depleted composition under reducing conditions occurs (1180°C IW-0.5), a glass with appropriate eucritic Fe/Mn and Fe/(Fe+Mg) ratios will be produced. The same calculations can be done using the bulk composition of Allegan-Murchison, instead of just the silicate portion with the same results being achieved. In this case, a much larger proportion of the original Fe budget (approximately 62 wt% FeO, assuming all Fe is FeO) must be removed by reduction. Although this modification process can be applied to any of the chondritic meteorite classes to achieve similar results [like the CM meteorites (Murchison)], it should be noted that the composition of any single chondritic type of meteorite (such as a CM) requires that larger amounts of olivine and/or Fe-metal be removed from the bulk composition before equilibrium partial melting. The composition of a single chondritic meteorite type also cannot satisfy the oxygen isotope constraints, like a mixture composition, such as natural Allegan-Murchison.

The eucrites, diogenites, and howardites all appear to be related. If the eucrites formed by equilibrium melting or crystallization processes, than the diogenites spatial and temporal relationship to the eucrites on the basaltic achondrite planetoid (BAP) would also need to be

addressed [The howardites, although important, are essentially breccias containing a combination of diogenitic and eucritic lithologies that have been altered by impact processes, and are probably not primary rock types like the eucrites or diogenites (McCARTHY et al, 1973; MASON et al, 1979). They therefore do not influence how the primary lithologies formed].

On BAP, the Fe-metal produced from the reduction of the silicate FeO to Fe-metal (and any pre-existing Fe-metal in the parent body) would most likely sink from its higher density and become a small core (~30% of total mass), while the olivine fractionation that occurred would go into the formation of an olivine rich mantle. The FeO reduction occurs because 1) the ambient oxygen fugacity in that area of the early solar system was near IW-1.0 naturally (SATO et al, 1973; STOLPER, 1977), 2) elemental carbon that is available within the BAP bulk composition undergoes a reaction with the silicate FeO to produce Fe-metal and CO₂, or 3) a combination of the ambient oxygen fugacity and some available carbon reacts with FeO producing Fe-metal. The exact process resulting in the FeO reduction may not be able to be determined. STOLPER (1977) found that the eucrites were saturated in five phases (pigeonite, olivine, plagioclase, spinel, and metal) at conditions very near IW-1.0. As STOLPER (1977) states:

"if conditions of pressure and oxygen fugacity can be found at which a given rock composition has more than three phases at its liquidus, it is probable that this rock was produced by quenching of a liquid and that the generation of this liquid involved equilibria between the liquid and the liquidus phases at the conditions at which the liquidus is multiply saturated."

It is therefore very likely that the oxygen fugacity was very near IW-1.0 on the parent body (and that eucrites are very likely to result from equilibrium processes).

The above quote from STOLPER (1977) also implies that melting just above the solidus temperature for a composition like the H-CM mixture, is equivalent to the liquidus temperature of the eucrites, and therefore the eucrites likely derive from a chondritic (H-CM like) source.

The silica-olivine-plagioclase pseudoternary in Figure 21 shows the paths taken by the liquids and bulk compositions during eucrite petrogenesis. The A-M composition shows the position the composition has before both olivine fractionation and Fe reduction, while the H-CM composition shown is after fractionation and reduction. The equilibrium crystallization path begins at the bulk H-CM composition and trends directly away from olivine during olivine crystallization, until reaching the olivine-pyroxene peritectic boundary. At this point, pyroxene is produced at the expense of olivine as the bulk composition moves along the dotted path marked (a) until the peritectic point is reached and eucritic glass is left. The equilibrium melting path overplots the equilibrium crystallization path but moves in the reverse direction.

The diogenites are orthopyroxenites and appear to be mostly monomict or polymict breccias. Since brecciation usually involves some degree of surface processing or impacting on the parent body and assuming 1) the meteorite sampling seen on Earth is representative of the lithologies on BAP and 2) no large scale tectonic processes occurred on BAP, than the diogenites probably occurred as surface or near surface lithologies, like the eucrites. This would imply that if eucrites are the products of equilibrium partial melting than diogenites cannot be derived from the same magma, but rather must have formed under different petrogenetic and regional conditions, to allow alternative processes (such as fractional crystallization) to occur. It would also imply that a layered crust (TAKEDA, 1979) probably cannot exist on BAP on any planetoid-wide scale, since this would probably force the derivation of one lithology (eucrites) from the residual of another lithology (diogenites). From these implications, we can only assume that BAP contains a heterogeneous crust that varies from region to region. Then how did the diogenites form? Although it is possible to create diogenites through equilibrium partial melting, the bulk composition from which it forms must be extremely pyroxene normative, plotting on the pseudoternary either just below the olivinepyroxene peritectic boundary in the olivine field or plotting somewhere inside the pyroxene field itself. The production of such a bulk composition from a chondritic source however becomes a problem, since the composition does not have a sufficiently large orthopyroxene

stability field (JUREWICZ et al, 1995) and processing involving either fractional crystallization and/or melting must also be considered to account for them.

A more reasonable possibility is that the diogenites formed by fractional crystallization of orthopyroxene from high temperature melts, like those in the 1300 - 1400°C synthetic H-CM experiments. These experimental glasses have compositions containing Fe/(Fe+Mg) ratios ~30 and Fe/Mn ratios of ~30. Fractionally crystallizing this melt would produce orthopyroxenites strikingly similar to diogenites, with the orthopyroxene being ~En₇₅ and Fe/Mn ratios ~26.

Fractional Crystallization and BAP

MASON (1962) first suggested that the eucrites were a product of fractional crystallization from a residual liquid that had previously crystallized diogenite cumulates. This residual liquid resulted when a magnesian (olivine-rich) parent magma (that was presumably chondritic) experienced olivine fractionation at depth inside its parent body. The fractionated olivine formed the mantle, while subsequent fractionation of mafic mineral cumulates, near the surface, produced cumulate orthopyroxenites, which are represented by the diogenites. The remaining residual liquid that was left than crystallized into the eucrites.

Fractional crystallization of a bulk composition like the synthetic H-CM or natural Allegan-Murchison would not yield a residual liquid that looked even remotely like a eucrite (Figure 21, Path b). If we assume that the diogenites formed from an olivine poor fractionally crystallized high temperature melt (like the 1300 - 1400°C synthetic H-CM glass in Figures 15 A & B), as discussed above, then the liquid remaining after orthpyroxene fractionation would be saturated with silica, pyroxene and plagioclase phases, resembling a terrestrial norite or quartz monozonite (MORSE, 1980). The reason for this silica rich melt is that during orthopyroxene fractionation the bulk composition leaves the olivine-pyroxene peritectic boundary and continues straight across the pyroxene field until intersecting the pyroxene-plagioclase boundary, between the eutectic and peritectic points. Fractionation of more

pyroxene and plagioclase phases will then persist until reaching the eutectic point, where an additional phase, a silica polymorph, will begin to crystallize to completion. The peritectic point, where eucrites would crystallize, is never reached during the entire fractional crystallization process. Even if the entire olivine-pyroxene phase boundary were a cotectic boundary [which for many reasons it is not, see DELANEY (1986a), DELANEY *et al* (1981), (1984), HEWINS and NEWSOM (1988), and LONGHI and PAN (1988b) for more details], at best, only a small instantaneous eucritic melt might be produced during the fractional crystallization process. However, the possibility that such a small amount of melt could account for a large proportion of the eucrites is highly unlikely. The major problem is there is no mechanism to halt the fractionating liquid at the peritectic point, and thereby inhibit it from going to completion.

One of few methods of deriving a eucritic melt, which previously crystallized the diogenites, would be to start with a composition that is substantially more Fe-rich [Fe/(Fe+Mg)~50] than the natural or synthetic H-CM compositions, and place it initially, deep (130km+) inside the parent planetoid (BARTELS and GROVE, 1991; GROVE and BARTELS, 1992). This depth, which is equivalent to a pressure of ~500 bars to 1 kbar, effects the olivine-pyroxene peritectic boundary so that upon pyroxene fractionation, the bulk composition does not trend markedly across the pyroxene field like the synthetic H-CM would, but stays slightly above the peritectic boundary, where it would eventually intersect the pyroxene-plagioclase boundary just above the peritectic point. This magnesian composition would thereby be able to fractionate the diogenites and produce a residual liquid to crystallize some eucritic lithologies. A problem with this model however, is that the eucritic melt produced cannot be inhibited from fractionating to completion at the eutectic point. The quantity of eucritic melt produced would therefore only consist of a small fraction of the original composition.

Fractional Melting and BAP

Unlike equilibrium melting and crystallization, fractional melting does not follow the reverse path as fractional crystallization, and, in fact, results in compositions that are radically different from it. During fractional melting, melt is continuously extracted from the bulk (solid) composition as it is formed. For a composition, like the synthetic H-CM or natural Allegan-Murchison, the melt forms at the peritectic point, where the minimum melting temperature for the system, olivine-plagioclase-silica, is located and where the bulk compositions of the eucrites plots. The melt is saturated in olivine, plagioclase, and pigeonite (+spinel+metal). As this melt is extracted, the composition of the residual solids begins to move directly away from the peritectic point (Figure 21, Path d). This extraction of melt continues until plagioclase is exhausted in the residual solids. At this point, the bulk composition intersects the olivine-silica leg of the pseudoternary. A subsequent increase in the temperature, however, will not produce any additional melt, since the peritectic point of the olivine-silica binary would need to be reached in order to produce melt (There is also a corresponding absence of change in the solids). Once the olivine-silica binary peritectic is reached, melt saturated in olivine and pyroxene begins to be generated and extracted. This melt extraction causes the remaining bulk composition to move directly away from the binary peritectic point, down the olivine-silica leg of the pseudoternary, towards olivine. Melt extraction continues until pyroxene is exhausted and melting is halted for a second time (The bulk composition arrives at olivine). Increases in the temperature beyond this point, will not result in the formation of melt, until the melting temperature for the remaining olivine composition is reached. Any melt produced (and extracted) above this implausibly high temperature will only be saturated in olivine, which will eventually be exhausted and the process ended.

The production of eucrites from the H-CM composition during fractional melting is extremely easy. A substantial amount of eucritic melt is created at the beginning of fractional melting, and assuming the sampling of basaltic achondritic meteorites found on Earth is representative of the surface of BAP, the amount of melt could easily account for why the

dominant lithology seen on BAP is eucritic. The diogenites might also form by fractional melting as a residual melt that is produced after the eucritic melt is extracted. The diogenitic melt would form at the olivine-silica binary peritectic, and assuming pyroxene fractionation occurs, an orthopyroxenite would crystallize. If the diogenites did not form as a residual after the production of eucrites, than they would require, much like in equilibrium melting or crystallization, that the primary H-CM bulk composition be modified from its present position in the olivine field to a more pyroxene normative position in the pseudoternary, not producing abundant olivine normative melts upon crystallization.

A Method to Produce a Planetoid by Mixing H and CM Chondrites.

The evolution of the basaltic achondrite planetoid (BAP) has been a problem ever since the discovery of the eucrites. Most calculations of it's bulk composition (CONSOLMAGNO and DRAKE, 1977; DREIBUS et al. 1976, 1977; DREIBUS and WÄNKE, 1979, 1980; MORGAN et al., 1977), have not dealt with the method of formation of the parent body represented by that composition. However, since the model of BOESENBERG and DELANEY (1994a & b) explicitly involves large scale mixing of two (or more) very distinctive components, the method by which they were mixed is of some importance. WETHERILL (1981) pointed out that impacts are fundamental to normal planet forming processes. Collisions were frequent in the early forming solar system. HARTMANN (1986) and HARTMANN and VAIL (1986) have argued that an extreme end member of the spectrum of impact processes involves giant impacts that can radically alter a pre-existing planetary body and completely redefine its characteristics. The BOESENBERG and DELANEY (1994a & b) view of the formation of the BAP planetoid simply applies this logic to the less extreme case of smaller asteroidal sized bodies. In the early solar system, the development of planetesimals produced H chondritic planetoids and CM chondritic planetoids that are represented by meteorites. Collision of two such objects provides a potential mechanism for mixing such apparently different precursor compositions. A result of such a "small" giant impact would almost certainly involve the dispersal of a large proportion of the mass of the impactors after the impact. If sufficient mass were available, perhaps enough material would fall back gravitationally to form a large asteroid like Vesta, the basaltic achondrite planetoid. Total melting of the resulting planetoid could occur, because of the large energy exchange associated with impact processes, causing homogenization of the body prior to the formation of a solid crust and mantle and the onset of the magmatism that is represented by the eucrites, diogenites and howardites. A requirement of a BAP planetoid model, however, is the almost total removal of the alkali elements and other volatile elements as the basaltic achondrites have much lower alkali contents than the precursor composition. While volatilization of the alkalis is plausible, the isotopic measurements of potassium isotopes by HUMAYUN and CLAYTON (1993) appear to preclude conventional liquid-vapor volatilization mechanisms following Rayleigh fractionation laws. HASHIMOTO (1990) and DAVIS et al., (1990) have shown negligible isotope fractionation during sublimation of solid forsterite samples, whereas liquid sample show marked fractionation of Mg isotopes. The isotopic indicators of volatile loss, therefore may not be independent of the mechanism by which the volatile loss occurs. Since the mixing model proposed implies that impact processes were important, it will be important to document the effect of shock modification of silicates on the mechanism of alkali volatilization and hence on volatile related isotope fractionation effects in the future. If a mechanism for alkali volatilization can be identified that avoids the constraints imposed by these potassium isotope measurements, then the "small giant impact" mechanism may be an appropriate way to construct an achondritic planetoid by mixing chondritic material from two or more sources.

Conclusions

1. Eucritic melts can be produced by equilibrium partial melting, equilibrium crystallization, or fractional melting of a chondritic composition similar to Allegan-Murchison mixture. The Fe-

Mn-Mg systematics, however, require that the bulk composition undergo a moderate amount of Fe-metal and olivine fractionation before one or more of the processes occurs.

- 2. The Allegan-Murchison cocktail composition of DELANEY (1993) produces a more favorable precursor composition than CM meteorites, like Murchison (or any other singular class of meteorite), even if the oxygen isotope constraint is ignored, since it requires less fractionation than any single chondritic type of meteorite (such as the CM chondrites).
- 3. The only compositional difference between the synthetic H-CM (or a modified form of natural Allegan-Murchison) glasses and true eucrites is the overabundance of alkalis in the glasses. The alkali content differences between the cocktail composition and natural eucrite samples requires that some unidentified mechanism (such as impact) deplete the alkali elements in the eucrites. Either the alkalis presently seen in H-chondrites were not present in the chondritic components that were mixed to form BAP or the formation of the planetoid led to the removal of the alkali elements. Further work is needed to resolve this problem.
- 4. Orthopyroxene fractionated from compositions similar to high temperature synthetic H-CM (1300-1400°C) partial melts can produce diogenitic lithologies, further supporting the argument that the basaltic achondrite planetoid is composed of at least two mixed chondritic classes.

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Figure Captions:

Figure 1: Oxygen isotope diagram, after CLAYTON (1992) and modified by DELANEY (1993) showing the basaltic achondrite field and several potential mixing components that may be mixed together to produce compositions on the achondrite mass fractionation line.

Figure 2: Schematic details of the Deltech VTOS/C vertical muffle tube gas mixing furnace.

- a) Alumina muffle tube
- b) Upper brass cooling head
- c) Lower brass cooling head
- d) Water outlet to upper cooling head
- e) Water inlet to upper cooling head
- f) Water outlet to lower cooling head
- g) Water inlet to lower cooling head
- h) Gas Inlet at rear of upper cooling head
- i) Gas Outlet at rear of lower cooling head
- i) Furnace thermocouple entering through the back wall of the furnace
- k) Furnace heating element
- I) Insulation
- m) Exterior metal casing surrounding furnace
- n) Hotspot location about 3 cm high
- o) Buss bar heating element connectors
- p) Power supply connectors
- q) Removable brass cap (The sample rod and experimental charges in Figure 3 are inserted into the furnace through this opening.)

Figure 3: Three views detailing the experimental setup on the sample rod.

- A) Generalized cross section of the sample rod showing the a) brass head, b) 40 cm long, four bore alumina tube, c) thermocouple wires, and d) support wires for the experimental charges (Figure is slightly exaggerated for clarity).
- B) Close-up and detailed view of the lower section of the sample rod showing the actual setup of the experimental charges (Figure is slightly exaggerated for clarity).
 - a) Four bore alumina tube
 - b) Alumina rings used to aid in the support of the charges
 - c) Thermocouple wires
 - d) Sample rod platinum support wires
 - e) Fe or Pt-Fe support wires for the experimental charges
 - f) Thermocouple butt weld
 - g) Experimental charges
- C) A different perspective illustrating how the experimental charges were supported and positioned on the Fe or Pt-Fe support wires while in the furnace.
- Figure 4: Temperature calibration setup showing the positions of the sample rod support wires, the thin Pt wires, and the Au (or Pd) wires.
- Figure 5: Graph of temperature taken by sample rod thermocouple versus the furnace display readings showing that the furnace display readings, although similar to actual temperatures, linearly deviate over a broad temperature range.
- Figure 6: Cross section through the lower end of oxygen fugacity sensor and thermocouple assembly. The four bore alumina rod (a) is sealed with epoxy to the inside of the yttrium-zirconia tube (b). Two of the bore holes in the alumina rod are used for the thermocouple

wires (c). A platinum wire (d), which is connected to the interior platinum foil (e) is then passed through a third bore hole. A platinum wire (f) is also attached to the exterior platinum foil (g). The fourth bore hole is used to pass oxygen into the interior of the fugacity sensor.

Figure 7: A) Schematic diagram of the original gas mixer showing that the gas flow from the high volume gas (CO) could create back pressure in the low volume gas (CO2) line. Depending on the flow rates that were used to calibrate the oxygen fugacity initially, and those used during each experiment, the back pressure applied to the low volume gas line would result in oxygen fugacities lower than expected. The effect would be more severe on low oxygen fugacities than high ones, because the greater flowrate from the high volume gas would create a larger back pressure and reduce the fugacity further. B) Schematic diagram of the rebuilt gas mixer showing the inlets for the CO2 and CO gases, the direction of flow, and the five pieces used in the assembly.

Figure 8: Log fO_2 vs. the A) observed and B) calculated gas mixture ratio showing the drastic differences in the observed conditions found within the furnace. In general, the observed conditions were more oxidizing than they were calculated to be by approximately ~0.5 log units.

Figure 9: Temperature vs. the A) observed and B) calculated gas mixture ratio demonstrating the drastic oxidizing conditions found in the furnace. The difference is caused by a leak, probably located near the edges of the muffle tube, where the cooling head comes in contact with the muffle tube. Air may also be able to enter along the edges of the sample rod when it is inserted into the top of the furnace.

Figure 10: Temperature vs. observed gas mixture ratio drawn from the final calibration results in Tables 8 and 9. This graph was used to determine the proper gas ratio necessary to achieve the desired oxygen fugacity conditions during the experiments.

Figure 11: Iron loss effects on the Series I experiments showing the resulting glasses and olivines plotted in an Fe/Mn vs. Fe/(Fe+Mg)x100 diagram. With increasing temperature, more platinum comes in contact with the experimental charge (the melt tends to migrate up the platinum wire) and allows greater amounts of iron to be removed. Therefore large fractionations occur at high temperature (abundant melt), such as in M17 and M19 (1580°C), and substantially less iron is removed at lower temperatures such as M4 (1180°C).

Figure 12: Silica-olivine-plagioclase pseudoternary diagram showing the effects of Fe loss on the bulk composition of synthetic Murchison. The phase boundaries seen are for an Fe/(Fe+Mg)x100=60. Pseudoternary is based on the work of LONGHI and PAN (1988b) and is projected from wollastonite.

Figure 13: Fe/Mn vs. Fe/(Fe+Mg) (molar) diagrams demonstrating the effect of iron loss on the bulk synthetic Murchison and H-CM compositions and their resulting glass compositions as a function of temperature and oxygen fugacity. The olivines in equilibrium with the glasses are also shown. The error bars shown on each experiment represent one standard deviation of compositional range. The H-CM composition is displaced toward the high Fe/(Fe+Mg) side of the Fe-loss trend. Notice in Figure A, the Fe loss curve for the glasses does not pass through the eucrite range, but it does pass through the eucrite range in Figure B for the H-CM glasses. The legend in Figure A applies to both figures.

Figure 14: Close-up view of the phase boundary intersections in the olivine-plagioclase-silica pseudoternary for an Fe/(Fe+Mg) ratio of 60. Pseudoternary is based on the work of LONGHI and PAN (1988b) and is projected from wollastonite. Eucrites (crosses) can be seen plotting on the peritectic point. Coordinates shown are for the four corners of the diagram and represent the (Olivine, Plagioclase, Silica) intersections. The intersection at the top right of

the diagram is the eutectic point. Solid line is the Orthopyroxene-Plagioclase tie line. The letters denoted within the legend are abbreviations for the natural Allegan-Murchison (AM), natural Murchison (NM), synthetic H-CM (H-CM), and the synthetic Murchison (SM) compositions. The numbers in parentheses following the compositional abbreviations are the temperature (in °C) and oxygen fugacity (relative to the IW buffer) of the experimental run.

Figure 15A and B: Two views of the olivine-pyroxene phase boundary in the olivine-plagioclase-silica pseudoternary for an Fe/(Fe+Mg) ratio of 30 (A) and 55 (B). Pseudoternary plots are based on the work of LONGHI and PAN (1988b) and is projected from wollastonite. Open symbols have an Fe/(Fe+Mg) ratio of 30 (A), while the solid symbols have an Fe/(Fe+Mg) ratio of 55 (B). Two of the experiments (solid circles and solid triangles) in Figure 15B are displaced from the olivine-pyroxene peritectic because of the fractionation of olivine during the quench. The letters denoted within the legend are abbreviations for the natural Allegan-Murchison (AM), natural Murchison (NM), synthetic H-CM (H-CM), and the synthetic Murchison (SM) compositions. The numbers in parentheses following the compositional abbreviations are the temperature (in °C) and oxygen fugacity (relative to the IW buffer) of the experimental run.

Figure 16: Normalization diagrams of the synthetic Murchison and H-CM experimental glasses to bulk average eucrites (Data taken from JAROSEWICH (1990)). The best compositional match to eucrites was produced in the H-CM 1180°C IW-0.5 experiment shown as the solid box. The open diamond (1200°C, IW-1.5) and cross (1200°C, IW-0.5) of the H-CM experiments show the drastic effects of altering the oxygen fugacity.

Figure 17: Fe/Mn vs Fe/(Fe+Mg) (molar) diagram demonstrating the effect of iron loss on the (a) bulk natural Murchison and (b) the natural Allegan (70%) - Murchison (30%) compositions and their resulting glass compositions as a function of temperature and oxygen fugacity. The olivines in equilibrium with the glasses are also shown. The error bars shown on each experiment represent one standard deviation of compositional range. Notice the Fe loss curve for the glasses does not pass through the eucrite range, but is shifted to the magnesian side of eucrites. Symbols shown in both diagrams are the same as in Figure 13 legend.

Figure 18: Normalization of literature experimental glasses to bulk average eucrites (Data taken from JAROSEWICH (1990), JUREWICZ et al. (1993), KUSHIRO and MYSEN, (1979), and STOLPER (1977)). Note the compositional variation in STOLPER'S glasses, which is dependent on what composition is being normalized to, even though he was melting eucrites.

Figure 19: Normalization diagrams of the natural Murchison and natural Allegan (70%) - Murchison (30%) experimental glasses to bulk average eucrites (Data taken from JAROSEWICH (1990)). A comparison with the work of JUREWICZ *et al.* (1993) in the next figure shows good agreement in terms of experimental results.

Figure 20: Fe/Mn vs Fe/(Fe+Mg)(molar) diagram showing how to modify the natural Allegan (70%) - Murchison (30%) composition into a composition similar to synthetic H-CM by using 12% reduction of FeO to Fe metal and 20% olivine fractionation.

Figure 21: Silica-olivine-plagioclase pseudoternary showing the paths taken by the liquids and the bulk composition during eucrite petrogenesis. The equilibrium crystallization path (a) and equilibrium melting path overplot each other but move in the reverse direction. The fractional crystallization path (b) follows the same trend as equilibrium crystallization initially, but moves off the peritectic boundary along the path marked as fractionation of pyroxene, then pyroxene and plagioclase, and eventually pyroxene, plagioclase, and a silica polymorph occurs. The solid evolution path taken during fractional crystallization follows the trend marked (c).

Fractional melting follows the path, marked (d), as liquid is continually removed from the bulk composition, first at the peritectic point and then at the Oliv-SiO $_2$ binary peritectic point.

<u>Table 1. Mixtures Constrained by Oxygen Isotope Ratios to Match Basaltic Achondrite Mass Fractionation</u>

<u>Muoo 1</u>	<u>radironatioi</u>			2	25 Allend-	34 Allend-	30 Mur-	43 Mur-
	Murchison	Allende	LL avg.	H avg.	75 H	66 LL	70 H	57 LL
SiO ₂	29.07	34.23	40.42	36.62	36.02	38.32	34.36	35.54
TiO ₂	0.13	0.15	0.15	0.12	0.13	0.15	0.12	0.14
Al ₂ O ₃	2.15	3.27	2.62	2.13	2.42	2.84	2.14	2.42
Cr2O3	0.48	0.52	0.49	0.51	0.51	0.50	0.50	0.49
Fe ₂ O ₃	0.00	0.00	0.00	0.41	0.30	0.00	0.28	0.00
FeO	22.39	27.15	17.24	10.00	14.29	20.61	13.72	19.46
MnO	0.20	0.18	0.35	0.31	0.28	0.30	0.28	0.29
MgO	19.94	24.62	25.40	23.24	23.59	25.13	22.25	23.05
CaO	1.89	2.61	1.94	1.73	1.95	2.16	1.78	1.92
Na ₂ O	0.24	0.45	0.91	0.85	0.75	0.75	0.67	0.62
K ₂ O	0.04	0.03	0.11	0.09	0.08	0.08	0.08	0.08
P2O5	0.23	0.23	0.22	0.27	0.26	0.22	0.26	0.22
FeS	7.24	4.03	5.33	5.36	5.03	4.89	5.93	6.15
Fe-met	al 0.13	0.17	2.68	15.81	11.90	1.83	11.10	1.58
Fe (tota	1) 22.00	23.66	19.47	27.27	26.37	20.90	25.69	20.56
C	1.85	0.29	0.20	0.11	0.16	0.23	0.63	0.91
Total	85.98	97.93	98.06	97.56	97.67	98.01	94.10	92.87
Fe/Mn	(t) 140.	169.	72.	114.	120.	91.	116.	93.
Fe/(Fe+	. ,	40.86	27.31	19.84	25.40	31.40	25.70	32.00

NOTE: First four columns are from JAROSEWICH (1990). Final four compositions were calculated by DELANEY (1993).

⁽t) Fe/Mn ratio based on total Fe.

Table 2. Chemical Compositions of Experimental Precursors.

A. Nominal Compositions of Murchison and the Allegan-Murchison Cocktail

	Natural <u>//urchison</u> *	70% Natural Allegan- 30% Natural Murchison*
SiO2	33.14	35.53
TiO ₂	0.15	0.11
Al2O3	2.45	2.10
Cr2O3	0.55	0.54
FeO	25.52	14.04
MnO	0.23	0.28
MgO	22.73	22.89
CaO	2.15	1.80
Na ₂ O	0.27	0.67
K2O	0.05	0.08
P2O5	0.26	0.27
NiO	1.99	2.16
Fe-metal	0.15	12.56
FeS	8.25	6.34
С	2.11	0.67
Fe (total)	25.22	27.50
Total	100.00	100.00
Fe/Mn (t)	140	125
Fe/Mn (s)	111	106
Fe/Mn (ms)		54

^{* =} Normalized anhydrous composition (based on analyses from JAROSEWICH (1990)).

⁽t) = Fe/Mn ratio based on total Fe

⁽s) = Fe/Mn ratio based on FeS free composition, but includes Fe metal.

⁽ms) = Fe/Mn ratio based on FeS and Fe metal free composition.

Table 2. (continued) B. Murchison Analogue Compositions for the Series I Experiments

Fe/Mn ra	tio 85*	65*	45*	20*	Nominal Murchison**
SiO2	37.65	38.06	37.48	38.32	35.37
Al2O3	2.88	2.76	2.91	2.63	2.62
Cr2O3	0.52	0.51	0.48	0.49	0.58
FeO	34.51	34.08	33.84	32.86	34.63
MnO	0.40	.52	0.71	1.63	0.24
MgO	22.82	21.98	22.12	22.45	24.26
CaO	2.49	2.61	2.19	2.41	2.30
Total	100.77	100.52	99.73	100.79	100.00

^{*} Average microprobe analyses of the four Murchison analogue compositions.
** Murchison normalized for the seven oxides contained within the compositions. NOTE: Nominal Murchison has an Fe/Mn ratio of 142 (total Fe)[based on analyses from JAROSEWICH (1990)].

Table 2. (continued)

C. Nominal and Synthetic Compositions for the Series II Experiments

	Nominal Synthetic <u>Murchison</u> *	Experimental Synthetic <u>Murchison</u>	Nominal Synthetic <u>H-CM</u> **	Experimental Synthetic <u>H-CM</u>
SiO2	37.18	40.19	44.64	46.04
TiO2	0.17	0.25	0.16	0.12
Al2O3	2.75	3.31	2.81	3.01
Cr2O3	0.61	0.65	0.67	0.82
FeO	28.85(s)	27.27(s)	19.10	19.90
MnO	0.26	0.38	0.35	0.53
MgO	25.50	22.79	29.06	25.50
CaO	2.42	2.91	2.31	2.59
Na ₂ O	0.31	0.30	0.84	0.99
NiO	1.95	1.91	ni	ni
Total	100.00	99.96	100.00	99.50
Fe/Mn	111	71	54	37

- * = Normalized anhydrous Murchison
- ** = Normalized silicate portion of the nominal H-CM (70% Allegan-30% Murchison) composition in Table 2A. Metal and sulfide portions were excluded from this composition.
- ni = Not included in the composition
- (s) = This composition excludes FeS, but includes Fe-metal as FeO. Therefore the target Fe/Mn ratio was 111, as opposed to the nominal (total Fe, including FeS) ratio of 142. NOTE: Nominal compositions are normalized from JAROSEWICH (1990), while the experimental compositions were determined by microprobe analysis.

Table 3. Compositional Starting Materials and their Purity Levels

Starting Material	Formula	Purity
Silicon (IV) dioxide powder	SiO2	99.999%
Titanium (IV) dioxide powder	TiO ₂	99.99%
Aluminum oxide powder	Al2O3	99.5%
Chromium (III) oxide powder	Cr2O3	99.997%
Hematite powder	Fe ₂ O ₃	97%
Manganese (II) oxide powder	MnO	99.5%
Magnesium oxide	MgO	99.5%
Calcium carbonate powder	CaCO3	99.95%
Sodium hydrogen carbonate	NaHCO3	~95%

Table 4. Temperature (°C) Correlation Table Between the Sample Rod Thermocouple (Tsr) and Furnace (Dfurn) Display Reading. The sample rod thermocouple shows the nominal temperature, based on the calibration to the melting points of Au and Pd.

Tsr	Dfurn	Tsr	Dfurn	Tsr	Dfurn
1050	1002	1240	1211	1430	1421
1060	1013	1250	1223	1440	1432
1070	1024	1260	1234	1450	1444
1080	1035	1270	1245	1460	1455
1090	1046	1280	1256	1470	1466
1100	1057	1290	1267	1480	1477
1110	1068	1300	1278	1490	1488
1120	1079	1310	1289	1500	1499
1130	1090	1320	1300	1510	1510
1140	1101	1330	1311	1520	1521
1150	1112	1340	1322	1530	1532
1160	1123	1350	1333	1540	1543
1170	1134	1360	1344	1550	1554
1180	1145	1370	1355	1560	1565
1190	1156	1380	1366	1570	1576
1200	1167	1390	1377	1580	1587
1210	1178	1400	1388	1590	1598
1220	1189	1410	1399	1600	1609
1230	1200	1420	1410		

Table 5. Results of the Thermocouple Calibration Experiments (Temp in °C)

Exp.	#	<u>#1</u>		<u>#2</u>		<u>#3</u>	
	Tsr	Dfurn	Tsr	Dfurn	Tsr	Dfurn	<u>Nominal</u>
Au	1034	1018	1061	1017	1064	1017	1064
Pd	1532	1556	1537	1555	1554	1555	1554

Tsr = sample rod thermocouple temperature
Dfurn = furnace display reading

Table 6. Initial Calibration of the Furnace Using the Oxygen Fugacity Sensor

Temp°C	;	Scal	е	m	/min	e.m.f.	IW
Tofs	Dfurn	co	CO ₂	CO	CO ₂	calc	<u>buffer</u>
1350	1405	32	123	60.4	17.43	-780	+0.5
1350	1405	63	123	100.9	17.43	-820	IW
1350	1405	75	84	117.0	9.98	-861	-0.5
1350	1405	75	46	117.0	4.72	-901	-1.0
1350	1405	75	17	117.0	1.89	-942	-1.5
1250	1298	24	123	50.2	17.43	-816	0.5
1250	1298	51.5	123	85.1	17.43	-853	IW
1250	1298	80	100	124.0	12.80	-891	-0.5
1250	1298	80	64	124.0	6.96	-929	-1.0
1250	1298	80	26	124.0	2.70	-967	-1.5
1150	1183	17	119	41.4	16.59	-850	+0.5
1150	1183	45	119	76.5	16.59	-886	IW
1150	1183	75	107	117.0	14.13	-921	-0.5
1150	1183	75	60	117.0	6.40	-957	-1.0
1150	1183	75	20	117.0	2.10	-992	-1.5

Scale = arbitrary and unitless flowmeter reading

Tofs = Oxygen fugacity sensor themocouple temperature
Dfurn = Furnace display reading

NOTE: The e.m.f. (calculated) is measured in millivolts and the IW buffer is measured in log units relative to the buffer.

Table 7. Testing for a Leak in the Furnace

Leak Test: Part 1 (ratio of CO:CO2 was 10:1)

Scale		ml/ı	min	EMF			
CO	CO ₂	CO	CO ₂	(obs)	$\Delta \log f(O_2)$		
10	32	33	3.3	-929	+0.52		
40	65	70	7.0	-930	+0.51		
60	82	97	9.7	-936	+0.42		
80	98	124	2.4	-939	+0.38		

The calculated e.m.f. value at these settings is -966 millivolts. Operating temperature: 1150°C

Leak Test: Part 2 (ratio of CO:CO2 was 20:1)

Scale		ml/ı	min	EMF			
CO	CO ₂	CO	CO ₂	(obs)	$\Delta \log f(O_2)$		
20	21.5	45	2.3	-957	+0.72		
50	41	83	4.2	-965	+0.61		
75	55.5	117	5.9	-969	+0.55		

The calculated e.m.f. value at these settings is -1008 millivolts. Operating temperature: 1150°C

Table 8. Calibration Using the Sample Rod Thermocouple and the Oxygen Fugacity Sensor

Temp	°C	Sc	ale	m	l/min	EMF	IW
Tsr	Dfurn	CO	CO2	CO	CO2	(calc)	buffer
1350	1333	15	121	39.0	17.01	-780	+0.5
1350	1333	50	123	83.0	17.43	-821	IW
1350	1333	75	95	117.0	11.90	-861	-0.5
1350	1333	75	50	117.0	5.20	-901	-1.0
1350	1333	75	13	117.0	1.61	-942	-1.5
1250	1220	10	122	33.0	17.22	-815	+0.5
1250	1220	45	123	76.5	17.43	-854	IW
1250	1220	75	113	117.0	15.33	-891	-0.5
1250	1220	75	69	117.0	7.66	-929	-1.0
1250	1220	75	32	117.0	3.28	-967	-1.5
					40.40	950	±0 E
1150	1109	10	128	33.0	18.48	-850	+0.5
1150	1109	45	131	76.5	19.13	-886	IW
1150	1109	75	116	117.0	15.96	-921	-0.5
1150	1109	75	72	117.0	8.10	-957	-1.0
1150	1109	75	35	117.0	3.55	-992	-1.5

Table 9. Fe Foil Calibration of Iron-Wüstite Buffer

Temp°C		Sc	ale	EMF	ml/min		EMF	
Tsr	Dfurn	CO	CO ₂	(obs)	CO	CO ₂	(calc)	
1350	1333	50	123	-817	83.0	17.43	-821	
1250	1220	45	123	-850	76.5	17.43	-854	
1150	1109	45	131	-882	76.5	19.13	-886	

Table 10. Electron Microprobe Operating Conditions and Standards

El	Crystal	Bkg(1)	Bkg(2)	CF	Bsin	Win	Bias	Standard
		<u>(mm)</u>	<u>(mm)</u>		(volts)	(volts)	(volts)	
Si	TAP	4.00	-4.00	0.179	0.5	2.0	1700	Plagioclase, Lake County
Ti	PET	1.50	-1.50	0.024	1.5	2.0	1600	Hornblende, Kakanui
ΑI	TAP	4.00	-5.00	0.144	0.5	2.0	1700	Anorthite, Great Sitkin Island
Cr	LIF	1.50	-1.50	0.385	8.0	2.0	1550	Chromite, Tiebaghi Mine
Fe	LIF	1.50	-1.50	0.503	8.0	2.0	1500	Fayalite (Fa 100)
Mn	LIF	4.00	-4.00	0.494	0.8	2.0	1550	Tephroite
Mg	TAP	4.00	-5.00	0.255	0.3	2.0	1700	Forsterite (Fo 100)
Ca	PET	1.50	-1.50	0.125	1.1	2.0	1600	Anorthite, Great Sitkin Island
Na	TAP	4.00	-4.00	0.014	0.4	1.5	1700	Plagioclase, Lake County
Ni	LIF	1.50	-1.50	0.003	0.5	1.0	1500	Olivine, San Carlos

El=element Bkg=Backgrounds CF=Correction factor Bsln=Baseline Win=Window

Table 11. Additional Calibration Standards

- Anorthoclase, Kakanui Hypersthene, Johnstown meteorite
 Augite, Kakanui Olivine, Springwater meteorite

- Fayalite, Rockport
- Orthopyroxene, Tatahouine meteorite

Table 12. Experimental Parameters and Resulting Phase Assemblages

A. Series I Experiments

Synthetic Murchison Fe/Mn 85

Temp (°C)	Time (hrs)	Type	Exp#	<u>Phases</u>
1180	40.5	cooling	M 09	glss, oliv, pig, spin, mtl
1180	39	heating	M 13	glss, oliv, pig, spin, mtl
1253	47	heating	M 03	glss, oliv, pig, spin, mtl
1300	21	heating	M 04	glss, oliv, spin, mtl
1325	16.5	cooling	M 08	glss, oliv, mtl
1580	1	heating	M 17	glss
1580	10	heating	M 19	glss

Synthetic Murchison Fe/Mn 65

Temp (°C	C) Time (hrs)	Type	Exp#	<u>Phases</u>
1180	40.5	cooling	M 10	glss, oliv, pig, spin, mtl
1180	39	heating	M 14	glss, oliv, pig, spin, mtl
1325	16.5	cooling	M 07	glss, oliv, mtl

Synthetic Murchison Fe/Mn 45

Temp (°C)	Time (hrs)	Type	Exp#	<u>Phases</u>
1180	40.5	cooling	M 11	glss, oliv, pig, spin, mtl
1325	16.5	cooling	M 06	glss, oliv, mtl

Synthetic Murchison Fe/Mn 20

Temp (°C	C) Time (hrs)	Type	Exp#	<u>Phases</u>
1180	40.5	cooling	M 12	glss, oliv, pig, spin, mtl
1325	16.5	cooling	M 05	glss, oliv, mtl

Table 12. (continued)

B. Series II Experiments

Temp (°C)	Time (hrs)	Type	IW	Exp#	<u>Phases</u>
1180	135.5	heating	-0.5	M 29	glss, oliv, pig, spin, mtl
1180	165	cooling	-0.5	M 35	glss, oliv, pig, spin, mtl
1180	144.5	heating	-1.0	M 61	glss, oliv, pig, spin, mtl
1180	142	heating	-1.5	M 30	glss, oliv, pig, spin, mtl
1180	164	heating	-1.5	M 47	glss, oliv, pig, spin, mtl
1200	66.5	heating	-0.5	M 26	glss, oliv, pig, spin, mtl
1200	161	cooling	-0.5	M 41	glss, oliv, pig, spin, mtl
1200	69	heating	-1.5	M 27	glss, oliv, pig, spin, mtl
1200	165	cooling	-1.5	M 39	glss, oliv, pig, spin, mtl
1300	112.5	heating	-0.5	M 43	glss, oliv, mtl
1300	118	heating	-1.5	M 45	glss, oliv, pig, mtl
1400	116	heating	-0.5	M 49	glss, oliv, mtl
1400	91.5	heating	-1.5	M 51	glss, oliv, pig, mtl

Synthetic H-CM

Temp (°C)	Time (hrs)	Type	IW	Exp#	<u>Phases</u>
1180	144.5	heating	-0.5	H 33	glss, oliv, pig, spin, mtl
1180	165	cooling	-0.5	H 34	glss, oliv, pig, spin, mtl
1180	144.5	heating	-1.0	H 60	glss, oliv, pig, spin, mtl
1180	164	heating	-1.5	H 46	glss, oliv, pig, spin, mtl
1200	144	heating	-0.5	H 31	glss, oliv, pig, spin, mtl
1200	161	cooling	-0.5	H 40	glss, oliv, pig, spin, mtl
1200	139	heating	-1.5	H 32	glss, oliv, pig, spin, mtl
1300	112.5	heating	-0.5	H 42	glss, oliv, pig, mtl
1300	118	heating	-1.5	H 44	glss, oliv, pig, mtl
1400	116	heating	-0.5	H 48	glss, oliv, mtl
1400	91.5	heating	-1.5	H 50	glss, oliv, pig, mtl

C. Series III Experiments

Natural Murchison

Time (hrs)	Type	IW	Exp#	
166	heating	-0.5	NM 55	glss, oliv, spin, mtl
166.5	heating	-1.0	NM 59	glss, oliv, pig, spin, mtl
169.5	heating	-1.5	NM 57	glss, oliv, pig, spin, mtl
67.5	heating	-1.0	NM 63	glss, oliv, mtl
	Time (hrs) 166 166.5 169.5	Time (hrs) Type 166 heating 166.5 heating 169.5 heating	Time (hrs) Type IW 166 heating -0.5 166.5 heating -1.0 169.5 heating -1.5	Time (hrs) Type IW Exp# 166 heating -0.5 NM 55 166.5 heating -1.0 NM 59 169.5 heating -1.5 NM 57

70% Allegan-30% Murchison

Temp (°C)	Time (hrs)	Type	IW	Exp#	
1180	166	heating	-0.5	AM 54	glss, oliv, pig, spin, mtl
1180	166.5	heating	-1.0	AM 58	glss, oliv, pig, spin, mtl
1180	169.5	heating	-1.5	AM 56	glss, oliv, pig, spin, mtl
1400	67.5	heating	-1.0	AM 62	glss, oliv, mtl

Table 13. Experimental Silicate Compositions

Series II Experiments

A. Synthetic Murchison

A. Synthetic Murchison										
			-0.5		<u>IW -1.0</u>		IW -			Eucrite
T(°C)	1180	1200	1300	1400	1180	1180	1200	1300	1400	Avg.*
Glass					=====	45.00	50.40	F2 F2	E4 00	49.15
SiO2	50.18	49.58	47.92	48.53	50.53	45.20	52.43	53.52	54.89	
TiO2	0.79	0.70	0.53	0.44	0.85	0.11	0.95	0.73	0.51	0.64
AI2O3	11.15	9.80	6.94	6.34	12.69	31.68	13.82	10.05	7.48	12.74
Cr2O3	0.28	0.33	0.77	0.87	0.41	0.28	0.41	0.83	0.91	0.34
FeO	19.71	21.54	24.94	25.54	16.07	1.93	11.68	12.32	6.17	17.96
MnO	0.32	0.36	0.60	0.65	0.40	0.11	0.50	0.73	0.72	0.52
MgO	6.05	7.07	11.31	12.09	6.31	2.60	8.42	12.39	20.97	6.96
CaO	10.25	9.50	6.13	6.31	10.16	16.84	10.63	9.22	7.17	10.28
Na2O	0.35	0.42	0.11	0.02	0.83	0.41	0.57	0.04	0.02	0.39
NiO	0.04	80.0	0.05	0.04	0.06	0.06	0.04	0.05	0.03	0.01
Total	99.08	99.39	99.30	100.83	98.31	99.17	99.47	99.88	98.90	98.99
Fe/Mn	60.93	59.82	41.11	38.90	40.95	16.71	22.86	16.73	8.46	34.25
Fe/(Fe+Mg)x100	64.15	62.44	54.60	53.49	58.02	29.92	42.88	35.18	13.97	59.10
Olivine					07.04	40.00	20.20	40.10	41.57	
SiO2	36.98	37.16	37.85	38.34	37.24	40.20	39.32	40.10		
TiO2	0.03	0.00	0.01	0.02	0.01	0.09	0.03	0.01	0.03	
Al2O3	0.05	0.06	0.04	0.04	0.05	0.23	0.08	0.08	0.02	
Cr2O3	0.27	0.33	0.56	0.46	0.38	0.77	0.48	0.61	0.53	
FeO	31.42	31.20	27.17	20.79	26.64	12.68	19.13	13.33	5.41	
MnO	0.40	0.43	0.52	0.43	0.45	0.52	0.63	0.60	0.47	
MgO	30.44	31.03	34.39	40.20	34.39	45.36	40.73	45.17	51.98	
CaO	0.34	0.33	0.21	0.16	0.29	0.29	0.29	0.25	0.12	
Na2O	0.01	0.00	0.00	0.01	0.02	0.01	0.00	0.00	0.02	
NiO	0.18	0.15	0.12	0.05	0.09	0.04	0.02	0.04	0.00	
Total	100.14	100.72	100.88	100.50	99.57	100.20	100.71	100.20	100.09	
Fe/Mn	77.21	72.88	51.55	48.11	59.01	24.28	29.93	21.91	11.37	
Fe/(Fe+Mg)x100	36.50	35.89	30.52	22.38	30.13	13.49	20.71	14.11	5.49	
Low Calcium Pyr)		53.35	56.91	55.53	57.00	56.94	
SiO2	54.06	53.89					0.15	0.07	0.03	
TiO2	0.08	0.07			0.09	0.07		0.49	0.03	
Al2O3	0.67	0.47			0.80	0.10	0.87	0.49	0.52	
Cr2O3	0.57	0.60			0.73	0.74	0.84		3.86	
FeO	18.90	18.94			16.51	9.37	14.48	9.32	0.32	
MnO	0.37	0.36			0.43	1.49	0.53 26.28	0.51	36.18	
MgO	22.56	23.45			24.20	27.58		31.52	0.25	
CaO	2.80	2.40			2.63	4.40	2.30	0.73		
Na2O	0.02	0.03			0.03	0.02	0.02	0.00	0.01	
NiO	0.11	0.07			0.04	0.00	0.02	0.04	0.06	
Total	100.15	100.29			98.81	100.67	101.02	100.48	98.43	
Fe/Mn	51.64	51.86			37.91	6.21	27.10	18.19	11.92	
Fe/(Fe+Mg)x100	31.78	31.00	÷		27.48	15.61	23.41	14.11	5.62	

Table 13. Series II Experiments (continued)

B. Synthetic H-CM

B. Synthetic I	T-CIVI	na.	-0.5		<u>IW -1.0</u>		IW	-1.5		Eucrite
T(°C)	1180	1200	1300	1400	1180	1180	1200	1300	1400	Avq.*
Glass	1100	1200	1000	1400						
SiO2	51.27	51.00	52.30	53.31	51.25	50.16	55.69	53.07	54.86	49.15
TiO2	0.54	0.58	0.42	0.19	0.60	0.69	0.57	0.47	0.34	0.64
Al203	12.35	12.97	8.32	4.09	13.86	17.28	14.95	9.87	8.60	12.74
Cr2O3	0.41	0.25	0.75	0.89	0.40	0.41	0.48	0.91	0.94	0.34
FeO	16.70	16.43	20.99	18.08	14.31	8.86	8.75	13.02	3.22	17.96
MnO	0.53	0.53	0.72	0.62	0.37	0.43	0.46	0.71	0.68	0.52
MgO	7.64	7.12	8.38	19.38	6.42	8.39	8.32	12.83	21.71	6.96
CaO	8.98	9.14	8.04	3.67	9.32	11.67	10.90	8.79	7.78	10.28
Na2O	0.54	0.48	0.16	0.04	0.99	0.35	0.26	0.05	0.02	0.39
Total	98.98	98.49	100.07	100.33	97.54	98.24	100.37	99.71	98.18	98.98
Fe/Mn	31.12	31.04	28.95	28.97	38.15	20.40	19.42	18.15	4.65	34.25
Fe/(Fe+Mg)x100	54.13	55.48	57.42	33.95	54.75	36.53	36.59	35.73	7.57	59.10
Olivine										
SiO2	38.13	38.04	39.19	39.71	37.42	39.28	40.01	40.06	40.16	
TiO2	0.02	0.01	0.00	0.01	0.02	0.04	0.02	0.01	0.01	
Al2O3	0.05	0.03	0.04	0.03	0.03	0.04	0.05	0.04	0.03	
Cr2O3	0.30	0.40	0.66	0.53	0.36	0.43	0.50	0.68	0.58	
FeO	26.29	24.55	21.55	15.51	24.57	14.45	13.73	14.26	3.14	
MnO	0.47	0.58	0.56	0.39	0.46	0.50	0.57	0.60	0.52	
MgO	35.25	35.83	38.88	43.97	35.82	45.36	45.46	45.05	53.39	
CaO	0.24	0.22	0.18	0.07	0.25	0.23	0.22	0.21	0.11	
Na2O	0.00	0.01	0.00	0.00	0.01	0.00	0.00	0.00	0.01	
Total	100.74	99.67	101.07	100.26	98.95	100.35	100.55	100.91	97.94	
Fe/Mn	55.54	41.97	37.79	38.93	53.87	28.25	24.09	23.48	5.96	
Fe/(Fe+Mg)x100	29.42	27.58	23.57	16.45	27.64	15.08	14.40	14.98	3.18	
Low Calcium Pyr	oxene (p	igeonite)							
SiO2	54.89	54.81	56.27		53.05	54.56	56.59	56.74	57.15	
TiO2	0.07	0.06	0.03		0.08	0.09	0.10	0.05	0.06	
Al2O3	0.72	0.61	0.42		0.68	0.68	1.04	0.50	0.22	
Cr2O3	0.72	0.63	0.87		0.66	0.78	0.98	0.89	0.63	
FeO	17.55	15.75	13.27		15.25	10.22	9.57	10.62	5.28	
MnO	0.43	0.57	0.47		0.44	0.48	0.58	0.49	0.35	
MgO	24.33	25.49	22.89		26.02	29.00	30.31	30.70	34.68	
CaO	2.19	1.81	0.54		2.07	2.61	1.86	0.66	0.21	
Na2O	0.01	0.03	0.01		0.04	0.03	0.03	0.00	0.00	
Total	100.91	99.77	100.78		98.29	98.44	101.07	100.65	98.74	
Fe/Mn	40.12	27.96	28.05		36.32	21.17	16.31	21.68	15.19	
Fe/(Fe+Mg)x100	28.63	25.50	20.34		24.58	16.34	14.92	16.14	7.84	

Table 13. (continued)

Series III Experiments

C.	Natu	ral	Mur	chison
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C. <u>Natural Mu</u>	rchisor	<u>1</u>			
T(°C)	1180	1180	1180	1400	Eucrite
IW-X	-0.5	-1.0	-1.5	-0.5	Avq.*
Glass					
SiO2	43.45	48.29	52.45	53.96	49.15
TiO2	0.82	0.62	0.59	0.35	0.64
Al2O3	14.65	13.32	14.56	7.47	12.74
Cr2O3	0.14	0.37	0.40	0.73	0.34
FeO	20.34	17.60	11.76	13.33	17.96
MnO	0.23	0.23	0.29	0.74	0.52
MgO	3.66	5.46	7.07	16.08	6.96
CaO	12.97	10.78	10.16	5.69	10.28
Na2O	0.44	0.75	0.87	0.05	0.39
NiO	0.10	0.09	0.06	0.02	0.01
Total	96.82	97.54	98.22	98.42	98.99
Fe/Mn	87.32	75.55	40.04	17.79	34.25
Fe/(Fe+Mg)x100	75.06	63.85	47.69	31.19	59.10
Olivine					
SiO2	35.22	37.51	38.54	40.11	
TiO2	0.01	0.01	0.02	0.01	
Al2O3	0.13	0.10	0.06	0.07	
Cr2O3	0.18	0.33	0.46	0.45	
FeO	33.22	28.79	19.85	11.45	
MnO	0.32	0.32	0.34	0.47	
MgO	28.86	33.12	40.33	47.12	
CaO	0.52	0.33	0.25	0.14	
Na2O	0.01	0.01	0.00	0.01	
NiO	0.17	0.15	0.06	0.05	
Total	98.64	100.67	99.93	99.88	
Fe/Mn	102.50	88.83	57.64	24.05	
Fe/(Fe+Mg)x100	39.09	32.66	21.56	11.94	
Low Calcium Pyr	oxene (pi	geonite)			
SiO2		52.69	54.80		
TiO2		0.07	0.10		
Al2O3		0.89	0.96		
Cr2O3		0.57	0.69		
FeO		18.62	13.22		
MnO		0.32	0.36		
MgO		24.47	27.28		
CaO		1.90	2.18		
Na2O		0.03	0.04		
NiO		0.06	0.06		
Total		99.62	99.68		
Fe/Mn		57.45	37.13		
Fe/(Fe+Mg)x100		29.76	21.26		

Table 13. Series III Experiments (continued)

<u>Table 13. Series III Experiments (continued)</u>							
D. Allegan (70%) - Murchison (30%)							
T(°C)	1180	1180	1180	1400	Eucrite		
IW-X	-0.5	-1.0	-1.5	-0.5	Avq.*		
Glass							
SiO2	52.43	51.05	52.68	53.73	49.15		
TiO2	0.50	0.54	0.59	0.33	0.64		
AI2O3	10.67	12.07	15.57	7.23	12.74		
Cr2O3	0.29	0.31	0.35	0.76	0.34		
FeO	17.78	18.21	9.77	19.76	17.96		
MnO	0.26	0.31	0.30	0.63	0.52		
MgO	5.68	5.11	7.76	9.66	6.96		
CaO	8.31	8.98	9.79	5.68	10.28		
Na2O	1.21	1.08	0.96	0.05	0.39		
NiO	0.05	0.12	0.02	0.06	0.01		
Total	97.17	97.78	97.79	97.89	98.99		
Fe/Mn	67.52	58.00	32.16	30.97	34.25		
Fe/(Fe+Mg)x100	63.12	65.90	40.87	52.53	59.10		
Olivine							
SiO2	37.24	36.99	39.64	38.82			
TiO2	0.02	0.02	0.01	0.01			
Al2O3	0.05	0.07	0.07	0.04			
Cr2O3	0.31	0.48	0.39	0.47			
FeO	29.20	29.10	16.37	17.28			
MnO	0.33	0.38	0.40	0.39			
MgO	32.37	32.64	43.35	42.43			
CaO	0.27	0.30	0.23	0.12			
Na2O	0.01	0.02	0.00	0.00			
NiO	0.21	0.04	0.07	0.15			
Total	100.01	100.03	100.55	99.73			
Fe/Mn	87.37	75.61	40.41	43.75			
Fe/(Fe+Mg)x100	33.47	33.19	17.41	18.52			
Low Calcium Pyr	oxene (pi	geonite)					
SiO2	53.59	53.41	55.65				
TiO2	0.07	0.06	0.10				
Al2O3	0.73	0.80	0.83				
Cr203	0.65	0.78	0.79				
FeO	17.85	17.88	12.80				
MnO	0.32	0.31	0.37				
MgO	24.20	24.48	28.29				
CaO	1.75	1.76	1.64				
Na2O	0.03	0.03	0.04				
NiO	0.13	0.05	0.03				
Total	99.32	99.58	100.55				

^{*} Eucrite Avg. based on analyses from JAROSEWICH (1990)

34.64

20.14

58.79

28.91

54.80

29.11

Fe/Mn

Fe/(Fe+Mg)x100

Table 14. Mineral Normative Calculations of the Experimental Glass Compositions

Synthetic Murchison

		[W-0.5			<u>IW-1.0</u>		IW-1.5			Eucrite
Temp°C	1180	1200	1300	1400	1180	1180	1200	1300	1400	Avg.*
Albite	3.34	4.00	1.04	0.19	7.88	3.71	5.26	0.37	0.18	
Anorthite	30.67	26.37	19.50	17.91	32.67	84.10	36.15	27.77	20.03	
Plagioclase	34.01	30.37	20.55	18.10	40.55	87.81	41.41	28.14	20.20	43.3±2.2
Clinopyrox.	18.71	18.91	10.12	11.73	16.50	0.00	14.47	15.09	12.02	
Orthopyrox.	39.79	45.71	65.93	65.39	35.27	9.84	34.11	45.68	59.57	
Pyroxene	58.50	64.62	76.05	77.12	51.76	9.84	48.58	60.77	71.60	51.3±2.5
Ilmenite	1.17	1.03	0.78	0.64	1.25	0.15	1.36	1.04	0.70	0.8±0.4
Chromite	0.44	0.51	1.19	1.33	0.63	0.41	0.62	1.24	1.31	0.2±0.2
Quartz	5.88	3.46	0.00	0.00	5.80	1.78	8.03	8.82	6.19	4.1±1.5
Olivine	0.00	0.00	1.43	2.82	0.00	0.00	0.00	0.00	0.00	

Synthetic H-CM

		IW-0.5			<u>IW-1.0</u>		IW-1.5			
Temp°C	1180	1200	1300	1400	1180	1180	1200	1300	1400	Avg.*
Albite	5.08	4.54	1.51	0.36	9.40	3.24	2.37	0.46	0.18	
Anorthite	32.78	35.03	23.14	11.03	35.29	46.95	40.21	27.23	22.95	
Plagioclase	37.86	39.57	24.65	11.39	44.69	50.18	42.58	27.69	23.12	43.3±2.2
Clinopyrox.	11.13	10.21	15.07	5.81	10.88	10.15	11.73	13.78	11.95	
Orthopyrox.	43.11	42.08	50.45	78.36	35.86	32.01	30.41	48.90	56.99	
Pyroxene	54.24	52.29	65.52	84.17	46.74	42.16	42.14	62.68	68.94	51.2±2.5
Ilmenite	0.79	0.85	0.62	0.27	0.88	0.99	0.81	0.67	0.46	0.8±0.4
Chromite	0.63	0.39	1.16	1.31	0.62	0.62	0.71	1.36	1.35	0.2±0.2
Quartz	6.49	6.90	8.05	2.87	7.07	6.05	13.76	7.60	6.12	4.1±1.5
Olivine	0.00	0.00	0.00	0.00	0.00	0.00	0.00	0.00	0.00	

Table 14. (continued)

Nint:		NA: 11	rrhic	20
14.111	11 41	171111	rchis	

Temp°C	1180	1180	1180	1400	Eucrite
IW-x	-0.5	-1.0	-1.5	-0.5	Avq.*
Albite	4.31	7.21	8.14	0.46	
Anorthite	41.51	35.33	37.35	20.62	
Plagioclase	45.82	42.54	45.54	21.08	43.3±2.2
Clinopyrox.	23.02	17.57	12.16	6.60	
Orthopyrox.	20.15	35.45	32.27	62.84	
Pyroxene	43.17	53.02	44.43	69.44	51.2±2.5
Ilmenite	1.25	0.92	0.86	0.50	0.8±0.4
Chromite	0.22	0.58	0.61	1.09	0.2±0.2
Quartz	0.00	2.94	8.60	7.88	4.1±1.5
Olivine	9.54	0.00	0.00	0.00	

Allegan (70%) - Murchison (30%)

Temp°C	1180	1180	1180	1400	Eucrite
IW-x	-0.5	-1.0	-1.5	-0.5	Avg.*
Albite	11.68	10.39	8.94	0.48	
Anorthite	25.47	30.10	39.60	20.90	
Plagioclase	37.15	40.49	48.54	21.39	43.3±2.2
Clinopyrox.	15.09	14.12	8.62	7.44	
Orthopyrox.	38.17	37.52	32.72	57.05	
Pyroxene	53.26	51.64	41.34	64.49	51.2±2.5
Ilmenite	0.75	0.81	0.85	0.49	0.8±0.4
Chromite	0.46	0.49	0.53	1.49	0.2±0.2
Quartz	8.39	6.58	8.74	12.44	4.1±1.5
Olivine	0.00	0.00	0.00	0.00	

^{*} from DELANEY et al (1984)

Table 15. Modal Analysis of the Series III Experiments

Natura	l Murc	hison

Temp°C	1180	1180	1180	1400
IW-x	-0.5	-1.0	-1.5	-0.5
Glass	19	19	19	38
Olivine	76	72	57	42
Pyroxene	0	1	9	0
Fe-metal	5	8	16	20
Chromite	<1	<1	<1	0

<u>Allegan (70%) - Murchison (30%)</u> Temp°C 1180 1180 1180 1400

Temp°C	1180	1180	1180	1400	
IW-x	-0.5	-1.0	-1.5	-0.5	
Glass	19	18	17	35	
Olivine	64	57	46	48	
Pyroxene	6	12	16	0	
Fe-metal	10	12	20	17	
Chromite	<1	<1	<1	0	

Note: Fe-metal includes all FeS.

Figure 1.

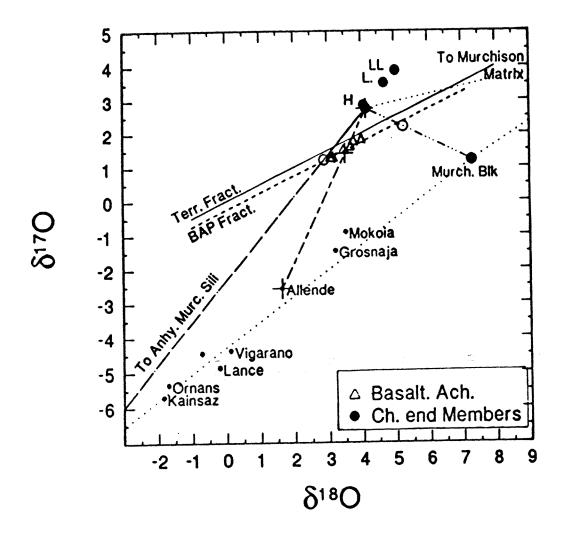


Figure 2.

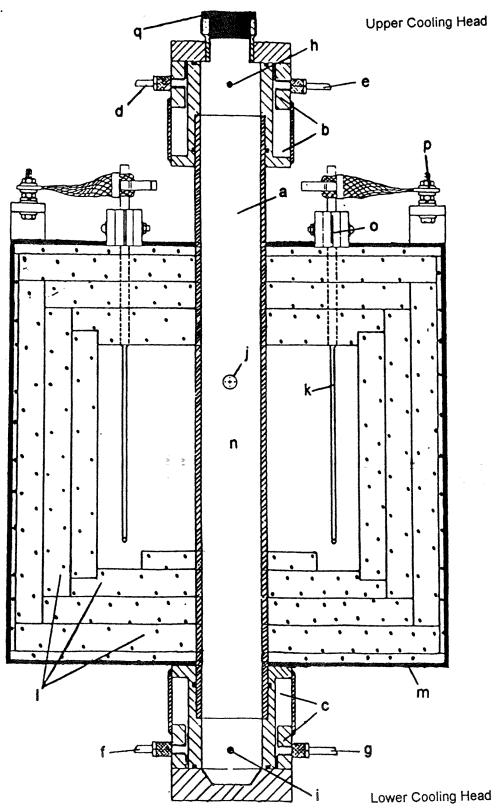


Figure 3.

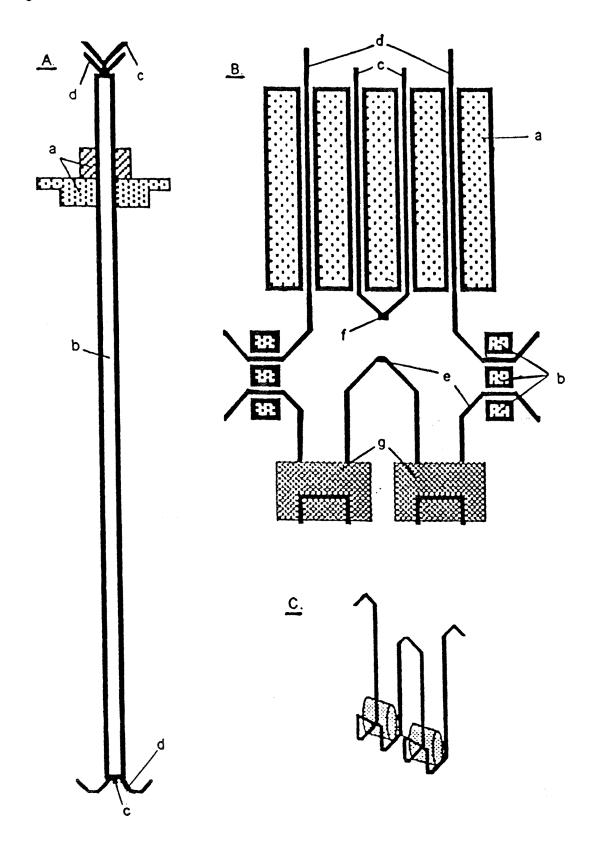


Figure 4.

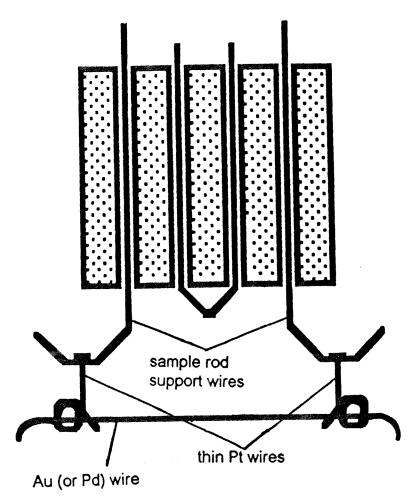


Figure 5.

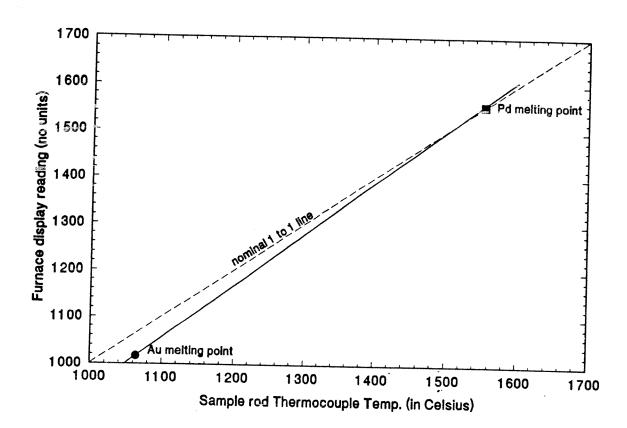


Figure 6.

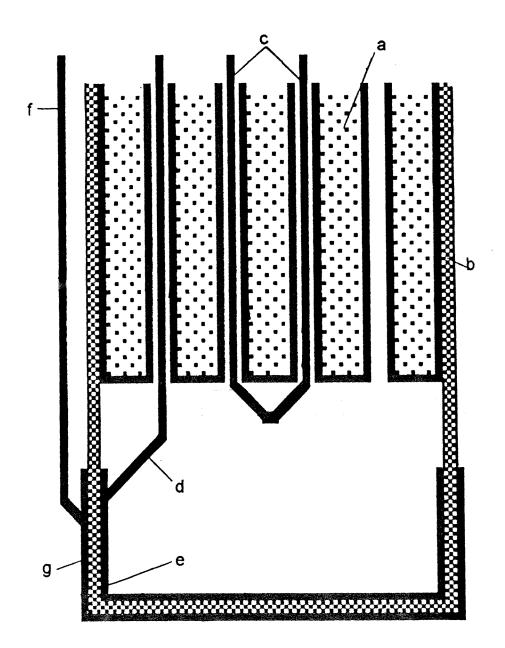


Figure 7A.

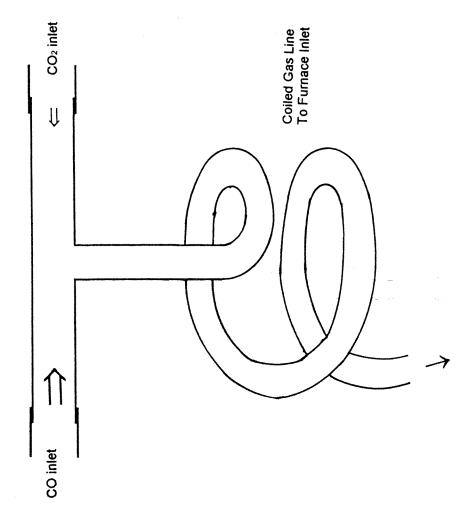


Figure 7B.

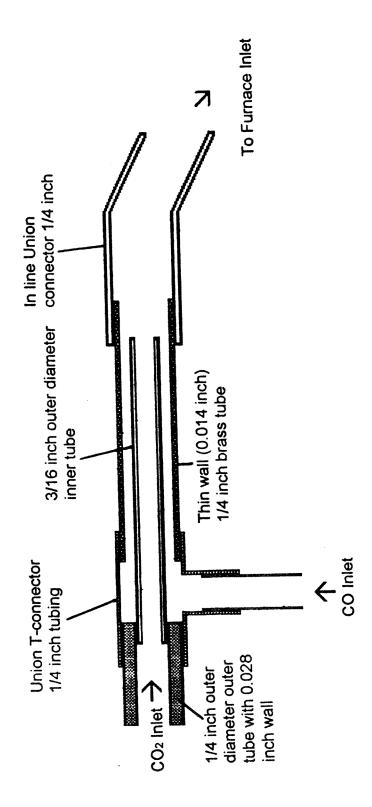


Figure 8.

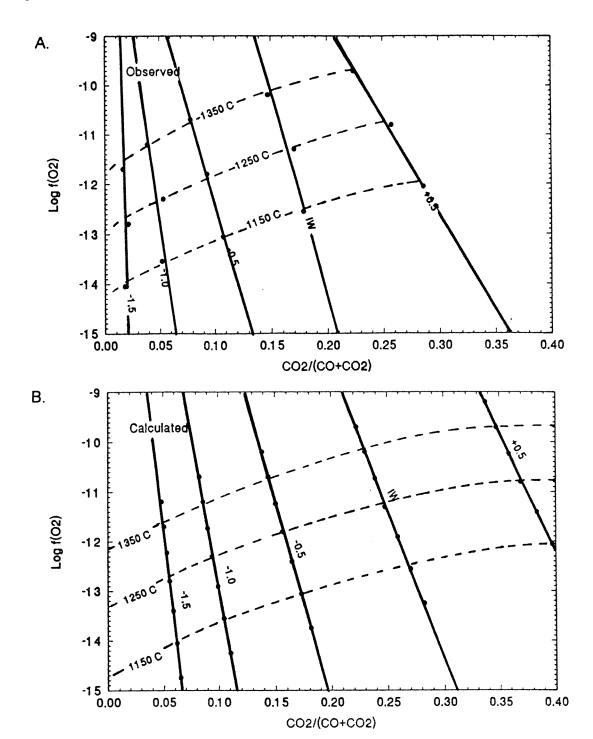
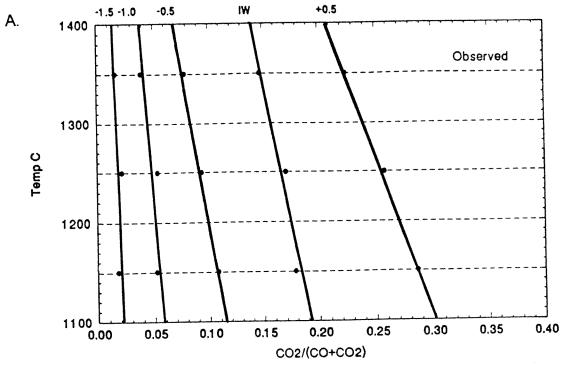


Figure 9.



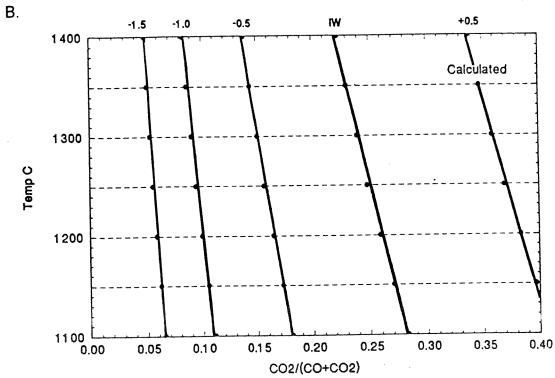


Figure 10.

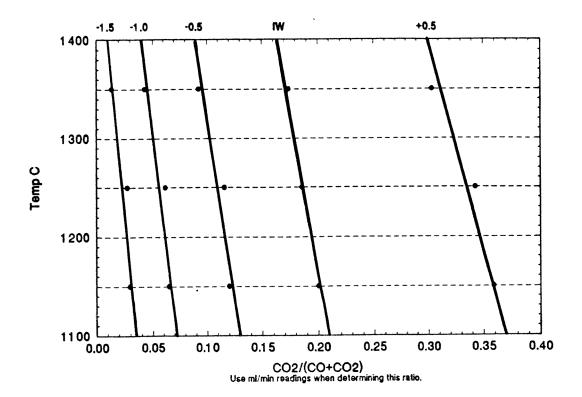


Figure 11.

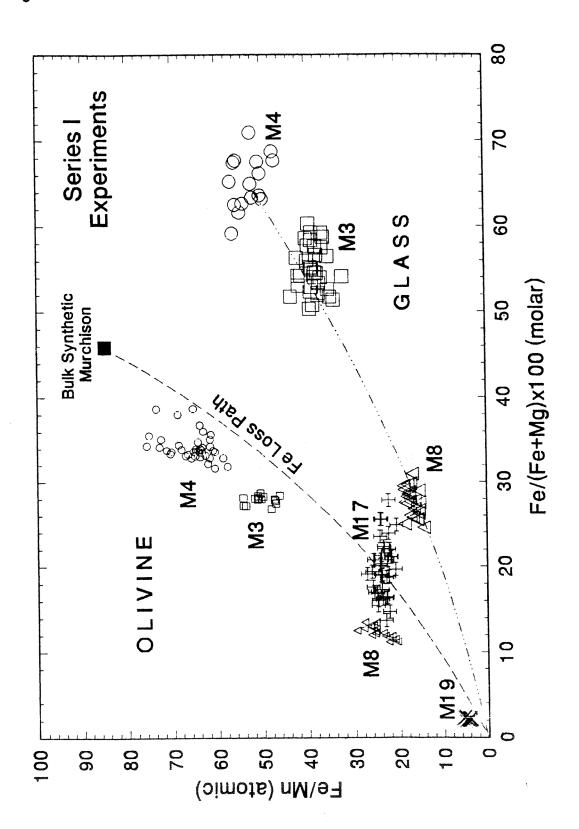


Figure 12.

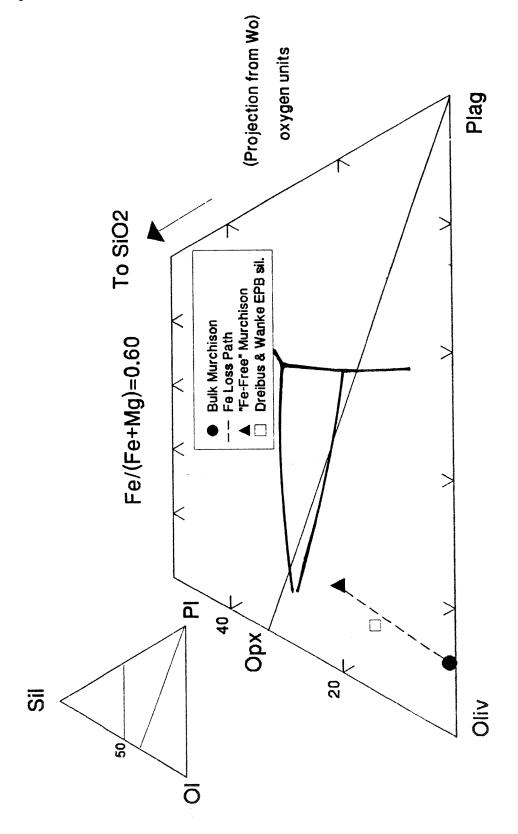


Figure 13.

Fe/(Fe+Mg)x100 (molar)

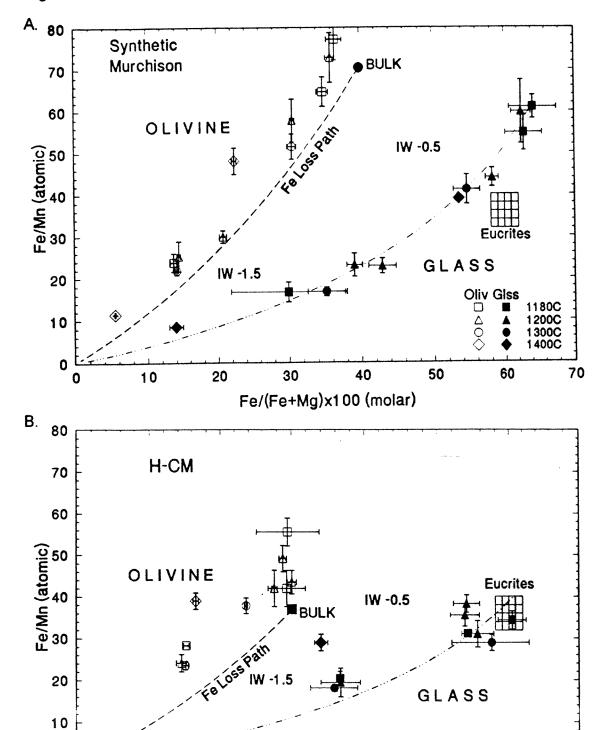


Figure 14.

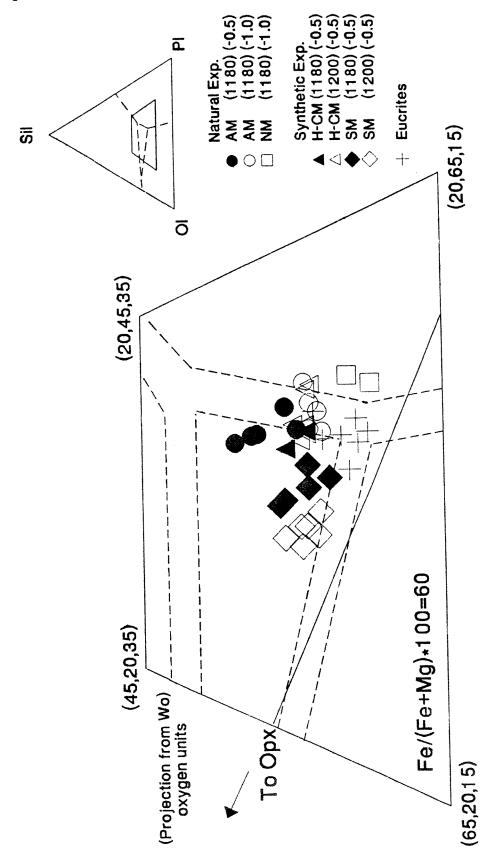


Figure 15.

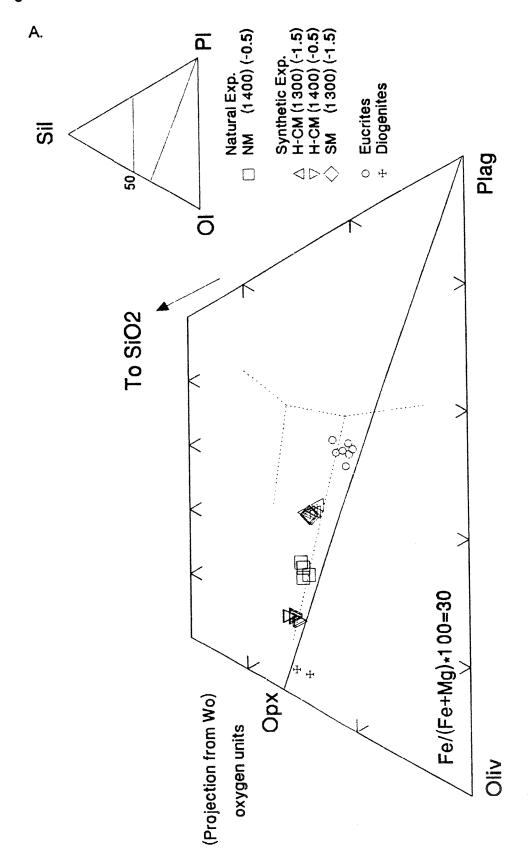
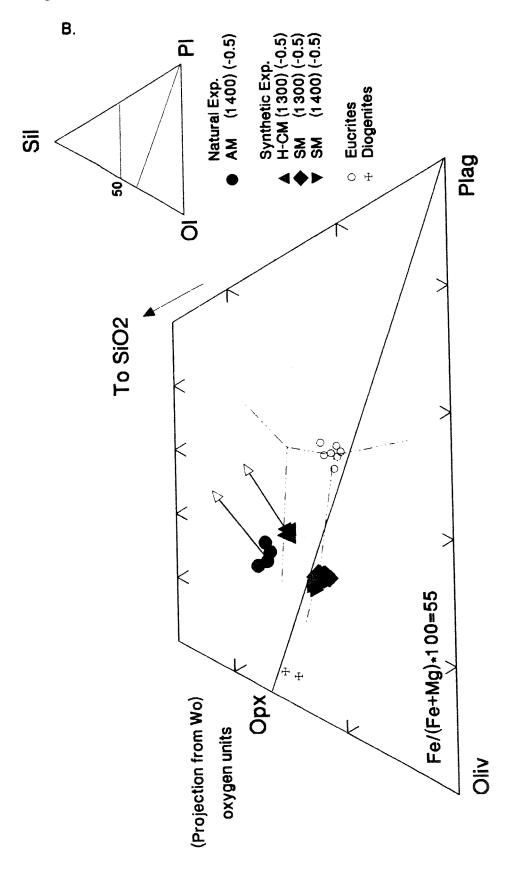
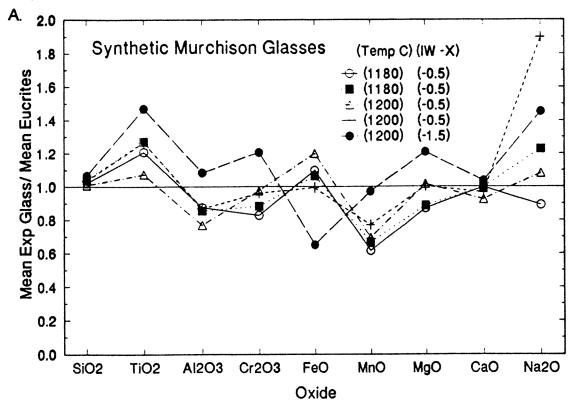
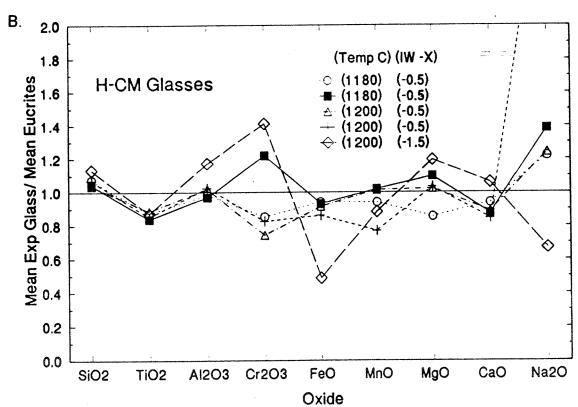


Figure 15.

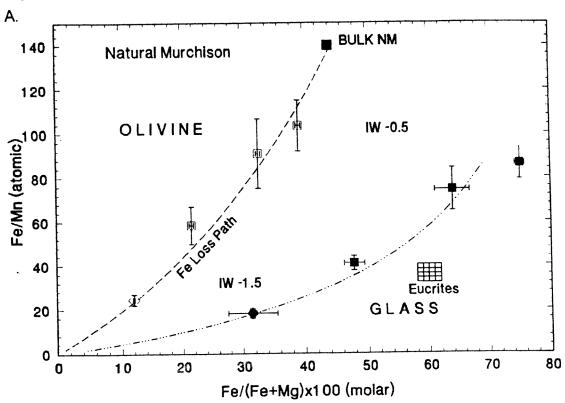












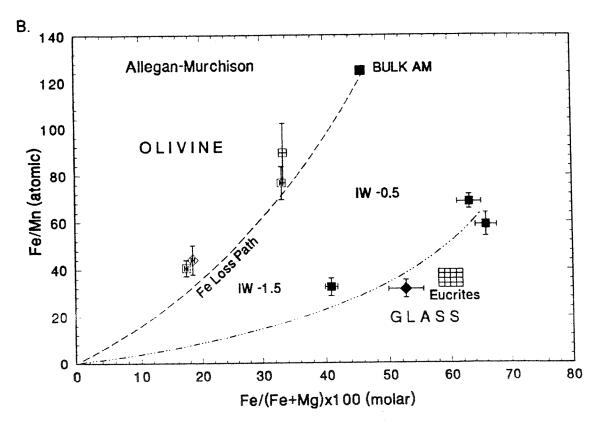


Figure 18.

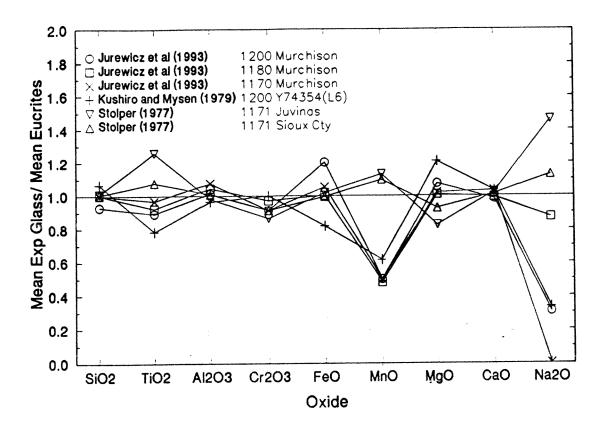
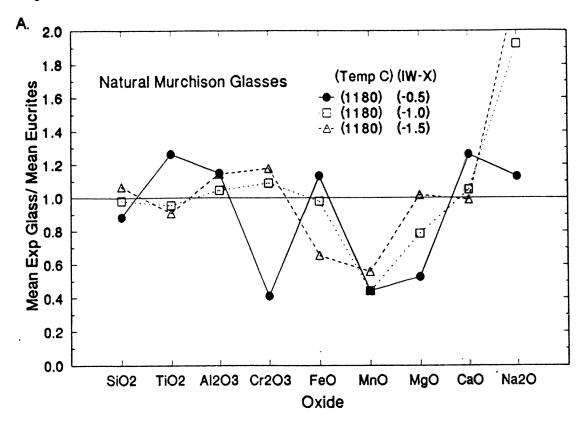


Figure 19.



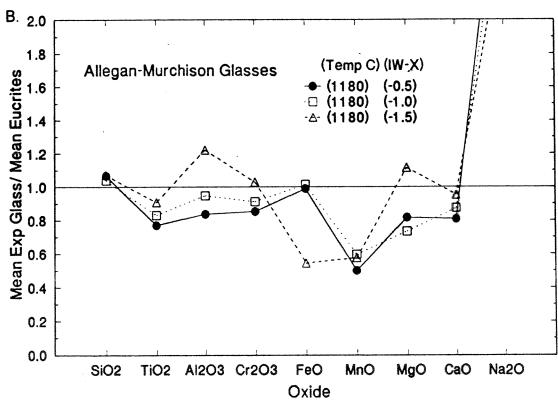


Figure 20.

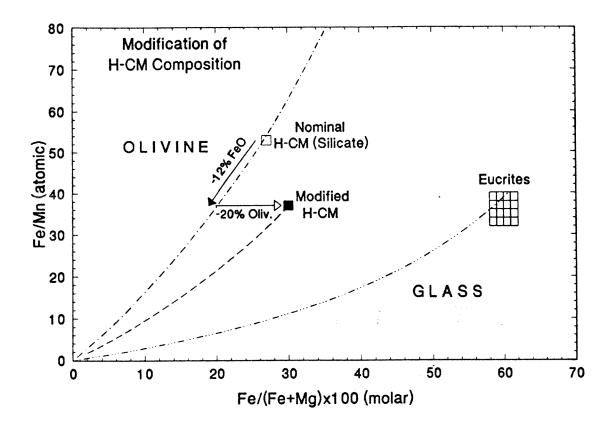
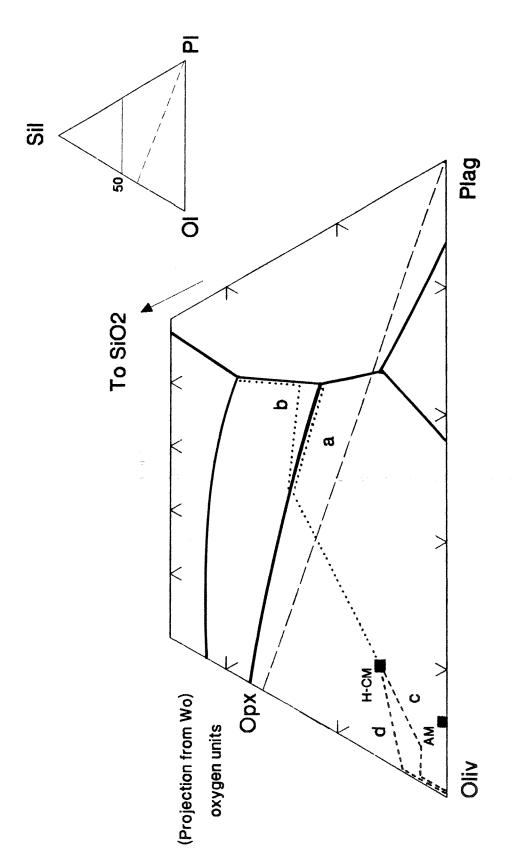


Figure 21.



Appendix I. Chemical Analyses of Electron Microprobe Mineral Standards

Mineral	SiO ₂	TiO ₂	Al ₂ O ₃	Al2O3 Cr2O3 FeO	FeO	Mno	MgO	CaO	Na ₂ O	X 30	Q Z	Total	Ref.
Forsterite, synthetic Fo 100*	42.7	,	,	1	,		57.3	1	,	,	,	100.00	-
Olivine (Fo 90), San Carlos, Gils Co., AZ* USNM 11312/444	40.81				9.55	0.14	49.42	0.05			0.37	100.34	7
Olivine (F183), Springwater meteorite** IISMM 2566	38.95		•	0.02	16.62	0.30	43.58		,	ı		99.47	7
Fayalite, synthetic Fa 100*	29.5		,	1	70.5				r	ı	ı	100.00	ო
Fayalite, Rockport, MA**	29.22	0.04	ı	1	9.79	2.14			,	ı	•	99.70	2
Convin 50270 Tephroite, synthetic*	29.8	ı			ı	70.3			ı	1	ı	100.10	4
Orthopyroxene, Tatahouine meteorite**	54.7	0.07	0.53	0.64	15.2	0.54	27.6	0.71	0.01			100.00	S.
Hypersthene, Johnstown meteorite** USNM 746	54.09	0.16	1.23	0.75	15.22	0.49	26.79	1.52	0.05	0.05	•	100.35	7
Augite, Kakanui, New Zealand** USNM 122142	50.73	0.74	8.73	•	6.4	0.13	16.65	15.82	1.27	ı	ı	100.47	7
Hornblende, Kakanui, New Zealand⁴ USNM 143965	40.37	4.72	14.90	ı	11.0	60.0	12.80	10.30	2.60	2.05	•	98.83	7
Anorthite, Great Sitkin Island, AL* USNM 137041	44.00	0.03	36.03	,	0.62	1	0.02	19.09	0.53	0.03	,	100.35	7
Plagioclase (Labradorite), Lake County, OR* USNM 115900	51.25	0.05	30.91	,	0.15	0.01	0.14	13.64	3.45	0.18		99.78	2
Anorthoclase, Kakanui, New Zealand** USNM 133868	66.44	t	20.12	1	0.20	ı	1	0.87	9.31	2.35		99.29	7
Chromite, Tiebaghi Mine, New Caledonia* USNM 117075	•	,	9.92	60.5	13.04	0.11	15.20	0.12	,		1	98.89	7

Calibration standard Additional standard used to test microprobe calibration

References:
1 Takei and Kobayashi (1974)
2 Jarosewich et al (1980)
3 Takei (1978)
4 Takei (1976)
5 Rutgers University Microanalysis Laboratory standard. Analyzed by J.S. Delaney

Appendix II. Chemical Analyses of Series I, II, and III Experimental Charges

A. Series I Experiments (Temp) (Exp type)

(Exp #)

Synthetic Murchison Fe/Mn 85

	FS			24.07 20.02 20.08 21.83 21.08 21.54 12.05 12.77 23.44
	WO			4.95 4.57 4.59 6.39 6.35 6.06 6.06 6.06 6.06 6.06 6.06
	FFM	36.90 56.97 59.87 61.01 52.97	14.08 113.88 15.51 19.04 16.94 13.93	25.33 20.98 21.67 22.39 22.39 20.39 24.24 26.27
	FEMN	20.70 22.43 24.17 23.94 22.93 36.30	30.93 36.24 35.76 35.60 38.08 34.04 32.86	20.63 22.33 20.25 20.50 17.77 20.36 22.39 23.21 22.31
	Ca	1467 1740 1847 1862 1912	w 4 w 4 4 w @ 4 w	96 89 89 117 127 127 120 120 120 120
	Mg	2227 979 793 684 812 1706	1682 1648 1765 1692 1666 1582 1623 1644	1353 1444 1424 1344 1383 1356 1358 1353 1353 1347
	Min	65 61 52 7 7	00 6 8 8 5 5 5 6	50 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1
	Fe	1341 1261 1146 961 257	277 230 230 274 307 374 333 319	467 388 399 399 415 406 406 397 351 430 346 425
	Ö	47 28 28 17 16	01 88 7 7 8 7 7 7 7	232288888888888888888888888888888888888
	¥	1896 2521 2573 2652 2741		74 77 70 77 74 74 74 84 84 83 93 95 100 100
	Ø	7992 8010 8073 8129 8069 1005	1007 1007 893 1006 1006 1006 1006	1973 1977 1976 1976 1980 1980 1990 1990
	TOTAL	100.30 97.89 100.82 99.28 101.40	101.33 100.70 101.54 100.79 100.84 100.44 101.05	100.67 101.33 100.24 101.09 100.27 101.35 101.35 100.72
œ	CAO	9.67 11.21 12.31 12.30 12.98	0.15 0.15 0.15 0.20 0.20 0.10	2.48 2.34 2.34 2.92 2.92 3.18 3.20 2.98 3.19 3.19
M09	MGO	10.55 4.53 3.25 3.96 46.54	45.86 44.34 48.67 42.05 42.05 43.36 44.32 45.88	25.15 27.39 26.66 25.27 25.82 25.87 25.85 25.75 25.75 25.63 25.63
	MNO	0.54 0.44 0.44 0.36 0.34	0.44 0.34 0.34 0.46 0.46 0.46 0.46	0.74 0.65 0.65 0.75 0.75 0.66 0.59 0.59 0.61
cooling	FEO	11.32 10.77 9.70 8.36	13.47 15.05 11.28 14.76 17.74 15.86 15.31	15.46 13.32 13.91 13.61 13.61 13.84 11.90 11.75
8	CR203	0.42 0.25 0.25 0.15 0.48	0.51 0.32 0.37 0.34 0.47 0.37	1.07 0.89 0.91 1.01 0.83 0.99 1.05 0.95 0.75 1.08
ပ္	AL203	11.36 14.76 15.59 15.93 16.91	0.0000000000000000000000000000000000000	1,11 1,13 1,168 1,76 2,23 2,23 2,24 2,25 2,25
1180°C	SIO2	glass 56.44 55.28 57.66 57.55 58.68 olivine	40.93 40.94 40.99 40.26 39.88 40.11 40.41	55.88 55.39 55.47 55.39 55.30 55.47 55.00 56.41

1180°C	ပ	_	heating		M13	m													
SIO2	AL203	CR203	FEO	MNO	MGO	CAO	TOTAL	S	¥	Ö	ů.	Mn	Mg	ප	FEMN	FFM	WO	FS	
glass																			
53.95	1.49	0.69	14.82	0.49	24.62	3.81	99.87	7850	256	79	1803	90	5341	594	29.86	25.03			
52.63	4.76	0.82	15.45	1.06	19.27	5.96	99.95	7722	823	92	1896	132	4215	937	14.39	30.37			
52.74	4.44	0.63	15,61	0.58	19.65	6.20	99.85	7739	768	73	1916	72	4299	975	26.57	30.47			
ofivine																			
38.70	0.08	0.58	23.81	0.87	36.71	0.39	101.14	1005	7	12	517	19	1421	-	27.02	26.42			
38.32	0.21	0.29	24.79	0.35	35.86	0.57	100,39	1005	9	9	544	60	1403	16	69.93	27.83			
38.02	0.30	0.25	23.41	0.30	36.79	0.62	99.69	666	6	5	515	7	1441	17	77.05	26.22			
37.87	0.38	0.88	25.68	0.99	34.01	0.52	100.33	1003	12	18	269	22	1343	15	25.61	29.41			
pigeonite		0	4 4 4 5	0	4	27.0		100	'n	ď	uC.	4	0 70	ć	77.63	24.94	6	23 50	
7		77.0	2 :	9	7	0		00	3	٠ د	3	ָ י	2	77	70.7	7.7	2	200	
54.81		0.22	14.13	0.39	24.31	6.13		1982	56	9	427	12	1311	238	35.77	24.42	11.95	21.50	
54.06		0.50	14.56	0.34	24.24	5.36		1972	34	14	444	Ξ	1319	210	42.28	25.05	10.57	22.41	
55.41	0.81	0.18	12.29	96.0	25.34	4.91	99.90	1996	34	c)	370	53	1361	189	12.64	21.03	9.72	18.99	
54.38		0.76	13.29	0.59	24.92	4.87		1968	25	77	402	18	1345	189	22.24	22.79	6.67	20.59	
jacico																			
1.37	65.53	0.15	15.30	0.10	16.88	0.63	96.66	35	1962	e	325	2	639	17	151.07	33.63			
000	60.09	0.19	18.73	0.15	15.93	0.00	101.09	0	1990	4	400	n	209	0	123.29	39.62			
90'0	90'.29	00.00	13.47	0.57	18.70	0.07	99.93	2	1996	0	285	12	704	7	23.33	28.43			
1	24.05	40.95	21.03	1.48	8.66	0.32	97.60	35	904	1033	561	40	412	=	14.03	55.39			
0.13	1.66	65.23	22.20	1.25	5.69	0.08	96.24	5	72	1897	683	36	312	က	17.54	66.05			

	FS																													21 76	20.00	2000	24.24	200	30.05	27.77	08.77	20.00	20.93	23.43	23.13	24.43					
	WO																													2 80	9 6	26.40	20.0	200	2.40	9 0	200	40.0	0.40	9 5	20.7	2.51					
	FFM	5,6 2,5	54.84	53.01	57.60	53.99	54.08	56.44	58.68	55.93	57.44		20 42	20.13	76.97	77.80	27.96	26.80	28.66	28.02	28.08	28.27	28.04	27.94	27.20	27.45	27.16	28.36	27.91	77 17	70.00	20.00	20.03	25.70	24.02	22.7.	27.57	24.00	40.17	23.88	73.60	24.85		60 95	58.33	58.51	
	FEMN	70.02	38.70	41.67	37.94	38.00	32.06	35.34	36.29	39.74	36.59		75 43	74.07	0.00	47.74	50.34	48.14	50.57	50.94	51.10	49.94	51.84	47.41	54.35	47.03	53.80	46.24	51.18	34 10	20.5	00.00	24.00	24.00	33.17	32.07	70.04	30.00	32.03	32.63	27.89	30.33		49.39	48.38	56.63	
	Ca	1452	1427	1349	1476	1398	1418	1508	1498	1453	1459		1	- 0	1 0		9	7	9	9	80	7	80	9	9	9	7	80	7	75	2 4	8 4	ດ ຜ	7 7	‡ ú	2	4 4	5 6	8	3 2	2	48		er.	m	0	
	Mg	1066	2067	2216	1874	2143	2061	1938	1781	1910	1806		1437	/74-	1409	147/	1436	1437	1396	1417	1428	1405	1433	1432	1437	1431	1438	1408	1412	4472	7 7 7 7	7 4 4	7041	7 4 4	4644	4440	904	5440	148/	1425	1446	1409		384	423	436	
	¥	ç	29	62	2	99	79	74	73	63	69		5	2;	_ (7.	11	÷	F	Ę	F	=	F	12	10	12	9	12	=	45	2 4	5 5	7 5	 	<u> </u>	2 !	- ;	2 ;	2	4	φ.	16		6	5 62	=	
	Fe	7090	2592	2570	2642	2595	2520	2607	2633	2504	2531		583	700	000	554	299	530	999	556	562	558	563	260	541	546	540	295	551	430	120	101			944	470	544	603	411	451	452	471		619	610	630	
	Ö	S	8	8	84	23	103	2	93	79	85		ç	7 (2 9	<u> </u>	13	14	12	14	13	12	14	12	14	7	12	10	S	ć	2 5	7 0	9 2	4 6	4 5	7	8 8	87	74	28	90	58		1434	1405	1219	
	¥	8181	1828	1830	1837	1843	1850	1875	1904	1911	1909		c	0 0	۰ د			-	-	-		-	7	7	7	7	2	7	7	4	2 4	D 0	2 2	9 6	9 8	9 6	8 8	ဌာ	36	98	37	9		540	556	675	
	Si	7525	7485	7462	7528	7476	7496	7478	7509	7542	7566		880	0 0	988	20.0	385	966	1000	994	985	1000	981	984	991	1000	885	966	1003	4070	0/6	1 200	1800	0/0	2007	200	19/1	1967	1966	1983	1965	1977		7	t vo	, 4	
	TOTAL	98.60	98.96	99.04	99.00	98.84	98.68	98.63	98.50	98.60	90.06		78 00	99.07	50.00	99.43	98.68	99.19	100,30	100.64	100.10	100.26	99.44	98.88	101.09	100.04	100.15	99,64	99.60	0	9 6	2000	80.83	000	67.66	66.00	99.56	99.68	98.98	99,33	100.21	98.66		CO RO	97.11	97.84	
_	CAO	88	8.76	8.30	9.05	8.58	8.70	9.21	9.14	8.93	9.00		30	0.40	17.0	C7.0	0.22	0.25	0.22	0.21	0.28	0.24	0.28	0.22	0.21	0.22	0.25	0.27	0.25	4	2 4	<u>.</u>	5	7	2.5	87	1.26	1.18	1.76	1.29	1.44	1.23		800	0.00	000	
M 4	MGO	8.64	9.12	9.80	8.26	9.45	60.6	8.51	7.81	8.44	8.01		36.05	20,00	55.70	35.85	35.82	36.33	35.47	36.15	36.15	35.72	36.00	35.83	36.94	36.43	36.63	35.53	35.77	01.40	20.00	00.00	707	20.51	20.00	20.93	26.99	26.76	27.46	26.33	26.88	25.77		7 66	90.8	8.90	; ;
	MNO	0.48	0.52	0.48	0.54	0.53	0.61	0.57	0.56	0.49	0.54		0.46	9.0	0.49	70.0	0.49	0.49	0.50	0.49	0.49	0.50	0.48	0.52	0.45	0.52	0.45	0.54	0.48	*	5 C	9. O	0.40	0.02	0.44	0.43	0.56	0.43	0.41	0.45	0.53	0.50		77	9	0.40	! i
heating	FEO	20.43	20.38	20.26	20.75	20.40	19.81	20.40	20.58	19.72	20.01		25 33	25.00	40.07	24.88	24.98	23.89	25.61	25.28	25.36	25.29	25.20	24.97	24.77	24.77	24.52	25.29	24.88	9	4.7	15.14	14.85	10.04	14.78	14.23	14.77	15.27	13.55	14.87	14.97	15.36		20.00	24.56	22.94	
ڪ	CR203	0.75	0.75	0.75	0.70	0.44	0.86	0.58	0.77	99.0	0.77		850	0000	0.00	0.60	0.61	0.67	0.57	29.0	09.0	0.56	0.67	0.57	0.69	0.09	0.57	0.47	0.25	4	5.4	0.75	C 80	0.85	0.87	0.75	96.0	1.02	0.82	0.92	1.05	96.0		08 23	50.08 50.54	46.92	•
ပ	AL 203	1011	10.20	10.24	10.24	10.28	10.32	10.41	10.56	10.68	10.71		9	3 6	00.0	0.02	0.02	0.04	0.04	0.04	0.04	0.04	90.0	90.0	90.0	90.0	90.0	90.0	0.08	•	o 0	5 1	φ (9	0 6	S	0	0	0	0.85	0	0		13 63	12.03	17.44	-
1253°C	SIO2	glass 49.31	49.23	49.21	49.46	49.16	49.29	48.95	49.08	49.68	50.02	quivilo	37.20	37.70	57.03	37.20	36.54	37.52	37.89	37.80	37.18	37.91	36.75	36.71	37.97	37.95	37.67	37.48	37.89	pigeonite	04.40	54.83	54.36	54.13	54.62	55.60	54.36	54.21	54.15	54.62	54.47	53.91	-	spinel	- 4 - 4	1.13	4

	FEMN FFM WO FS																	55.71 62.50 56.76 65.22 51.87 63.34 52.15 64.94 52.15 64.94 50.14 63.57 50.08 66.16 55.60 67.51 46.94 67.66 55.49 67.66 55.49 67.65 55.20 70.95																		
	Mg Ca	·		•	•			. ,	. , , , , ,	. , , , , , ,		. , , , , , , , , ,						2332 1027 2333 994 1754 1264 1958 1273 1970 1178 1902 1178 1843 1342 1661 1317 1669 1368 1542 1398 1572 1398 1571 1390 1489 1390			000 000 128 138 138 138 138 138 138 138 138 138 13	000 000 000 000 000 000 000 000 000 00	000 128 128 138 138 138 138 138 138 138	902 128 138 138 138 138 138 138 138 138 138 13	0.00 0.00 0.00 0.00 0.00 0.00 0.00 0.0	00 00 00 00 00 00 00 00 00 00 00 00 00	0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0	905 907 128 138 138 138 138 138 138 138 138 138 13	0 6 6 7 7 7 8 8 8 8 8 8 8 8 8 8 8 8 8 8 8	0 6 6 7 7 7 8 8 8 8 8 8 8 8 8 8 8 8 8 8 8	0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0	206 821 122 125 125 125 125 125 125 125 125 1	00 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0		0 6 6 7 7 7 7 8 8 8 8 8 8 8 8 8 8 8 8 8 8	00 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0
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	Ò	i	71	7.3	74 44 50	77 44 50 87	1, 4, 0, 0, 0, 0, 0, 0, 0, 0, 0, 0, 0, 0, 0,	17 44 67 79 70 70	1,488 65 7,5 2,488 65 7,5	1. 4 6 8 8 7 7 2 8	17 4 6 6 6 7 7 7 8 8 8 8 9 8 7 7 8 8 8 9 9 9 9 9 9	1, 4, 8, 8, 8, 8, 8, 8, 8, 8, 8, 8, 8, 8, 8,	1, 4, 6, 6, 6, 7, 7, 8, 8, 8, 8, 8, 8, 8, 8, 8, 8, 8, 8, 8,	7 4 6 6 6 7 7 7 8 8 8 8 8 8 8 8 8 8 8 8 8 8	7 4 4 6 6 6 7 7 7 8 8 8 8 8 8 8 8 8 8 8 8 8 8	1. 4.00 6 7. 7. 7. 8. 8. 8. 8. 8. 8. 8. 8. 8. 8. 8. 8. 8.	7 4 6 6 6 6 6 6 6 6 6 6 6 6 6 6 6 6 6 6	L 4 8 8 8 8 8 8 8 8 8 8 8 8 8 8 8 8 8 8	1. 4 6 6 6 7 7 8 8 8 8 8 8 8 8 8 8 8 8 8 8 8	7.440007772888888888900 C	7488888888888 6 C C	7.400077758888888900 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0	7.460007752888888897 0.001 4.4600077588888897 0.001	74888888888	7.4000777588888888900 0 0 0 4 4 0 0 0 0 0 0 0 0 0 0 0 0 0	7.40007775888888888970 0.044070	7.40007778888888897 0.00144070 9.440499999999999999999999999999999999		7.400077758888888890 0 0 0 0 4 4 0 0 m s 1 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0	7.4000777888888888926.0007445πα1056.0007445πα1056.000445πα1056.000445πα1056.000445πα1056.000445πα1056.0004456.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476.000476		7.400077758888888890 6.440007778888888890 7.4000777888888890 7.400007888888890	<u> </u>	 7.488888888889 6.6054450 6.6054450 6.6054450 6.605460 <li< td=""><td>7.4888888888889 . 48867788888888899 . 50544500550555</td><td>7.48888888888890 0 0 0 0 4 4 0 0 0 0 0 0 0 0 0 0 0 0 0</td></li<>	7.4888888888889 . 48867788888888899 . 50544500550555	7.48888888888890 0 0 0 0 4 4 0 0 0 0 0 0 0 0 0 0 0 0 0
	TOTAL Si AI		7199	7199	7199	7199 7154 7169 7413	7199 7154 7169 7413 7388	7199 7154 7169 7413 7388 7470	7199 7154 7169 7413 7388 7470	7199 7154 7169 7413 7388 7470 7361	7199 7154 7169 7413 7388 7470 7361 7355	7199 7154 7169 7413 7388 7470 7361 7355 7360 7361	7199 7154 7169 7413 7388 7470 7361 7355 7380 7361	7199 7154 7169 7413 7388 7470 7361 7355 7361 7361 7361 7361	7199 7154 7169 7413 7438 7470 7361 7361 7361 7361 7361 7361	7199 7154 7169 7413 7388 7470 7361 7355 7361 7451 7451 7451	7199 7154 7169 7413 7388 7361 7361 7361 7361 7361 7361 737 7361 737	97.47 7199 1529 98.83 7154 1582 98.63 7169 1452 98.643 7413 1718 99.643 7388 1539 97.29 7470 1568 98.87 7355 1762 98.63 7380 1675 98.63 7380 1659 98.737 1659 98.737 1659 98.737 1659 98.737 1659 98.737 1659 98.737 1659	7199 7164 7169 7413 7388 7470 7361 7355 7361 7451 7451 7451 7451 7451 7451 7451 745	7199 7154 7154 7413 7388 7355 7355 7351 7351 7451 7451 7292 7341 7292 7341	7199 7154 7159 7413 7470 7361 7356 7380 7380 7380 7380 7380 7380 7381 7381 7282 7391 7292 7391 7292	7199 7154 7154 7156 7358 7355 7355 7355 7357 7357 7357 7357	7199 7154 7154 7413 7388 7355 7355 7351 7351 7451 7451 7451 7451 7451 7451 7451 74	7199 7154 7156 7410 7410 7355 7356 7351 7351 7351 7351 7351 7351 7351 7351	7199 7164 7169 7413 7388 7386 7360 7361 7361 7361 7361 7361 737 737 737 737 737 737 737 737 737 73	7199 7154 7154 7413 7486 7355 7366 7367 7367 7367 7367 737 7292 7292 7292 7292 7292 7292 7292	7199 7154 7154 7413 7388 7386 7361 7351 7351 7451 7451 7451 7451 7451 7451 7451 74	7199 7154 7156 7410 7386 7355 7356 7351 7351 7351 7351 7351 7351 7351 7351	7199 7154 7154 7154 7355 7355 7356 7357 7357 7357 7357 7357	7199 7154 7154 7413 7388 7386 7361 7361 7361 7361 737 7235 7282 7292 7292 998 998 999 998 998 998 998	7199 7154 7158 7247 7355 7356 7351 7351 7351 7351 7351 7351 7351 7351	7199 7154 7154 7154 7155 7155 7155 7155 7155	7199 7154 7413 7413 7413 7410 7361 7361 7451 7451 7451 7451 7451 7451 7451 745	7199 7154 7413 7388 7386 7355 7351 7351 7451 7451 7451 7451 7451 7451 7451 74	7199 7154 7154 7413 7361 7361 7361 7361 7361 7361 7361 73	7199 7154 7154 7154 7155 7388 7388 7381 7381 7381 7381 7381 7292 7292 7292 1000 998 999 999 997 998 997 998 998 998
	CAO		7.25	7.25	7.25 6.02 5.68	7.25 6.02 5.68 7.51	7.25 6.02 5.68 7.51 7.61	7.25 6.02 5.68 7.51 7.51	7.25 6.02 5.68 7.51 7.51 7.51	7.25 6.02 5.68 7.51 7.51 7.88 7.88 6.99	7.25 6.02 5.68 7.51 7.51 7.51 8.99 8.07	7.25 6.02 5.68 7.51 7.51 7.88 6.99 8.07	7.25 6.02 7.58 7.51 7.51 7.51 7.88 6.99 8.07 7.94	7.25 6.02 5.68 7.51 7.51 7.88 7.88 7.99 7.99 8.02	7.25 6.02 6.02 7.51 7.51 7.51 7.88 8.07 7.82 8.30 8.30	2.25 6.02 7.51 7.51 7.51 7.65 7.85 7.82 8.02 8.02 8.10	7.25 6.02 7.51 7.51 7.51 7.51 7.89 8.00 8.00 8.30 8.30 8.30	992 7.25 9 978 602 9 9778 602 9 9778 602 9 9779 7.49 7.51 9 8741 7.61 9 8744 7.61 9 8744 7.61 9 8744 7.61 9 8744 7.61 9 8744 7.61 9 8744 7.61 9 8744 7.61 9 8745 88 80	7.25 6.02 6.02 7.51 7.51 7.51 7.51 7.51 8.30 8.30 8.30 8.30 8.30 8.31 8.31 8.31	2.25 6.00 6.00 7.55 7.55 7.75 7.85 7.82 7.82 7.82 8.30 8.30 8.30 8.10 8.10	2.25 6.00 7.55 7.55 7.55 7.55 8.00 7.80 8.00 8.00 8.00 7.00 7.00 8.00 8	2.25 6.602 5.602 7.51 7.51 7.51 7.88 8.30 8.30 8.30 8.30 8.30 8.30 8.30 8	2.25 6.02 5.60 5.60 5.60 5.60 5.60 5.60 5.60 5.60	2.25 6.02 7.55 7.55 7.55 7.55 8.02 8.02 8.02 8.10 8.10 8.10 9.02 7.02 7.02 7.02 7.02 7.02 7.02 7.02 7	2.25 6.60 6.75 6.75 6.75 6.75 6.85	2.25 6.602 7.51 7.51 7.51 7.51 7.51 8.72 8.83 8.83 8.83 8.83 8.83 8.83 8.83 8.83 9.83	2.25 6.602 7.51 7.51 7.51 7.51 7.51 7.51 8.602 8.602 8.702 8.702 8.702 8.702 8.702 9.702 1.002 1.	7.25 6.60 7.51	2.25 5.602 5.602 5.602 5.603 5.6	2.25 5.60 5.60 5.75 5.75 5.81 5.75 5.82 5.83 5.83 5.83 6.83	7.25 6.60 7.51 7.51 7.51 7.51 7.51 8.72 8.73	2.25 5.602 5.75 5.75 5.75 5.85	2.25 5.60 5.60 5.75 5.75 5.60 5.75 5.60 5.75 5.60 5.75 5.60 5.75	2.55 6.60	25.7 6.00	2.25 6.60
M04	MNO MGO		28 0.46	28 0.46	28 0.46 80 0.52 34 0.52	28 0.46 80 0.52 34 0.52 87 0.45	28 0.46 80 0.52 34 0.52 87 0.45 79 0.51	28 0.46 80 0.52 34 0.52 87 0.45 79 0.51 88 0.49	28 0.46 80 0.52 34 0.52 87 0.45 79 0.51 71 0.47	28 0.46 80 0.52 34 0.52 87 0.45 79 0.51 88 0.49 65 0.51	28 0.46 30 0.52 34 0.52 87 0.45 77 0.49 88 0.49 71 0.47 95 0.50	28 0.46 80 0.52 34 0.52 87 0.45 88 0.49 71 0.47 65 0.51 88 0.53	28 0.46 80 0.52 34 0.52 37 0.51 79 0.51 88 0.49 65 0.51 71 0.47 72 0.51 88 0.53	28 0.46 80 0.52 34 0.52 34 0.57 79 0.51 79 0.51 65 0.51 70 0.51 70 0.50 88 0.50 88 0.50 17 0.48	28 0.46 80 0.52 3.4 0.52 3.4 0.52 7.9 0.51 7.1 0.47 6.5 0.51 3.9 0.50 88 0.53 1.7 0.48 1.7 0.48 1.7 0.48	28 0.46 80 0.52 3.4 0.52 3.4 0.52 7.9 0.51 7.1 0.47 6.5 0.51 9.9 0.50 1.7 0.48 1.7 0.48 1.6 0.53 1.6 0.53	28 0.46 80 0.52 34 0.52 34 0.52 70 0.51 88 0.49 88 0.50 89 0.50 89 0.50 110 0.53 31 0.53 31 0.53 31 0.53 31 0.53 31 0.53 31 0.53	28 0.46 80 0.52 34 0.52 34 0.52 79 0.45 79 0.45 65 0.51 65 0.53 67 0.53 67 0.54 67 0.54 67 0.54 67 0.55 67 0.55 67 0.55 67 0.55 67 0.55	28 0.46 80 0.52 34 0.52 34 0.52 79 0.51 79 0.51 11 0.51 17 0.48 16 0.53 10 0.50 31 0.57 55 0.54	28 0.46 89 0.52 89 0.52 89 0.52 89 99 99 99 99 99 99 99 99 99 99 99 99	28 80 80 80 80 80 70 70 80 80 80 90 90 90 90 90 90 90 90 90 90 90 90 90	28 0.046 88 0.52 934 0.552 934 0.552 938 938 938 938 938 938 938 938 938 938	28 80 80 80 80 80 80 80 80 80 8	28 80 80 80 80 80 80 80 80 80 8	28 80 80 80 90 90 90 90 90 90 90 90 90 9	28 80 80 80 80 80 80 80 80 80 8	28 00 46 00 48 00	28 80 80 80 80 80 80 80 80 80 8	28 88 0.52 9.34 0.55 9.34 0.55 9.34 0.55 9.34 0.55 9.34 0.55 9.34 0.55 9.34 0.55 9.34 0.55 9.34 0.55 9.34 0.55 9.34 0.55 9.34 0.55 9.34 0.55 9.34 0.55 9.34 0.55 9.34 0.55 9.34 0.55 9.34 0.55 9.34 0.55 9.34 9.34 0.45	28 80 80 80 80 80 80 80 80 80 8	28 80 80 80 80 80 80 80 80 80 8	28 88 934 0.55 28 934 935 94 94 94 94 94 94 94 94 94 94 94 94 94	28 89 9 9 9 9 9 9 9 9 9 9 9 9 9 9 9 9 9	28 80 8 80 9 9 9 9 9 9 9 9 9 9 9 9 9 9 9	28 80 80 80 80 80 80 80 80 80 8	28 89 89 89 89 89 89 89 89 89 8
ating	3 CR203 FEO		0.56 26	0.56 26	0.56 26 0.35 28 0.39 29	0.56 26 0.35 28 0.39 29 0.64 25	0.56 0.35 0.39 0.64 0.64 25	0.56 0.35 0.39 0.39 0.64 0.64 25 0.61 25	0.56 26 0.35 28 0.39 29 0.64 25 0.64 25 0.64 26	0.56 0.35 0.39 0.64 0.64 0.64 0.64 0.64 0.65 0.65 0.66	0.56 0.39 0.39 0.64 0.64 0.64 0.65 0.68 0.72 0.72	0.56 0.39 0.39 0.39 0.64 0.64 0.64 0.68 0.58 0.58 0.58 0.67 0.67 0.67 0.67 0.67 0.67 0.67 0.67	0.56 0.39 0.39 0.64 0.64 0.58 0.58 0.72 0.72 0.66 0.66 0.72 0.66 0.66 0.72 0.66 0.66 0.72 0.66 0.72 0.72 0.72 0.72 0.72 0.72 0.72 0.72	0.56 0.39 0.39 0.09 0.64 0.64 0.65 0.72 0.72 0.65 0.65 0.66 0.66 0.66 0.66 0.66 0.66	0.56 0.39 0.39 0.04 0.04 0.04 0.04 0.05 0.05 0.05 0.05	0.56 0.35 0.35 0.05 0.05 0.05 0.05 0.05 0.05	0.56 0.39 0.39 0.39 0.64 0.64 0.65 0.65 0.65 0.65 0.65 0.65 0.65 0.65	0.56 0.39 0.39 0.09 0.09 0.09 0.09 0.09 0.09	0.56 0.35 0.39 0.39 0.64 0.64 0.64 0.75 0.65 0.64 0.64 0.64 0.77 0.70 0.70 0.70 0.70 0.70 0.70 0.7	0.35 0.35 0.39 0.39 0.04 0.04 0.05 0.05 0.05 0.05 0.05 0.05	0.35 0.35 0.39 0.39 0.39 0.39 0.39 0.39 0.39 0.39	0.35 0.35	0.56 0.035 0.035 0.04 0.04 0.04 0.04 0.04 0.04 0.04 0.0	0.056 0.039 0.039 0.039 0.039 0.04 0.04 0.04 0.04 0.04 0.04 0.04 0.0	0.056 0.035 0.	0.35 0.35	0.35 0.35	0.35 0.35	0.056 0.035 0.	0.356 0.356 0.357 0.357 0.357 0.357 0.357 0.357 0.358	0.00 0.00	0.056 0.036 0.04 0.056 0.057 0.057 0.057 0.057 0.057 0.057 0.057 0.057 0.057 0.057 0.057 0.057 0.058 0.0	0.38	0.056	0.039	0.00 0.00
1300°C h	SIO2 AL203																	91ass 44.91 44.93 43.88 47.19 47.79 46.89 46.89 46.75 47.03 46.75 46.75 46.74																		,,

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	FFM	28.53	25,96	26.23	28.20	25.69	27.52	27.75	26.59	29.00	28.79	26.74	28.50	27.61	29.59	28.94	11,63	11.49	11.87	12.21	12.70	11.88	12.02	12.33	12.16	12.74	13.10	12.93	13.09	13.46	13.39
	FEMN	17.53	17.29	15.38	18.11	16.77	15.14	16.28	15.34	18.78	17.52	16.48	17.83	17.33	17.89	15.34	22,55	21.13	21.82	23.78	25.30	23.35	26.41	25.37	25.52	28.08	25.90	26.53	26.64	27.89	25.37
	S	986	943	975	980	066	1001	961	970	994	1022	993	1129	1009	1051	1044	n	4	4	හ	4	4	4	ო	4	ო	හ	၈	₹7	6	က
	Mg	3168	3268	3131	3200	3273	3003	3219	3210	3149	3156	3208	2782	3085	2903	2923	1764	1742	1735	1746	1729	1725	1720	1724	1726	1730	1719	1719	1718	1703	1681
	W	74	89	74	71	69	11	78	78	2	75	73	64	69	2	80	10	Ξ	F	10	9	10	O	10	တ	6	9	9	9	2	10
	Fe	1294	1170	1140	1285	1156	1170	1266	1191	1315	1306	1197	1134	1203	1249	1223	234	228	235	244	253	234	236	244	240	254	261	257	260	266	262
	Ö	112	85	20	83	88	104	66	96	100	107	101	110	113	108	66	10	Ü	14	12	O	12	15	13	12	ග	F	13	13	2	5
	A	1184	1171	1179	1191	1190	1194	1202	1212	1228	1230	1218	1231	1253	1275	1286	-	-	7	-	-	-	-	•		-	0	7	-	0	•
	S	8267	8334	8403	8269	8297	8401	8262	8295	8240	8218	8275	8440	8293	8326	8327	986	266	966	988	995	1004	1003	1000	900	982	995	994	994	1001	1012
	TOTAL	98.62	99.80	98.92	99.17	99.82	99.70	99.63	99.42	98.95	99.44	99.79	98.64	99.27	98.84	99.94	101.33	101.07	101.18	100.32	101.03	100.34	100.69	100.51	100.30	100.05	100.48	100.12	99.35	99.32	100.02
œ	CAO	6.44	6.28	6.45	6.44	6.59	99.9	6.35	6.42	6.51	6.72	6.59	7.44	6.66	6.90	6.94	0.11	0.14	0.14	0.13	0.14	0.15	0.17	0.10	0.15	0.11	0.11	0.13	0.15	0.13	0.11
M08	MGO	14.87	15.64	14.89	15.12	15.65	14.36	15.29	15.27	14.82	14.91	15.30	13.17	14.64	13.70	13.96	48.40	47.79	47.55	47.31	47.16	46.97	46.97	46.89	46.89	46.73	46.56	46.41	45.98	45.58	45.40
	MNO	0.61	0.57	0.62	0.59	0.58	0.65	0.65	0.65	0.58	0.62	0.61	0.53	0.58	0.58	0.67	0.50	0.52	0.52	0.49	0.48	0.48	0.43	0.46	0.45	0.43	0.48	0.46	0.46	0.45	0.49
cooling	FEO	10.83	96.6	99.6	10.82	9.85	9.97	10.72	10.10	11.03	11.00	10.18	9.57	10.18	10.51	10.41	11.42	11.13	11.49	11.80	12.30	11.35	11.50	11.82	11.63	12.23	12.59	12.36	12.41	12.71	12.59
ŏ	CR203	66.0	0.77	0.63	0.83	0.80	0.94	0.89	0.86	0.89	0.95	0.91	0.98	1.01	96.0	0.89	0.53	69.0	0.73	0.63	0.47	0.60	0.76	0.66	0.64	0.44	0.58	0.67	0.67	0.51	99.0
ပ	AL203	7.03	7.09	7.09	7.12	7.20	7.22	7.22	7.29	7.31	7.35	7.35	7.37	7.52	7.61	77.77	0.04	0.02	90.0	0.02	0.02	0.04	0.02	0.0	0.04	0.02	000	90.0	0.03	00.0	0.02
1325	SIO2	glass 57.85	59.47	59.58	58.25	59,15	29.90	58.51	58.83	57.81	57.89	58.85	59.58	58.68	58.58	59.30	olivine 40 33	40.78	40.69	39.94	40.46	40.75	40.84	40.54	40.50	40.09	40.18	40.03	39.66	39.94	40.75

	FS																
	WO																
	FFM		20.10	18.92	23.86	21.26	18.72	20.72	19.35	16.97	18.82	21.06	17.58	19.64	19.11	16.27	16.98
	FEMN		24.28	24.06	22.40	24.04	22.91	25.77	26.13	24.96	22.41	21.94	25.93	20.86	27.20	22.95	25.63
	Ca		545	515	677	534	513	543	503	460	480	551	479	999	475	379	453
	Mg		6351	6550	5953	6181	6536	6354	6530	6801	6533	6302	6814	6459	6880	7129	6882
	Mn		29	64	84	2	99	92	61	26	89	78	24	11	90	61	22
	Fe		1615	1544	1892	1688	1521	1678	1582	1401	1530	1702	1465	1598	1640	1397	1419
	Ö		85	55	88	85	87	88	91	85	83	87	9/	88	0	9/	87
	¥		677	629	099	651	647	652	639	628	624	629	620	617	611	905	296
	Š		7135	7128	7134	7206	7131	7125	7115	7106	7163	7146	7071	7122	7014	2006	7083
	TOTAL		100.45	99.90	100.89	100.57	100.26	99.05	100.03	100.52	100.99	100.84	100.19	100.24	100.67	100.81	99.85
7	CAO		3.50	3.30	4.31	3.43	3.30	3.43	3.22	2.98	3.11	3.54	3.08	3,62	3.05	2.46	2.91
Σ	MGO		29.30	30.16	27.23	28.52	30.21	28.83	30.04	31,67	30.44	29.08	31.52	29.71	31.77	33.26	31.79
	MNO		0.54	0.52	0.68	0.57	0.54	0.52	0.49	0.46	0.56	0.63	0.46	0.62	0.49	0.50	0.45
eating	FEO		13.28	12.67	15.43	13.88	12.53	13.57	12.97	11.63	12.71	14.00	12.08	13.10	13.50	11.62	11.68
_	CR203		0.80	0.48	0.77	0.80	0.76	0.76	0.79	0.75	0.73	0.76	99.0	0.76	0.00	0.67	0.76
ပ့	AL203 CR20		3.95	3.84	3.82	3.80	3.78	3.74	3.72	3.70	3.68	3.67	3.63	3.59	3.57	3.57	3.48
1580°C	SIO2	glass	49.08	48.93	48.65	49.57	49.14	48.20	48.80	49.33	49.76	49.16	48.76	48.84	48.29	48.73	48.78

	FS															
	WO															
	FFM		2.16	2.28	2.35	2.10	2.23	2.13	2.18	2.14	2.15	2.25	2.02	2.10	2.07	
	FEMN		4.69	5.28	5.18	4.31	4.26	4.31	4.41	5.88	4.24	5.49	4.94	3.80	4.25	
	Ca		558	537	548	554	544	548	548	550	550	552	552	547	534	
	Mg		6785	6753	6853	6884	6807	6801	2929	6765	6828	6781	6823	6844	6842	
	Mu		32	99	32	34	37	32	34	52	36	58	58	38	34	
	Fe		151	158	166	148	156	149	151	148	151	157	141	148	146	
	ö		30	42	52	0	20	8	44	4	38	36	52	0	4	
	A		694	20	697	669	698	702	700	706	714	709	709	715	716	
	Si		7694	7704	2099	7666	1991	7685	7691	9692	7654	7681	7677	7675	7654	
	TOTAL		100.58	99.89	100.97	100.58	100.84	100.13	100.80	100,41	100.36	100.90	100.81	100.98	100.84	
•	CAO		3.85	3.68	3.79	3.82	3.76	3.76	3.79	3.79	3.78	3.82	3.82	3.79	3.69	
Ŝ	MGO		33.63	33.25	34.06	34.12	33.79	33.56	33.61	33.49	33.74	33.71	33.91	34.07	33.99	
	MNO		0.28	0.26	0.28	0.30	0.32	0.30	0.30	0.22	0.31	0.25	0.25	0.34	0.30	
eating	FEO		1.33	1.39	1.47	1.31	1.38	1.31	1.34	1.31	1.33	1.39	1.25	1.33	1.29	
Ĕ	CR203		0.28	0.39	0.23	0.00	0.47	0.28	0.41	0.38	0.35	0.34	0.23	0.00	0.38	
ပ	AL203 CR203		4.35	4.36	4.38	4.38	4.38	4.38	4.40	4.42	4.46	4.46	4.46	4.50	4.50	
1580°C	SIO2	glass	56.86	56.56	56.76	56.65	56.74	56.54	56.95	56.80	56.39	56.93	56.89	56.97	56.69	

Synthetic Murchison Fe/Mn 65

	FS																					22.84	20.78	21.06	21.45	19.56	20.31	20.34
	WO																					5.03	4.37	4.62	5.17	5.44	5.72	6.23
	FFM	29.00	32.35	63.57	64.37	45.54	7	1 4	17.64	12.04 24.04	12.48	12.29	12.00	15.88	11.78	13.26	17,43	16.42	11.22	16.11		22.90	22.05	22.26	21.60	19.97	21.14	21.47
	FEMN	15.49	14.22	18.74	21.12	16.14	90	20.00	20.52	24.14	23.60	21.67	27.94	24.04	19.34	23.79	22.42	25.67	21.99	22.45		15.34	16.24	15.76	13.85	14.21	14.10	15.24
	Ca	1235	1296	1949	1884	1851	r	> 0	0 <	ŧ u) (°) m	4	4	4	ന	7	4	က	œ		92	85	87	94	95	100	107
	Mg	2729	3072	529	531	828	702.7	5 0	001	1747	17.45	1729	1769	1665	1742	1720	1640	1643	1770	1633		1395	1427	1417	1423	1454	1418	1369
	W	74	φ 4 τ	5.0	20	47	c	ם נו י	ច ក្	2 ‡	- ;	: <u>;</u>		13	12	F	16	5	9	14		28	52	56	58	56	27	25
	Fe	1145	1338	1017	1049	757	100	777	300	333	250	244	242	317	234	265	349	325	225	316		422	411	413	400	369	387	381
	Ö	94	8 7	8	Ξ	27	(1 0	~ 0	10	~ α	oo	တ	F	æ	9	5	œ	9	50		33	32	58	52	34	28	27
	A	1675	1/26	2652	2728	2714	C	۰ د	- •	•	- +			-	-		-	۴	7	တ		39	42	43	44	25	52	79
	S	8082	7020	8221	8188	8188	000	3 5	4001	000	066	000	984	991	266	995	986	1001	990	993		1977	1971	1974	1976	1966	1973	1980
	TOTAL	98.33	99.05 98.87	97.95	97.53	69.66	9	0000	90.70	00.00	20.00	99.22	99.28	98.85	100.40	98.61	98.07	99.48	98.16	98.61		100.00	100.14	98.96	99.28	99.26	29.62	98.33
	CAO	8.06	8.38	12.77	12.30	12.49	9	0.00	0.22	0.0	2 6	5 5	0.1	0.13	0.17	0.11	0.25	0.15	0.11	0.28		2.39	2.20	2.24	2.43	2.45	2.60	2.74
M10	MGO	12.80	14.28	2.49	2.49	4.16	!	20.74	43.97	45.11	45.70	40.73	47.47	43.79	47.41	45.68	42.43	43.48	47.17	42.81		25.90	26.60	26.10	26.33	27.01	26.38	25.19
	MNO	0.61	0.77	0.45	0.41	0.40	4	0.40	C. 0	000	000	0.00	0.41	0.61	0.58	0.52	0.71	0.59	0.48	0.65		06.0	0.83	0.85	0.94	0.85	0.90	0.81
ooling	FEO	9.57	11.09	8.54	8.77	6.54	3	5 6	14.61	10,01	5.0		11.60	14.85	11.36	12.53	16.12	15.34	10.69	14.78		13.98	13.65	13.57	13 19	12.23	12.85	12.50
8	CR203	0.83	0.70	0.18	0.10	0.25	(0.52	0.37	0.3	0.54	24.0	0.2	0.56	0.41	0.32	0.50	0.39	0.31	0.98		117	1.	101	0.86	108	66 0	0.94
ပ	AL203	9.94	10.15	15.80	16.19	16.65	(0.00	0.02	0.02	70.0	20.0	0.0	000	0.04	0.04	0.0	0.04	0.08	0.30	_	_	_		•			1.83
1180°C	SI02	glass 56.52	53.68	57.72	57.27	59.20	olivine	40.30	39.86	38.68	38.34	74.90	39.36	38.87	40.43	39.41	38.02	39.49	39.32	38.81	atinoapid	54 75	54 77	54 19	54 51	54.43	54.72	54.32

																								_				•	~				
	FS																	21.7	21.4	20.5	27.00	203	20.91	22.8	23.3	20.00	1 0	20.00	Ž,				
	WO																	10.04	11.06	9.53	0.0	8 88	9.72	7.12	10.83	10.07	1	10.13	117				
	FFM	52.53	52.13	53.00	52.66		28.60	28.06	28.78	27.88	25.30	28.01	28.33	26.64	28.02	28.75	29.07	24.13	24 12	24 92	24.96	24.52	23.16	24 62	28.20	23.20	200	22.90	73.6/	0.4.60	34.39	53.21	
	FEMN	23.61	24.02	23.93	23.41		31,11	29.21	33.31	31.97	21.09	34.83	22.18	26.00	23.15	38.22	23.83	19.84	13.66	20.20	13.50	70.8	21.51	10.03	13.41	26.46	2 4 4 6	24.14	09:7	20.64	70.67	20.42	
	Ca	1698	1664	1710	1687		12	F	15	13	11	13	16	17	24	20	18	187	213	188	176	170	0.00	. t	20°	260	200	0 0	150	٥	ж c	27	
	Mg	1406	1429	1377	1394		1390	1402	1376	1377	1446	1408	1380	1412	1354	1366	1369	1253	1269	1310	1208	1267	1339	1317	1234	1000	2007	1324	1306	9	548 565	206	
	Mn	69	68	99	02		18	19	17	17	24	16	52	20	23	15	24	20	5	3 8	3 8	3 6	4 5	*	7 6	† u	2 9	9	99	,	<u> </u>	8	
	Fe	1633	1631	1630	1628		564	554	563	539	498	554	555	520	536	222	571	405	4.3	445	443	0 00	9 0	2 4	1 4	2 6	000	398	422	9	349	609	
	Ö	25	56	30	52		S	7	13	16	7	7	4	æ	æ	12	4	g	ď	5	10	, 5	3 5	2 6	3 6	† C	2 (77	22	ć	32	1062	
	¥	1925	1920	1936	1945		-	2	က	4	5	S	80	=	16	16	16	25	2 6	8 %	3 6	5 6	3 7	4	9 4	2 6	2 :	52	83		1927	457	
	Š	8134	8144	8133	8132		1003	1000	1003	1012	1002	666	1003	1001	1014	1001	994	2043	2010	1080	1001	000	1070	0.70	200	7081	0/8	19//	1974	;	7,7	275	i
	TOTAL	98.50	98.47	98.26	97.92		100.29	100.43	101.05	100.68	100.54	101,36	98.57	60 66	99.31	101.55	100.59	99 94	00.00	100.00	100.32	1007	00 00	40000	00.43	88,00	100.00	100.26	100.96		101.35	99.67	
4	CAO	10.87	10.66	10.92	10.74		0.42	0.38	0.52	0.45	0.39	0.45	0.55	09.0	0.83	0.70	0.62	4 86	9 4	0.40	2 4		4.4	9 4	0.00	5.50	0.40	5.08	3.89	;	0.29	0.00	
M14	MGO	6.47	6.58	6.32	6.38		35,28	35.68	35.17	35.28	37.24	36.21	34.44	35.70	34.21	35.15	34.74	23.36	00.00	24.40	24.40	200	24.70	2	74.04	75.27	23.83	24.62	24.31		17.13	10.79	
	MNO	0.56	0.55	0.55	0.56		0.81	0.85	0.76	0.76	1.07	0.72	1.10	0.89	1.03	99.0	1.07	0.67	5 6	9 6	4 5	,	0 6	0.0	7.57	90.1	0.48	0.54	1.82	i	0.53	1 12	!
heating	FEO	13.39	13.38	13.33	13.28		25.52	25.15	25,64	24.61	22.85	25.40	24.71	23.44	24.15	25.55	25.82	13.46	0 0	13.00	14.73	2 4	12.41	13.01	14.01	14.67	12.88	13.20	14.01		16.43	10.44	2
Ž	CR203	0.22	0.23	0.26	0.22		0.25	0.34	0.64	0.77	0.35	60.0	0.20	0.38	0.39	0.58	0.20	0.22	4 6	77.0	0.0	100	7.7	44.0	0.80	0.82	0.66	0.76	0.89		1.75	42.71	i r
ပ	AL203 CR203	11.20	11.18	11.24	11.26		0.04	90.0	0.09	0.13	0.15	0.17	0.26	0.36	0.51	0.51	0.51						, c								64.41	12.34	4.04
1180°C	SIO2	glass 55 79	55.89	55.64	55.48	erixilo	37.97	37.97	38.23	38.68	38.49	38,32	37,31	37.72	38.19	38.40	37.63	pigeonite 56.78	0 0 0	22.77	04.0	00.00	55.56	24.02	54.53	54.06	54.49	54.83	54.79	spinel	0.81	0.15 8.75	2

	FS												5	13.38	13.10	14.01	14.10	14.13	13.62	1487	ì
	WO												,	1.40	1.45	1,61	2.30	2.97	3.38	60 7	1,04
	FFM		30.00	48.54	39.30	43.37		14.17	14.45	14.31	14.42	14.61	,	13//	13,30	14.24	14.43	14.57	14.10	15.61	5
	FEMN		18.98	19.04	16.82	17.37		21.39	23.96	22.80	22.37	22.68	9	16.49	16.25	15.92	16.82	15.61	16.04	45.22	27.61
	Ca		1394	1651	1507	1645		2	9	2	9	o,	į	/7	28	31	43	56	63	S	36
	Ма		3697	1592	2433	1966		1694	1695	1674	1662	1654		1626	1648	1614	1563	1557	1536	1400	704
	Mn		82	83	26	91		13	7	12	1 3	13	:	16	16	17	16	17	16	•	0
	Fe		1621	1581	1639	1575		282	288	282	282	285		262	255	271	266	268	255	010	9/7
	Ö		82	9/	83	93		Ξ	13	14	12	-	;	20	21	27	30	56	56	č	7
	₹		1575	1580	1700	1931		 -	(-	*	-	9		19	22	32	37	52	73	ć	9
	S		7359	8304	7820	7843		994	686	1002	1009	1007		2006	1994	1989	2005	1992	1990	0404	19/6
	TOTAL		97.16	99.47	97.02	97.16		99.68	98.86	97.69	97.27	97.64		98.64	98.09	97.76	99.27	98.46	98.91	0	88.50
	CAO		8.65	10.69	9.42	10.33		0.18	0.21	0.20	0.20	0.31		0.71	0.74	0.81	1.15	1.48	1.67		2.43
Š	MGO		16.48	7.41	10.93	8.87		45.27	44.82	43.87	43.41	43.34		30.79	31.02	30.19	29.73	29.32	29 13		28.10
	MNO		0.67	0.68	0.77	0.72		0.62	0.56	0.57	0.58	0.58		0.53	0.52	0.56	0.53	0.57	0.53		0.61
oling	FEO		12.88	13.11	13.12	12.67		13.43	13.59	13.16	13.14	13.32		8.85	8.56	9 03	9 03	0 6	8.61		9.40
ర	CR203		0.69	0.67	0.75	0.79		0.54	0.64	0.70	0.60	0.56		0.70	0.75	0.94	108	0.91	0.94		0.76
ပ	AL203		8.88	9.30	99.6	11.02		0.04	0.04	0.04	0.02	0.21							178		
1325°	SIO2	glass	48.91	57.61	52.37	52.76	olivine	39.60	39,00	39.15	39.32	39.32	pigeonite	56.61	55.97	55.47	56 R6	55.92	56.07	20.4	55.86

45	M11
Murchison Fe/Mn	cooling
Synthetic	1180°C

FS		22.56 24.53 24.92 23.00 23.02 22.45
wo		3.96 4.4.68 4.62 5.32 5.32
FFM	54.92 65.47 66.24 67.21 68.42 68.64 66.64 66.64 66.64 66.28 71.67 50.74 50.74 11.80 11.80 11.83 11.83 11.83 11.63 11.63	23.49 25.73 26.19 24.11 24.13
FEMN	16 23 16 23 16 23 16 80 16 80 17 84 17 15 10 17 16 11 18 11 17 76 18 11 18 11 18 12 18 13 18 13	12.23 11.13 12.16 12.16 10.89
Ca	1787 1865 1896 1896 1821 1821 1841 1844 1845 1977 7	78 90 93 87 102
Mg	865 710 623 573 573 574 819 819 745 1021 1705 1705 11669 11669 11669 11669 11669 11669	1407 1330 1307 1322 1361
M	8 8 8 8 8 8 8 8 8 8 8 8 8 8 8 8 8 8 8	36 36 36 36 36 36
Fe	1153 1317 1387 1337 1265 1265 1265 1391 1129 1129 1293 259 316 399 283 283 283 324	443 474 479 431 429
Ö	266 27 27 27 27 27 27 28 28 29 20 20 20 20 20 20 20 20 20 20 20 20 20	26 27 35 30 32 32
¥	2371 2538 2629 2589 2733 2649 2779 3030 3183 1183 4	33 47 73 76 83
S	8259 8098 8098 8104 8108 7954 8134 8134 8152 7736 7777 1005 994 1007 1007 1007	1974 1979 1978 1985 1958
TOTAL	98 67 99 00 99 04 100 99 100 93 100 93 100 93 101 28 101 29	101.71 101.84 101.94 100.05 102.23
CAO	12.03 12.03 12.03 12.03 12.03 13.24	2.36 2.36 2.24 2.34 2.63
MGO	4 08 2 3 3 3 3 3 3 3 3 3 3 3 3 3 3 3 3 3 3	26.50 24.95 24.49 24.57 25.75 24.95
MNO	0.68 0.79 0.79 0.75 0.75 0.68 0.62 0.62 0.63 0.63 0.63 0.63 0.63 0.63 0.63 0.63	1,19 1,28 1,16 1,16 1,15 1,29
FEO	9.70 11.01 11.01 11.40 11.72 8.50 10.72 9.07 11.71 11.71 12.56 18.82 18.82 18.82 18.73 18.	14.87 15.86 16.00 14.29 14.97
CR203	0023 0023 0010 0010 0010 0010 0010 0010	0.94 0.94 1.24 1.05 1.13
AL 203	1415 1508 1568 1568 1616 1616 1616 1600 0002 0002 0002 0002	0.79 1.10 1.72 1.81 1.95
SIO2	glass 58.10 56.35 56.35 56.35 56.35 57.81 55.43 58.43 58.44 57.52 52.52 52.52 59.81 40.43 99.94 40.56 40.67	55.41 55.35 55.26 55.02 55.20 54.38

1325	325°C cooling	oling	Σ	90M															
SIO2	AL203	CR203	FEO	MNO	MGO	CAO	TOTAL	Š	₹	Ö	Fe	M	Mg	Ca	FEMN	FFM	WO	FS	
glass																			
53.44	8.52	0.89	13.59	1.14	14.92	7.61	100.11	7739	1454	102	1646	140	3221	1181	11.77	32.87			
53.44	8.54	0.85	14.23	1.06	14.86	7.68	100,66	7718	1454	26	1719	130	3200	1189	13.25	34.05			
53.51	8.60	0.80	14.07	0.98	14.23	8.09	100.28	7752	1469	85	1705	120	3074	1256	14.18	34.80			
53.44	8.60	0.76	13.73	1.06	15.12	7.72	100.43	7718	1464	87	1658	130	3256	1195	12.79	32.88			
53.31	8.60	0.80	14.29	1.12	14.91	7.82	100.85	7694	1463	91	1725	137	3208	1209	12.60	34.02			
53.59	8.75	0.75	13.23	1,10	14.53	8.06	100.01	7756	1493	86	1601	135	3135	1250	11.88	32.87			
53.53	8.88	0.98	12.45	0.93	15.09	7.98	99.84	7731	1512	112	1504	114	3249	1235	13.22	30.90			
53.70	9.03	0.73	12.72	1.05	15.04	7.89	100.16	7734	1533	83	1532	128	3230	1218	11.96	31.33			
53,59	9.03	0.83	12.80	1.03	15.04	7.85	100.17	7723	1534	95	1543	126	3231	1212	12.27	31.49			
53.66	9.07	0.89	12.71	1.01	15.39	7.93	100.66	9692	1533	101	1525	123	3291	1219	12.43	30.87			
53.87	9.13	0.98	12.84	1.06	15.09	7.89	100.86	7713	1541	111	1537	129	3221	1210	11.96	31.46			
a file sign																			
AD 58	000	0.50	1000	980	45.05	4,0	17001	4003	•	Ç	727	Ş	1604	•	40 00	40 60			
40.64	20.0	9 6	12.04	9 6	20.04	; ;	100	2 5	- •	2 0	000	, t	1670	ŧ c	19.00	0,4			
20.04	20.0	2 0	2 5	000	500.1	- 6		500	- ,	0 9	007	- !	0/01	0 (78.01	4.44			
40.04	0.0	0.09	13.03	0.80	45.43	17.0	101.54	555	- ,	5	6/7		/991	۱۵	16.70	14.19			
40.71	0.04	0.53	14.01	0.74	45.08	0.17	101.28	1006	-	10	290	15	1661	'n	18.69	14.73			
40.58	0.02	0.47	14.32	0.74	45.02	0.18	101.33	1004	-	o	296	16	1661	മ	19.11	15.02			
40.82	0.02	0.51	14.91	0.85	44.85	0.13	102.09	1005	-	10	307	18	1646	n	17.32	15.58			
40.05	0.02	0.57	14.74	0.00	44.74	0.20	101.22	966	+	7	307	19	1659	S	16.17	15.45			
40.11	0.04	0.56	14.32	0.76	44.69	0.15	100.63	1000	₩	;	299	16	1662	4	18.60	15.11			
40.50	00.0	0.63	14.74	0.87	44.45	0.20	101.39	1004	0	12	306	18	1643	2	16.73	15.54			
40.20	0.04	0.54	14.54	0.84	43.94	0.14	100.24	1007	•	Ξ	305	18	1641	4	17.09	15.51			
pigeonite 58.19		0.72	8.38	0.71	32.45	0.73	101,58	1998	16	20	241	27	1661	27	11.65	12.52	1.38	12.35	
57.66		0.85	9.56	0.76	31.92	0.69	101.93	1986	50	23	275	22	1639	52	12.42	14.22	1.30	14.04	
57.76	0.53	0.88	9.92	0.77	31.35	0.64	101.85	1993	22	24	286	23	1613	24	12.72	14.90	1.22	14.72	
57.59		1.08	9.55	0.66	31.34	0.62	101.56	1990	53	30	276	19	1614	23	14.29	14.45	1.19	14.28	

Synthetic Murchison Fe/Mn 20

	FS																		18.75	20.70	19.33	20.53
	WO																		5.42	5.18	4.45	5.83
	FFM	45.97	53.80	56.83	69.09	55.42		16.22	15.61	14.51	11.50	14.77	11.85	13.55	15.66	15.75	13.89		19.83	21.83	20.23	21.80
	FEMN	4.09	5.09	4.71	5.10	4.79							6.62						4.13	4.06	4.20	3.95
	Ca	1626	1760	1739	1872	1783		4	4	4	က	2	2	ო	က	9	ღ		103	66	85	110
	Mg	958	707	582	456	601		1585	1617	1642	1696	1626	1692	1648	1601	1604	1650		1356	1318	1372	1289
	Mn	251	210	226	200	211		51	46	44	32	42	35	41	46	20	44		98	26	88	86
	Fe	1029	1068	1063	1021	1010		317	307	286	225	289	232	265	306	309	273		357	395	370	387
	Ö	34	53	32	20	x 0		σ	6	0 0	æ	9	œ	Ξ	89	6	7		36	93	30	32
	A	2332	2545	2596	2626	2707		0	-	~~	-	-		-	-	-	•		46	20	92	11
	S	8293	8198	8223	8241	8161		1014	1006	1006	1014	1014	1012	1013	1015	1008	1009		1987	1984	1971	1976
	TOTAL	100.02	99.72	99.50	99.46	99.54		100.63	100.29	99,95	100.43	100.16	101.29	101.91	100.80	101.00	100.04		100.88	100.97	101.57	101.90
0 1	CAO	10.84	11.67	11.52	12.41	11.84		0.15	0.13	0.13	0.13	0.17	0.18	0.13	0.13	0.21	0.13		2.69	2.57	2.24	2.88
M12	MGO	4.59	3.37	2.77	2.17	2.87		42.25	43.03	43.76	46.15	43.48	46.34	45.05	42.90	42.96	44.16		25.44	24.59	25.88	24.29
	MNO	2.12	1.76	1.89	1.68	1.77		2.40	2.14	2.07	1.69	1.96	1.69	1.96	2.17	2.35	2.05		2.85	3.20	2.93	3.25
cooling	FEO	8.79	9.07	9.02	8.67	8.59		15.05	14.59	13.60	10.91	13.78	11.33	12.90	14.61	14.76	13.03		11.93	13.15	12.45	12.99
ర	CR203	0.31	0.26	0.31	0.18	20.0		0.47	0.44	0.41	0.39	0.32	0.42	0.56	0.41	0.44	0.37		1.27	1.07	1.07	1.14
ပွ	AL203	14.13	15.34	15.63	15.82	16.34		0.00	0.02	0.02	0.02	0.02	0.02	0.02	0.04	0.04	0.04	_				1.83
1180°C	SIO2	glass 59.24	58.25	58.36	58.53	58.06	olivine	40.31	39.94	39.96	41.14	40.43	41.31	41.29	40.54	40.24	40.26	afinoapin	55.60	55.20	55.45	55.52

	FS																6.12	6.63	6.41	6.93	7.94	
	WO																1.65	2.12	1.50	3.88	4.90	
	FFM	12.07	12.26	15.06	11.50	13.55	12.39	15.76	24.72	27.37	28.48		7.15	7.41	7.58		6.22	6.77	6.50	7.21	8.34	
	FEMN	234	2.76	2.49	2.22	2.46	2.38	2.60	2.51	2.48	2.53		3.17	3.18	3.27		2.02	2.15	2.26	2.30	2.24	
	Ca	90	982	1046	856	882	1007	1280	1766	1905	1965		12	Θ	φ		32	41	58	74	89	
	Mg	4773	5409	3857	4535	4608	5006	4160	2259	1910	1821		1764	1770	1787		1726	1702	1740	1642	1522	
	M	302	289	296	281	313	316	322	340	343	341		44	46	46		29	9	55	22	64	
	£	607	1967	736	626	771	753	838	853	849	861		139	145	150		118	128	125	132	144	
	ŏ	110	26	06	95	91	91	65	91	99	23		1	12	თ		22	20	25	22	22	
	A	106	1235	1246	1256	1397	1650	1755	2149	2249	2365		4	-	-		24	36	45	82	120	
	S	7681	7293	8030	7840	7597	7153	7335	7711	7759	7693		1010	1006	966		1998	1992	1973	1969	1984	
	TOTAL	100 14	100.27	100.62	99.87	98.05	99.35	95.76	97.51	98.88	100.24		100.83	100.50	69.66		101.43	101,94	101.47	101.77	100.37	
LO.	CAO	603	6.45	7.02	5.72	5.72	6.56	8.20	11.29	12.35	12.90		0.46	0.24	0.22		0.88	1.13	0.80	2.03	2.41	
MOS	MGO	87.00	25.53	18.60	21.79	21.49	23.43	19.15	10.38	8.90	8.59		48.91	48.83	48.81		34.09	33.71	34.31	32.37	29.56	
	ONM	254	2.40	2.51	2.38	2.57	2.60	2.61	2.75	2.81	2.83		2.14	2.22	2.21		2.04	2.08	1.92	1.99	2.20	
ooling	FEO	بر 20	6.70	6.33	5.36	6.41	6.28	6.88	6.99	7.05	7.24		6.88	7.15	7.32		4.17	4.52	4.39	4.64	5.00	
ខ	CR203	80 0	0.50	0.82	0.83	0.80	0.80	0.56	0.79	0.60	0.47		0.57	0.64	0.48		0.83	0.76	0.92	0.83	0.79	
ပ်	AL203	7.22	7.37	7.60	7.63	8.24	9.77	10.22	12.49	13.26	14.11		0.13	0.04	0.02		0.59	_			-	
1325°C	SIO2	glass	51.32	57.74	56.16	52.82	49.91	50.34	52.82	53.91	54.10	quivilo	41.74	41.38	40.63	piaeonite	58.83	58.83	58.02	57.87	57.46	

		FS															30.21	29.58	31.35	29,10	29.50									
		WO															4.59	4.65	5.19	6.32	5.90									
		FFM	61.95	58.61	65.63	63.44		35.77	35.86	36.34	35.48	36.17	35.73	35.88	38.38		31.67	31.03	33.07	32.01	31.35		71.24	70.02	69.00	68.86	04.1			
		FEMN	55.76	63.22	64.83	60.89		77.76 79.08	78.93	79.16	69.11	82.88	74.04	68.93	80.20		51.50	44.58	44.45	50.76	59.37	:	73.19	68.22	76.41	77.30	80.03			
		Z	0	21	o	, <u>_</u>		w 4	t	0	2	9	ı Qı	'n.	4 4		ю	10	0.0	o +-	-	,	0 4	o (4		۰ ۰	n	12637	13004	
		Na	92	6	S 2	3 8		~ <	0	0	-	7	7	0 (o c	,	·-	0	0 (ν ω	က	,		- 64	0	← 0	>		0 0	
		Ca		1541				o و	ာ့တ	10	=	10	10	o (9 C	2	88	89	00 9	123	15		ۍ 4	1 10	æ	9 4	4	4	00	
		Mg		1678 15				1247	54	46	47	46	37	20	7 7	5				1231	•		301	23	36	9 9	70	0	၈ဝ	
		Min		38 16		•		0 0 5 5	Ċ											12 12	·		11					0	0 4	
		Fe		2430				699	90	16	91	-	93	e e	2.0	2	88	70	07	585	75	1	773	32	7	51	:	21	54	
		ŏ		35 24											0 1					18 5									26 10954 46 13175	
		A								_	-	m	ന	2,	- 0	ı				20 20			457 1441 467 1416						00	
		TI /		82 1901				- c	4 ***	0	0	*~ ·	- (0 0	0 0)				v 60			24 45						5 0	
		S						C1 L2	ōΦ	4	_	<u>ල</u>	ო :	4.0	ے د <u>ر</u>	ı	ž.	œ.	·- 9	ວເວ	ζ.		~ u						00	
			·	3 7589		•		1012									·	•		1985								·		
		TOTAL	99.9	98.78	2.88	98.9(100 42	100.4	99.4	101.13	99.66	100.67	99.1	101.0		101.08	100.0	98.60	100.40	99.8		98.03	97.0	96.87	97.86	98	134.84	135.46	
	M29	욅	0.00	0.17	0.00	90.0		0.22	0.06	0.00	0.23	0.28	0.25	0.21	0.10		0.10	0.33	0.0	0.09	0.04		0.00	0.06	0.05	0.26		72.48	74.74	
	_	NA20	0.27	0.31	0.39	0.44	;	000	0.00	0.00	0.01	0.04	0.04	0.00	000		0.01	0.00	0.00	0.0	0.04		0.0	0.03	0.00	0.02	0.00	0.00	000	
	_	CAO	66.6	9.51	10.57	10.11	,	0.32	0.31	0.35	0.39	0.35	0.34	0.32	0.33		2.29	2.27	2.50	3, 13	2.91		0.14	0.08	0.21	0.17	5.0	90.0	0.00	
	eating	MGO	6.80	7.44	6.16	6.39	į	30.72 29.45	31.06	30.51	31.17	30.54	30.79	30.55	29.68		23.15	22.85	21.74	22.45	22.72	4	5.84 8.4	6.25	6.63	6.56	0.10	0.00	0.00	
	ž	MNO	0.36	0.30	0.32	0.33		0.39	0.39	0.39	0,44	0.37	0.41	4 6	0.59	<u>:</u>	0.37	0.41	0.43	0.37	0.31		0.36	0.39	0.35	0.34	U.31	0.00	0.00	
		FEO	20.33	19.21	19.61	20.35	9	30.72	31.18	31.27	30.80	31.06	30.75	30.72	33.21		19.30	18.51	19.36	19.02	18.64	;	26.69 26.96	26.95	27.09	26.62	17.97	31.91	60.56	
no:	IW-0.5	CR203		0.29				0.07												0.62			52.66						0.15	
rchis	≧	- }																												
Synthetic Murchison		2 AL203		10.66				0 0												0.66			11.19						000	
theti	ပ္ပင	T102	0.75	0.72	0.82	0.76		0.05												0.10			0.93					0.10	0.00	
Syn	1180°C	SI02	glass 50.57	50.17	49.70	49.75	olivine	37.18	37.14	36.65	37.67	36.52	37.59	30.08	36.82		pigeonite 54.77	53.98	52.80	53.96	54.13	spinel	0.21	0.13	0.39	0.40	60 n	metal 0.06	0.00	

(Exp type) (Exp #)

B. Series II Experiments (Temp) $f(O_2)$

	FS																													
	WO																													
	FFM	6	24.00	63 10	60.09	60.95	59.88	62.87	99'09	63.54	62.28	61.19		35.31	34.47	34.98	34.49	35.80	35.39	33.75	34.52	33.53	34.77	34.24	36.11	34.64	34.43		75.37	76.09
	FEMIN	9	00.00	50.00	61.25	54 70	55.81	63.51	56.31	53.40	54.95	60.29		67.42	66.39	67.47	64.36	59.61	59.65	56.16	68.14	65.18	67.79	69.89	63.62	65.14	64.00		82.49	74.50
	ź	c	o •	ŧ ţ	<u>-</u>	. 0	4	4	0	7	46	0		4	4	2	9	က	S	4	4	က	S	4	4	က	4		7	7
	S.	900	2 5	1 9	5 4	128	152	119	164	144	126	149		0	0	0	0	-	0	0	0	0	0	-	0	0	0		0	0
	్ట	90	200	1697	1582	1610	1581	1684	1570	1661	1666	1630		10	10	10	10	=	10	10	o	10	10	10	12	9	Ξ		7	S.
	М	1574	1422	1368	1552	1509	1555	1466	1517	1407	1441	1553		1258	1287	1265	1268	1269	1281	1297	1292	1297	1281	1295	1229	1264	1255		224	285
	Ψ	Ş	? ?	4 4	4	44	43	40	43	47	45	42		10	10	0	9	12	12	12	9	9	10	10	F	9	10		တ	5
	Fe	0450	2476	2428	2486	2424	2385	2550	2405	2536	2453	2514		692	683	989	673	714	708	299	687	629	688	679	701	675	664		711	946
	Ö	÷	1 0	r g	37	34	58	37	58	37	33	4		7	9	9	7	80	6 0	7	9	2	7	9	_	9	7		1558	1440
	A	, 0	107	1826	1818	1815	1813	1899	1842	1897	1892	1936		-		-	7	5	7	7	7	5	7	က	၈	ო	7		448	362
	F	ď	G a	8 8	6	88	95	94	98	93	69	92		-	-	0	-	-	-	-	~-	0	0	-		0	0		12	15
	Š	7665	2007	7694	7652	7701	7702	7552	7702	7591	7607	7516		1006	266	1006	1009	287	983	966	992	1005	966	994	1014	1012	1018		10	6
	TOTAL	08	00.43	90.00	86.66	99.89	100.49	98.46	100.54	99.24	99.68	97.72		100.75	100.57	99.45	100.82	100.50	101.41	100.38	100.47	101.64	102.04	101.61	100.28	101.60	101.18		119.99	95.76
M35	읮	ç	3 6	0.13	0.01	00.0	0.03	0.03	0.00	90.0	0.38	0.00		0.17	0.18	0.24	0.29	0.14	0.24	0.17	0.19	0.15	0.25	0.19	0.17	0.13	0.18		0.10	0.08
2	NA20	97	5 5	0.48	0.49	0.44	0.53	0.40	0.57	0.49	0.43	0.50		0.00	0.00	0.00	0.00	0.01	0.00	0.00	0.00	0.00	0.00	0.01	0.00	0.00	0.00		0.00	0.00
	CAO	97	200	10.51	986	10.05	9.95	10.28	9.88	10.23	10.33	9.89		0.34	0.34	0.34	0.36	0.38	0.35	0.35	0.31	0.34	0.34	0.34	0.41	0.34	0.38		90.0	0.13
poling	MGO	9	9 6	809	6.95	6.77	7.03	6.43	6.86	6.23	6.42	6.77		31.32	32.02	31.11	31.70	31.29	31.89	32.28	32.07	32.78	32.27	32.55	30.43	31.87	31.59		5.31	5.22
8	ONM	2,4	200	0.36	0.32	0.35	0.34	0.31	0.34	0.37	0.35	0.32		0.45	0.45	0.44	0.46	0.52	0.52	0.52	0.44	0.45	0.45	0.43	0.48	0.46	0.46		0.36	0.41
	FEO	10 34	10.50	19.26	19.85	19.39	19.22	19.94	19.39	20.01	19.48	19.54		30.73	30.26	30.07	29.99	31.40	31.42	29.58	30.37	29.71	30.90	30.44	30.93	30,35	29.82		30.08	30.94
IW-0.5	CR203	34	, c	0.33	0.31	0.29	0.25	0.31	0.25	0.31	0.28	0.34		0.34	0.28	0.29	0.34	0.35	0.38	0.34	0.29	0.26	0.31	0.29	0.31	0.29	0.31		69.72	49.80
	AL203	10.07	10.13	10.28	10.30	10.30	10.37	10.54	10.54	10.62	10.66	10.68		0.04	0.04	0.04	90.0	90.0	90.0	90.0	90.0	90.0	90.0	0.08	90.0	90.0	0.21		13.45	8.39
ပ	T102	0.83	0.77	0.84	0.80	0.78	0.85	0.82	0.77	0.82	0.82	0.82		0.03	0.03	0.02	0.03	0.05	0.05	0.05	0.03	0.02	0.02	0.03	0.03	0.02	0.02		0.57	0.55
1180°C	SIO2	glass	50.25	51.05	51.09	51.52	51.92	49.40	51.94	50,10	50.53	48.86	olivine	37.33	36.97	36.90	37.59	36.30	36.50	37.03	36.71	37.87	37.44	37.25	37.44	38.06	38.21	100	Spinel 0.34	0.24

	FS		26.28	25.91 25.91	
	WO		11.	5.11	
	FFM	55.11 55.11 55.18 55.18 55.18 55.18 56.10 50.24 60.10 60.10 60.10 50.10	30.30 30.97 27.69	27.30 27.30 63.12	4.0
	FEMN	51.76 74.480 74.480 74.480 74.480 74.69 74.69 74.69 75.75 75.56 75 75 75 75 75 75 75 75 75 75 75 75 75		37.75 37.75 61.42	
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	S	7560 7574 7532 7498 7590 7590 7575 7575 7575 7579 75619 7620 995 1000 1005 998 998	989 998 1978	19/4 1966 23	32 3
	TOTAL	98.46 97.06 97.20 97.20 98.62 98.64 99.15 99.15 99.15 99.25 99.25 99.25 99.25 99.25 99.25 99.25 99.25 99.25 99.25 99.25 99.25	99.89 99.83 98.75	98.40 98.40 95.92	96.13 136.49 137.18
M61	<u>Q</u>	0.000 0.000	0.00	0.10	0.09 29.34 30.32
Σ	NA20	0.77 0.78 0.089 0.889 0.889 0.082 0.981 0.992 0.992 0.003 0.003	0.03	0.03	
	CAO	9.96 10.12 10.12 10.12 10.12 10.26 10.37 10.37 10.30 9.93 0.31 0.31 0.32 0.32	0.25	2.53 2.53 0.15	000
eating	MGO	7.25 7.25 7.27 7.23 7.23 7.23 7.23 7.23 7.23 7.23	34.54 33.92 24.18	24.31	0.0 80.0 20.0
þe	MNO	0.035 0.035	0.39	0.43	000
	FEO	16.25 15.88 15.89 16.67 16.66 16.38 16.83 16.83 16.83 16.83 16.83 16.83 16.83 16.78 17.71 15.17 15.17 27.11	26.98	16.44 16.44 24.26	23.88 106.52 106.68
W-1.0	CR203	0.38 0.37 0.32 0.32 0.32 0.34 0.35 0.35 0.35 0.37 0.37	0.47	0.80	0.20 0.20 0.13
=	AL203 (12.15 12.19 12.19 12.19 12.15 12.15 13.09 13.19 13.19 13.19 13.19 13.10	0.08	0.87	15.44 0.06 0.00
ပ	TI02 /	0.00 0.00 0.00 0.00 0.00 0.00 0.00 0.0		0.10	1.08 0.07 0.00
1180°C	SIO2	9gass 50.54 50.04 49.44 49.44 49.47 50.10 50.10 50.10 51.39 51.39 51.39 51.39 51.39 51.39 51.39 51.39 51.47 37.31	36.97 37.29 pigeonite 53.48	53.63 52.93 spinel 0.68	0.39 metal 0.15 0.00

	FS	l								9	22	7					
											14.52						
	WO									8.78	90.6	8.66					
	FFM	20.97	32.12 36.67	13.64	14 16	13.95	13.77	12.83	13.55	15.61	15.56	15.77	9	27.00	30.36	40.80	
	FEMN	14.27	18.99 16.88	27.41	23.88	23.51	21.95	21.26	24.92	6.21	6.05	6.30	4	4.0	30.02	31.52	
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	S	6283	6242 6508	991	982	666	1005	686	1019	2009	2014	2015	'n) (C	5	65	24
	TOTAL	98.81	100.42 98.27	99.46	100,79	99.37	98.74	101.84	100.12	100.68	100.39	100.54	97 15	98.47	98 46	97.80	129.30 127.89
M30	욅	0.07	0.00	0.10	0.05	0.00	0.19	0.04	0.04	0.00	0.00	0.00	0	800	0.16	0.00	7.33
_	NA20	0.29	0.63	0.03	0.00	0.00	0.01	0.00	0.03	0.02	0.01	0.02	5	9 0	000	0.02	0.05
_	CAO	16.89	16.96 16.68	0.28	0.34	0.32	0.56	0.27	0.28	4.40	4.51	4.32	0.10	200	0.11	0.84	0.04
heating	MGO	3.30	1.60	45.08	44.09	44.19	43.43	47.26	44.90	27.58	27.34	27.45	10.44	13.78	13.28	12.96	0.03
<u> </u>	MNO	0.11	0.13	0.46	0.54	0.54	0.56	0.58	0.50	1.49	1.51	1.48	95	0.80	0.51	0.51	0.18
	FEO	1.59	1.71	12.77	13.06	12.86	12.45	12.49	12.62	9.37	9.26	9.44	15.96	15.09	15.70	16.28	121.39 120.52
W-1.5	CR203	0.05	0.04	0.86	2.25	1.13	0.86	0.57	0.53	0.74	0.75	0.74	60 79	43.60	45.10	40.71	
	AL203	31.84	31.26	0.28	0.81	0.42	0.68	0.15	90.0	0.10	0.08	0.02	6 14	21.63	19,63	20.91	0.02
ပ	T102	0.05	0.07	0.13	0.18	0.12	0.17	0.02	0.08	0.07			2.65	3 1	3.49	3.47	0.00
1180°C	SIO2	glass 44.62	46.16	olivine 39.47	39.47	39.79	39.83	40.46	41.08	oigeonite 56.91	56.88	66.99	spinel	23	747	5.09	netal 0.11 0.13

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	FFM	16.26	3202	29 93	13.66	13.22	13.42	13.75	40.19	
	FEMN	11 65	26.41 18.35	15.53	22.07	15.13	18.01	17.12	22.35	
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	e N	1 25	\$ \$	124	c	0	-	0	-	11
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	Σ	1056	987 559	457	1700	1675	1634	1656	613	0
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	Ē.	208	5 5 5 6 5 6 7	23	27.1	258	255	287	425	21464
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	₹	5015	5534 5547	5769	6	15	35	9	748	n
	F	56	48	18	-	5	6	7	84	86
	Š	6278	5921 6076	5944	985	066	666	1001	10	0
	TOTAL	99.18	100.59	100.92	98.37	99.27	89.56	99.00	97.31	133.13
M47	욅	0.00	0 0	0.07	0.12	0.14	0.16	0.11	90.0	13.12
-	NA20	0.49	0.48	0.48	0.00	0.01	0.03	0.00	0.02	0.04
D 1	CAO	16.64	17.07	17.89		0.77			0.09	0.08
heating	MGO	5.03	2.69	2.21	44.86	44.68	43.85	44.12	13.01	0.00
£	ON	0.15	0.0	0.11	0.57	0.80	0.67	0.73	0.71	00.00
	FEO	1.77	2.23	1.73	12.74	12.26	12.22	12.66	16 07	119.12
IW-1.5	CR203	0.05	0.13	0.05	0.60	0.49	0.44	0.54	43.40	0.25
	TIO2 AL203 CR203	30.22	33.70	35.28	0.32	0.52	1.18	0.34	20.07	0.01
ပ္	TI02	0.25	0.0	0.17	0.07	0.25	0.17	0.12	3.53	0.53
1180°C	SI02	glass 44.59	43.60	42.84	olivine 38.76	39.35	39.99	40.02	spinel 0.33	metal 0.00

	FS																							29.22	.88	32	97	19										
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		61.34	960	0 6	7 05	32		56	87	34	12	75	92	54	84	63	61	79	44	53	84	35.62		30.90 5						66.69	95	52	74	30				
	FFM																																					
	FEMIN	61.10	90.9	0.40 0.10	523	71.1		71.1	82.0	74.5	9.69	83.1	83.1	67.9	68.8	76.6	72.9	77.94	67.0	67.7	75.9	74.3		52.14	46.1	60.2	46.8	53.9		73.03	86.7	73.2	78.2	86.0				
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	Ca	1546	1597	1370	1594	1543		o	თ	2	တ	თ	თ	=	0	80	=	80	10	6	10	6		110	82	103	82	92		9	4	4	4	တ	C	o c	0	
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	F.	2760	2/25	7000	2777	2841		719	702	704	708	205	704	702	709	704	734	693	691	697	703	669		280	571	579	588	584		752	800	784	774	756		17/98	12716	
	Ö	49	- ;	4 5	4 4	54		7	7	7	90	7	7	7		13	9	80	2	6	7	7		13	21	50	19	7		1368	1571	1539	1429	1167	ţ	5 5 4	ō ao	
	¥	1717	1/59	1700	1832	1777		0	-	-	-	7	7	ო	e	4	œ	+		-	-	0		o	58	22	31	-		532	331	364	463	713	,	0 0	20	
	F	77	9 6	0 0	9 6	82		0	0	0	0	0	0	0	0	0	0	-	0	0	0	0		0	က	7	7	7		27	27	25	28	27	•	ne	വ	
	S	7541	7003	207	7520	7500		266	1010	892	1007	1003	1008	266	1001	993	1003	1018	1014	1005	1007	1005		1971	1970	1976	1980	2009		9	7	ന	4	7	•	= 0	4	
	TOTAL	99.59	98.29	9 0	99.58	99.98		100.78	99.55	100.03	99.84	100.59	100.61	100.57	100.83	100.58	101.35	100.65	100.32	100.71	99.53	100.35		99.64	100.82	100.84	99.51	100.63		99.43	98.68	98.09	98.80	97.87		136.51	136.61	
M26	ON N	0.17	רבים	0.00	0.07	0.00		0.15	0.10	0.00	0.15	0.01	0.00	0.22	0.18	0.15	0.15	0.15	0.13	0.22	0.09	0.28		0.08	0.04	0.14	0.00	0.08		0.22	60.0	0.03	0.04	0.07	1	70.54	64.98	
2	NA20	0.34	0.42	9 6	0.49	0.40		00.0	0.00	0.00	0.00	0.00	0.00	0.00	0.00	0.00	00.0	0.01	00.0	0.01	0.02	0.00		0.03	0.03	0.03	0.03	0.05		0.00	0.00	0.01	0.01	0.00	,	00.0	0.03	
	CAO	9.47	9.00	9.40	9.2	9.47		0.32	0.31	0.35	0.31	0.32	0.32	0.38	0.34	0.29	0.39	0.28	0.34	0.31	0.33	0.31		2.77	2.10	2.63	2.07	2.43		0.13	0.11	0.10	0.11	0.14		0.00	000	
eating	MGO	7.46	7.7	4 6	6.45	7.08		31.02	30.64	31.52	30.54	31.22	30.99	31.27	31.27	31.26	30.00	30.77	30.99	31.17	30.67	31.05		23.64	24.08	23.52	23.16	22.84		6.20	5.21	5.55	909	6.86		0.0	0.00	
he	MNO	0.35	4.5	20.0	0.30	0.31		0.44	0.37	0.41	0.44	0.37	0.37	0.45	0.45	0.40	0.44	0.39	0.45	0.45	0.40	0.41		0.36	0.40	0.31	0.40	0.35		0.36	0.31	0.36	0.34	0.31		0.05	0.03	
	FEO	21.66	21.16	50.12	21.73	22.33		31.71	30.76	30.94	31.04	31.17	31.17	30.99	31.39	31.07	32.51	30.79	30.58	30.87	30,76	30.86		19.01	18.69	18.92	18.98	19.11		26.63	27.25	26.71	26.95	27.02		65.81	71.27	
IW-0.5	CR203	0.41	0.00	70.0	35.0	0.45		0.34	0.32	0.34	0.39	0.34	0.35	0.34	0.04	0.63	0.45	0.39	0.22	0.40	0.33	0.34		0.44	0.73	0.69	0.65	0.50		51.29	56.61	55.48	52.65	44.12		0.09	0.05	
_	AL203 C	9.56	90.0	000	9.09 10.19	9.91		0.00	0.02	0.02	0.04	90.0	90.0	0.08	0.08	0.13	0.25	0.02	0.04	0.04	0.03	0.01		0.21	0.68	0.52	0.70	0.26		13,38	8.01	8.81	11.45	18.08		0.00	0.02	
O	TIO2 A	0.67						0.00	0.00	0.00	00.0	0.00	0.00	0.00	00.0	00.0	00.0	0.05	0.02	00.0	0.01	0.01						0.07		1.05	1.03	0.95	408	1.06		0.02	0.03	
1200°C	SIO2	glass 49.50					olivine															37.08	ationapid	53.10	53.95	53,99	53.47	54.95	spine	0.17	90.0	60.0	0.41	0.21	metal	0.00	0.04	

	FS														,	90.00	24.63	25.30	26.07	24.79						
	WO														Š	4.0	4.04	3.92	3.98	4.57						
	FFM	57.25	59.06	58.30	30.73	30.42	30.95	30.49	30.37	31.22	30,17	30.51	30.42	30.77	•	70.14	25.67	26.33	27.15	25.98		75.85	/0.5/			
	FEMN			42.34		56.05														36.19		84.82	11.69			
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	Na	232	204	216	C	0	0	0	0	-	0	-	0	, -	c	7	7	7	~ ~~	7		0 1			00	
	S	1637	1565	1596	œ	ග	æ	G	G	10	9	თ	우	89	ŕ	2	79	11	77	89		4.	4	(00	
	Mg	1598	1538	1544	1355	1354	1353	1358	1362	1337	1359	1363	1357	1345	1	13//	1382	1371	1336	1355		275	334	•	55	
	Mn	52	9 2	27	0	=======================================	12	=	2	10	9	-	12	ō	ç	5	<u></u>	4	15	13		# ;	71	c	7 1	
	Fe	2209	2289	2230	909	596	612	900	299	611	591	603	599	602		487	482	495	503	480		897	831	į	10051	
	ö	45	36	8 4	œ	9	7	7	ထ	œ	9	9	9	80	į	ç	16	18	16	20		1435	1385		5.6	
	A	1990	1933	1974	-	- 74	7	2	7	ო	7	7	ო	2	,	20	7	9	32	38		404	444		o 0	
	F	96	3 2	35	c	0	0	0	-	-	0	0	0	0	Ó	7	7	က	7	က		19	28	(၁'က	
	S	7571	7657	7632	1003	1007	666	1003	1002	1006	1007	666	1003	1006	3	1881	1991	1982	1993	1984		10	10	,	39	
	TOTAL	97.13	100.13	98.77	101 07	101.78	99.58	100.77	101.68	100.20	101.29	101.53	100.05	100.96	,	91.19	101.42	101.20	100.91	99.60		66.66	98.69		141.08 140.92	
M41	9	0.05	0.04	0.03	90.0	0.15	0.11	0.11	0.15	0.13	0.10	0.09	0.11	0.17	i	50.0	0.05	0.04	0.05	0.00		60.0	0.14	1	82.37 82.61	
Σ	NA20	0.78	0.71	0.74	00	00.0	00.0	0.00	0.00	0.01	0.00	0.01	0.00	0.01	i i	0.03	0.03	0.03	0.01	0.03		0.00	0.01		0 0	
	CAO	96.98	9.84	9.97	0 28	0.32	0.29	0.31	0.32	0.34	0.35	0.32	0.35	0.29	0	5.00	5.06	1.99	1.99	2.27		0.11	0.11		00.0	
ooling	MGO	7.00	6.95	6.89	34 44	34.70	33.79	34.42	34.85	33.64	34.72	34.79	34.17	34.17	1	25.73	25.92	25.60	24.87	24.95		5.32	6.48	;	0.02	
ၓ	MNO	0.40	0.39	0.40	0.43	0.48	0.53	0.49	0.45	0.44	0.44	0.48	0.53	0.41	;	0.44	0.44	0.45	0.49	0.43		0.36	0.41		0.00	
	FEO	17.25	18.44	17.74	27.43	27.25	27.24	27.13	27.30	27.42	26.93	27.44	26.86	27.26		16.39	16.11	16.47	16.71	15.76		30.93	28.70		58.54 57.87	
IW-0.5	CR203	0.35	0.31	0.34	0.38	0.31	0.32	0.34	0.31	0.37	0.29	0.31	0.29	0.38	į	0.54	0.56	0.63	0.56	0.70		52.37	50.98		0.06	
	AL203 (11.03	11.05	11.14	0.04	90.0	90.0	90.0	90:0	0.08	0.08	0.08	60.0	0.17		0.43	0.51	0.72	0.83	0.89		9.90	10.88		0.02	
ပွ	TI02 /	0.83	0.82	0.81	0.02	0.02	0.02	0.02	0.03	0.03	0.02	0.02	0.02	0.00						0.10		0.63	0.70		0.00	
1200°C	SIO2	glass 49.46	51.58	50.78	olivine 37.99	38.49	37.22	37.89	38.21	37.74	38.36	37.99	37.63	38.10	pigeonite	55.45	55.67	55.17	55.32	54.47	Spine	0.28	0.28	metal	0.06	

	FS			15.15 14.25 14.36 14.67 13.67 14.25 14.25 14.25		
	WO			5.84 5.39 5.02 5.02 5.02 5.04 5.04 5.04 5.04		
	FFM	38.18 40.47 38.20 38.29	14.60 14.30 14.73 14.41 14.19 14.23 14.15	16.09 15.06 15.12 15.38 14.77 14.23 15.39 15.02 14.77	34.69 32.95 36.55 37.33 34.51	
	FEMN	22.33 23.45 16.89 24.45	24.06 21.45 22.97 23.34 22.91 21.43 30.54	20.25 21.27 16.07 22.17 15.95 12.52 21.74 16.47 15.86	29.04 33.02 34.24 38.93 35.77	
	ž	8877	000000-0	00411-110610	-	2029 2134 1998 2038 2049 2273 1955 2216
	R _a	102 131 118	0000000	000000000	0000-	00000047
	S	1647 1703 1617 1619	rrr9r8r	113 105 99 92 86 76 86 100 98	0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0	r 2 0 0 7 0 0 r
	Ма	1517 1435 1482 1555	1665 1681 1681 1682 1685 1660 1695	1527 1553 1573 1575 1576 1570 1518 1555 1555	621 711 666 591 675	23 696 10 10 498 128 16
	Ψ.	64 43 66 04	25525510	21 13 13 14 15 15 17 17 17 17 17 17 17 17 17 17 17 17 17	2111 00	44004400
	e L	963 1005 951 990	287 283 294 280 277 277 281	295 278 283 289 271 264 278 278 266	336 355 390 357 361	21817 20628 21953 21872 21129 21099 21904
	ত	56 48 48	22224121	23 26 28 28 33 34 35 36 37	796 812 1028 907 865	30 30 27 17 17 17 17 17 17 17 17 17 17 17 17 17
	A	2042 2009 2022 2008		23 38 50 53 53 66 66 75	1201 1007 880 1073 1068	0 10 10 10 10 10 10 10
	F	77 77 77	00000++0	0001446000	15 17 16	000000+8
	S	8238 8247 8281 8253	1005 998 987 1004 996 996 996 996	1991 1979 1952 1948 1977 1965 1979 1979	3 77 81 8	37 252 4 4 131 205 24 4
	TOTAL	98.62 99.63 98.60 99.34	101.32 101.34 97.99 101.36 100.50 100.12 100.12	100.69 100.27 100.27 101.29 100.20 101.87 101.26 101.26	97.10 98.39 98.75 100.26 95.95	130.22 133.47 132.87 136.73 133.91 132.83 131.97
M27	9	0.03 0.04 0.06 0.06	0.00 0.00 0.00 0.00 0.23 0.10	0.00 0.14 0.00 0.00 0.00 0.00 0.00 0.00	0.04 0.05 0.05 0.06	11.44 12.61 11.47 12.06 12.04 13.21 11.17
2	NA20	0.37 0.48 0.43 0.41	0.00 0.00 0.00 0.00 0.00 0.00 0.01	000000000000000000000000000000000000000	0.00 0.00 0.01 0.01	0.00 0.00 0.00 0.00 0.00 0.00 0.00 0.0
	CAO	10.86 11.31 10.67 10.75	0.25 0.27 0.27 0.24 0.25 0.29	3.01 2.80 2.62 2.28 2.28 2.28 2.28 2.28 2.28 2.28	0.01 0.31 0.18 0.17 0.18	0.0000000000000000000000000000000000000
eating	MGO	7.19 6.85 7.03 7.42	45.25 45.25 44.24 45.37 45.32 44.72 46.09	29.13 29.75 29.00 30.24 29.48 30.46 29.86 29.46 29.46	14.03 16.17 14.61 13.53	0.07 2.22 0.03 1.58 0.40 0.02 0.05
he	MNO	0.36 0.36 0.47 0.34	0.57 0.63 0.58 0.58 0.62 0.62 0.51	0.49 0.44 0.59 0.54 0.57 0.57 0.58 0.58	0.46 0.43 0.44 0.37 0.39	0.01 0.00 0.00 0.01 0.01 0.00
	FEO	8.14 8.55 8.04 8.42	13.89 13.69 13.73 13.74 13.46 13.31 13.61	9.05 9.08 9.08 9.21 9.13 9.21 9.20 9.20	13.53 14.38 15.26 14.59	118.32 117.26 121.24 124.37 119.44 117.93 117.83
IW-1.5	CR203	0.50 0.43 0.48 0.43	0.63 0.60 0.61 0.64 0.69 0.56 0.56	0.83 0.92 1.07 1.05 0.94 1.17 0.99 1.13 1.20	33.92 34.80 42.56 39.18 35.78	0.18 0.09 0.18 0.18 0.01 0.01 0.10
	AL 203	12.24 12.13 12.13	0.04 0.06 0.08 0.01 0.11 0.06	0.55 0.91 1.19 1.27 1.27 1.53 1.61 1.68	34.35 28.97 24.43 31.08 29.65	0.00 0.02 0.04 0.04 0.04 0.08
ပ	T102	0.72 0.79 0.72 0.68	0.0000000000000000000000000000000000000	0.00 0.00 0.00 0.27 0.18 0.21 0.30 0.22	0.65 0.65 0.75 0.68 0.68	0.00
1200°C	SIO2	glass 58.21 58.70 58.56 58.71	olivine 40.69 40.43 38.49 40.71 40.33 40.66	pigeonite 56.63 56.52 55.30 55.75 56.24 56.29 55.62 57.06 55.92	spinel 0.11 2.61 0.47 0.60 0.26	metal 0.17 1.20 0.02 0.04 0.62 0.96 0.11

	FS														24 09	20.28	21 94	200
	WO														4 29	4 54	4 44	00 1
	FFM	45 16	45.36	41.34	42.32	4191	41.20		21.21	21.28	20.63	20.36	20.09		25 17	21.25	22.95	24.25
	FEMN	24 57	25.55	22.30	21 74	20.75	22.25		31.97	30.76	28 10	29.34	29.46		29.66	24.31	29.31	25.12
	ž	ĸ	0	9	က	12	9		С	C			0		0	0	-	-
	Na	164	152	187	159	141	151		С	0	0	0	0		2	-		,
	Ca	1659	1633	1606	1620	1714	1655		00	60	7	œ	80		83	86	87	8
	Ма	1850	1797	1715	1847	1842	1842		1559	1530	1548	1541	1558		1374	1410	1431	1381
	Mn	64	9	26	65	99	90		13	14	4	14	13		16	16	15	8
	Đ.	1577	1542	1248	1403	1376	1333		423	417	406	398	395		467	385	431	448
	Ö	52	45	48	42	48	47		10	6	10	10	10		23	21	24	26
	Ā	2291	2290	2359	2369	2411	2414		2	2	7	e	က		30	35	36	43
	F	102	101	108	66	107	106		0	-	-				က	2	4	4
	Ø	7523	7594	7725	7584	7509	7562		989	1001	1002	1010	1002		1986	2005	1966	1972
	TOTAL	99.59	99.37	99.18	99.97	99.34	99,35		100.63	101.12	100.78	100.75	100.27		101.26	101.38	99.93	101.52
M39	ON	0.04	0.00	0.05	0.03	0.10	0.05		0.00	0.01	0.05	0.03	0.01		0.01	00.0	0.04	0.04
2	NA20	0.58	0.54	0.67	0.57	0.50	0.54		0.01	00.0	00.0	0.00	0.00		0.03	0.01	0.01	0.03
	CAO	10.63	10.48	10.44	10.52	11.03	10.70		0.29	0.29	0.27	0.28	0.31		2.17	2.28	2.25	2.50
ooling	MGO	8.52	8.29	8.01	8.62	8.52	8.56		40.81	40.39	40.77	40.72	40.94		25.75	26.83	26.56	25.97
O	MNO	0.52	0.49	0.46	0.53	0.54	0.49		0.61	0.63	0.67	0.63	0.62		0.52	0.53	0.48	0.59
	FEO	12.94	12.68	10.39	11.67	11.35	11.04		19.75	19.63	19.07	18.72	18.50		15.62	13.05	14.25	15.01
IW-1.5	CR203	0.45	0.39	0.42	0.37	0.42	0.41		0.47	0.44	0.48	0.48	0.51		0.82	0.77	0.85	0.91
	TIO2 AL203 CR203	13.34	13.36	13.94	13.98	14.11	14.19		90.0	0.08	0.08	0.09	60.0		0.72	0.85	0.91	1.02
ပ	TIO2	0.93	0.92	8	0.92	0.98	0.98		0.02	0.03	0.03	0.03	0.05				0.15	
1200°C	SIO2	glass 51.64	52.22	53.80	52.76	51.79	52.39	ofivino	38.59	39.62	39.36	39.77	39.24	pigeonite	55.49	56.89	54.43	55.30

	FS																									
	WO																									
	FFM		55.28	51.50	54.80	53.37	52.70	55.57	53.24	52.83	55.11	55.56	53.51		30.12	30.92	31.10	29.87	30.75	31.09	31.40	31.20	29.37	30.22	30.31	29.95
	FEMN		40.85	41.45	37.90	40.65	37.71	41.99	38.38	40.11	44.97	41.33	45.37	,	49.17	49.83	52.80	51.86	55.80	51.62	56.02	52.50	48.53	50.30	54.53	45.69
	Ž		12		,-	9	4	9	9	S	7	10	ဌာ	•	-	-		S	_	7	0	0	5	S	2	S
	Na Na		27	33	33	27	34	33	34	34	39	33	36		0	0	0	0	-	0	0	0	-	-	0	0
	်		1044	961	1039	1034	1037	1057	1034	1055	1025	1016	882	,	7	9	9	9	9	9	မ	9	9	9	9	9
	Mg		2587	2827	2644	2716	2710	2606	2721	2750	2633	2575	2831		1367	1339	1347	1361	1349	1359	1344	1343	1366	1371	1360	1369
	Μg		81	74	87	79	83	80	83	79	74	90	74	•	12	12	12	-	=	12	=	12	12	7	Ξ	65
	F.		3298	3080	3311	3198	3111	3359	3194	3169	3323	3321	3343		594	909	613	585	604	618	620	614	573	599	969	591
	ŏ		85	91	101	96	66	102	6	91	92	103	87		13	12	7	=	=	=	12	Ξ	=	12	13	12
	¥		1212	1213	1217	1249	1252	1234	1262	1271	1234	1248	1285		-	-	Ψ-	·	-	-	-	-	-	-	7	2
	F		99	28	99	92	92	61	62	62	61	63	54		-	0	0	0	0	0	0	0	0	0	0	C
	S		7446	7484	7396	7403	7441	7376	7396	7379	7403	7414	7340		998	1008	1001	1007	1006	882	1000	1003	1009	993	1000	866
	TOTAL		100.48	99.34	100.49	97.79	97.65	100.12	97.17	97.36	101.01	101.03	101.19		100.87	100.48	101.45	99.90	102.27	100.47	100.37	100.66	100.85	100.73	101.26	101 30
M43	ON N		0.10	0.01	0.01	0.05	0.03	0.05	0.05	0.04	90.0	0.08	0.04	;	90.0	0.05	90.0	0.24	0.05	0.08	0.00	0.01	0.22	0.24	0.24	0 22
2	NA20		60.0	0.11	0.11	0.09	0.11	0.11	0.11	0.11	0.13	0.11	0.12		000	0.00	0.00	0.0	0.01	0.00	0.00	0.00	0.01	0.01	00.0	000
	CAO		6.34	5.83	6.30	6.13	6.16	6.37	60.9	6.23	6.25	6.20	5.41		0.24	0.21	0.20	0.20	0.21	0.22	0.21	0.21	0.22	0.22	0.21	000
heating	MGO					11.57									34.69	33.84	34.27	34.31	34.70	34.16	33.79	33.91	34.85	34.67	34.62	34.89
Ĕ	ONW		0.62	0.57	29.0	0.59	0.62	0.61	0.62	0.59	0.57	0.62	0.57		0.54	0.54	0.52	0.50	0.49	0.53	0.49	0.52	0.53	0.53	0.49	0.58
	FEO		25.65	23.93	25.72	24.29	23.68	25.94	24.10	23.97	25.96	25.95	26.19	:	26.89	27.25	27.81	26.26	27.69	27.71	27.80	27.65	26.05	27.00	27.06	26.84
IW-0.5	CR203		97.0	0.75	0.83	0.77	0.80	0.83	0.72	0.73	97.0	0.85	0.72	,	0.64	0.56	0.54	0.53	0.51	0.53	0.56	0.54	0.53	0.58	0.61	0.58
	AL203 C		69.9	69.9	6.71	6.73	6.76	6.76	6.76	6.82	6.84	6.92	7.14		0.04	0.04	0.04	0.04	0.04	0.04	0.04	0.04	0.04	0.04	90.0	0.08
\circ	TIO2 A		0.50	0.50	0.57	0.55	0.55	0.52	0.52	0.52	0.53	0.55	0.47		0.03	0.00	0.02	0.00	0.00	0.00	0.00	0.02	0.02	0.02	0.02	0.00
1300°C	SIO2	lass												•												

	FS							4.36	4.88	2.93	14.67	5.31	2.68	2.55
	WO										1.35			
	FFM	32.01 30.31	1.95	80.8	3.96	4.34	13.80	4.55	5.10	3.12	14.87	5.51	2.89	2.74
	FEMN	16.10 15.34 3					21.39	•	•		21.14	•	•	•
	Z	φο <i>ί</i>	·	-	-				ເ		0			
	e N	4-1	<u>π</u> ∞	0	0	0 0	00	0	0	0	, -	0	0	0
	Ca	1405	375	φ	2	ro é	ဥ ဖ	56	28	28	27	52	8	28
	Mg	3025 14		678	069	686	1674	990	604	657	1632	622	647	657
	W W	922 3		·	•		7 £			•	14		•	•
	E E	1467	1479	277	276	284	276	285	288	253	287	300	246	244
	ŏ	99		-	13	77	<u> </u>	20	71	20	33	54	22	24
	₹	1619 1649	1627	₩.	7	0.0	o 4	5	18	50	77	73	55	23
	F	74 79	75	0	0	0 (o 	-		7	7	7	7	2
	S	7646 7628	7633	1003	266	986	100	1979	2000	1991	1985	1985	1994	1989
	TOTAL	99.39 98.56	99.91	100.07	100.50	99.93	100.86	101.08	98.85	101,41	100.72	101.40	99.41	100.46
M45	ON	0.05	0.02	0.04	0.04	0.03	0.03	0.05	0.10	0.04	0.00	0.05	0.05	0.00
Σ	NA20	0.05	0.03	000	00.00	0.00	0.0	0.00	0.00	00.0	0.01	0.00	000	0.00
_	CAO	9.02 8.47	8.88	0.21	0.20	0.20	0.22	0.69	0.73	0.76	0.71	0.67	0.81	0.76
eating	MGO	13.96	14.20	45.10	45.58	45.15	45.68	32.00	30.29	32.25	31.37	31.34	31.45	31.97
he	MNO	0.74	0.73	0.63	0.59	0.56	0.62	0.52	0.49	0.53	0.46	0.45	0.58	0.56
	FEO	12.07	12.23	13,28	13.28	13.57	13.43	9.80	69.6	8.76	9.85	10.34	8.38	8.40
IW-1.5	CR203	0.76	0.78	0.57	0.64	0.61	0.69	0.72	0.76	0.75	0.82	0.77	0.79	0.89
=	- 1	9,45	9.55	0.04	90.0	90.0	0.15	0.36	0.43	0.49	0.51	0.51	0.53	0.57
O	TIO2 AL203	0.68	0.69	0.00	0.02	0.00	0.02	0.05	0.05	0.07	90.0	90.0	0.08	0.08
300°C	SIO2	glass 52.61 52.24						ite						

1400°C	ပ		IW-0.5		=	heating		≥	49																
3102	TIO2	TIO2 AL203 CR203	CR203	FEO	MNO	MGO	CAO	NA20 NIO		TOTAL	Ø	j=	₹	ర	e e	M _n	Σ	ő	e Z	ž	FEM	FFW	0 X	FS	
																		l			i				
lass																									
18.52	0.46	6.26	0.89	25.44	0.65	12.11	6.18	0.03	0.05	100.59	7444	•	1132		3264		•	716	6	9	38.64	53.35			
8.58	0.47	6,16	0.88	25.39	0.63	12.23	6.35	0.02	90.0	100.77	7442	•	1112		3253			742	9	7	39.79	53.08			
18.44	0.45	6.41	0.86	25.45	0.65	12.01	5.91	0.01	0.03	100.22	7451	•	1162		3274			374	က	4	38.66	53.56			
98.8	0.46	6.38	0.87	25.71	99.0	11.94	99.9	0.05	0.04	101.60	7432	53	1144		3271	85 2	•	1086	9	2	38.46	53.94			
18.71	0.42	6.33	0.87	25.41	0.65	12.01	5.98	0.01	0.03	100.42	7473	•	1145	106	3261		2747	383	60	4	38.60	53.52			
1																									
Minne																									
17.93	0.00	0.04	0.47	20.55	0.40	40.31	0.18	00.0	0.00	99.88	983	0		9	446		558	2	0	0	50.73	22.14			
98.38	0.08	0.04	0.45	20.91	0.48	40.23	0.14	00.0	0.00	101.72	1000	8		Ø	444		523	4	0	0	43.01	22.46			
17.87	0.03	0.04	0.45	20.74	0.45	40.62	0.18	00.0	0.11	100.49	978	,-	-	0	448		563	3	0	2	45.51	22.16			
18.34	00.00	0.04	0.47	20.21	0.40	39.88	0.15	00.0	0.05	99.54	994	0	,	10	438		542	4	0	-	49.89	22.04			
18.02	0.00	0.04	0.50	21.11	0.41	40.18	0.18	0.03	0.10	100.57	982	0	•	9	456	6	1547	S	2	7	50.84	22.66			
8.51	0.02	90.0	0.44	21.21	0.43	39.96	0.15	0.01	0.04	100 83	066	c	c	đ	456		533	٧	c		70 87	22 84			

Synthetic H-CM

																								_		
	FS																					26.34	26.26	22.32	22.54	23.21
	WO																					8.09	5.23	5.56	4.14	3.23
	FFM	57 61	57.84	59.55	62.30	61.89	62.55	61.34		27.17	32.26	27.77	31.80	31.87	26.53	27.90	26.27	31.73	26.76	32.12		27.93	27.29	23.21	23.31	23.39
	FEMN	32.64	36.64	35.39	32.64	31.39	36.94	34.45		47.73	40.83	43.13	31.57	46.25	42.81	44.27	40.06	39.25	44.60	40.87		44.35	40.23	15.39	22.12	23.60
	Na	155	152	136	122	146	116	123		-	0	0	0	0	0	-	0	0	0	0		2	ო	0	0	0
	Ca	1459	1473	1540	1535	1570	1526	1546		æ	80	7	o	60	80	80	7	თ	7	თ		144	26	94	98	98
	Mg	1499	1482	1285	1210	1216	1203	1243		1419	1295	1402	1332	1309	1434	1423	1443	1319	1415	1303		1260	1323	1409	1405	1453
	Σ	92	28	26	65	99	24	90		=	15	13	50	13	12	13	13	16	12	15		7	12	58	50	19
	e.	2126	2112	1974	2106	2083	2105	2067		534	624	544	630	619	522	556	519	620	521	624		493	501	434	433	450
	ŏ	33	33	37	34	4	30	53		9	9	9	9	S.	9	S	80	2	œ	40		16	15	19	18	16
	A	2151	2164	2154	2156	2195	2178	2209			-	-		-	-	-	-		7	7		12	17	27	27	32
	Ξ	92	56	64	63	64	97	64		0	0	0	-	0	0	0	0	0	0	0		-	-	က	7	7
	S	7684	9692	7831	7807	7754	7805	7768		1006	1024	1012	866	1020	1006	994	1002	1013	1015	1014		2022	2008	1980	1993	1966
H33	TOTAL	99.42	99.27	29.65	100.12	99.31	100.20	99.27		99.48	100.68	99.01	100.31	98.05	102.00	99.88	101.25	100.33	98.71	100.98		99.87	98.87	101.99	101.36	101.30
I	NA20	0.54	0.53	0.48	0.43	0.51	0.41	0.43		0.01	0.00	0.00	0.00	0.00	0.00	0.01	0.00	0.00	0.00	0.00		0.07	0.04	0.00	0.00	0.00
	CAO	9.21	9.29	9.82	9.78	9.91	9.74	9.78		0.27	0.28	0.25	0.31	0.28	0.28	0.28	0.27	0.31	0.25	0.32		3.68	2.45	2.48	2.25	1.79
heating	MGO	6.80	6.72	5.89	5.54	5.52	5.52	5.65		36.06	32.73	35.38	33.36	32.25	37.41	36.08	37.36	33.20	35.77	32.98		23.20	24.13	26.65	26.51	27.31
Ĕ	MNO	0.52	0.46	0.45	0.52	0.53	0.46	0.48		0.50	0.68	0.56	0.88	0.58	0.56	0.56	0.59	0.70	0.52	0.68		0.36	0.40	0.94	0.65	0.63
	FEO	17.19	17.07	16.13	17.19	16.85	17.21	16.75		24.17	28.12	24.46	28.14	27.17	24.28	25.11	23.94	27.83	23.49	28.15				14.65		
IW-0.5	CR203	0.28	0.28	0.32	0.29	0.35	0.26	0.25		0.31	0.28	0.26	0.28	0.23	0.28	0.23	0.37	0.26	0.37	0.50		0.57	0.50	0.67	0.63	0.58
-	TIO2 AL 203 CR203	12.34	12.41	12.49	12.49	12.60	12.64	12.70		0.02	0.05	0.02	0.04	0.04	0.04	0.04	0.04	0.04	90.0	0.08		0.28	0.40	0.64	0.64	0.77
ပ္	T102	0.58	0.50	0.58	0.57	0.58	0.58	0.58		0.02	0.00	0.00	0.05	0.02	0.02	0.00	0.00	0.00	0.02	0.02	_	0.05				
1180°C	SIO2	glass 51.96	52.01	53.51	53.31	52.46	53.38	52.65	olivine	38.12	38.57	38.08	37.25	37.48	39.13	37.57	38.68	37.99	38.23	38.25	pigeonite	55.49	54.62	55.86	56.05	60.33

	FS													28.23	31.83	30.12	26.17	26.14	21.82		
	WO													5.29	4.54	4.00	3.38	3.17	5.78		
	FFM	54.19	54.07	53.92 54.39		25.26	26.17	32.78	34.68	34.56	26.48	26.01		29.81	33.34	31,38	27.09	26,99	23.16		98'29
	FEMN	31.28	30.96	32.43		52.51	58.11	90.69	59.77	53.35	54.24	51.76		39.30	43.16	42.59	45.97	36.62	33.07		56.84
	S.	141	169	138		-	0	0	0	0		0		0	0	0	0	0	'n		0
	S	1423	1434	1413		7	9	4	10	80	9	9		103	88	79	92	61	112		ಣ
	Mg	1701	1682	1682		1455	1459	1311	1276	1278	1454	1442		1274	1221	1292	1340	1355	1395		343
	Ā	67	99	64		on.	o	F	=	13	9	9		14	14	14	=	14	13		13
	F.	2091	2057	2082		495	520	645	683	682	527	510		547	618	282	502	909	424		751
	ŏ	49	848	4 0 0		S	9	7	7	7	S	7		22	77	17	16	56	21		1521
	¥	2119	2205	2150		-	-		-	 -	7	7		23	56	56	34	38	40		328
	F	09	61	0		0	0	0	, -	0	0	0		7	-	-	7	7	က		52
	S	7637	7587	7634		1012	266	1008	1003	1004	966	1009		1996	1993	1974	2004	1981	1978		10
H34	TOTAL	98.70	99.26	98.80		100.71	101.73	100.05	100.36	100.52	100.73	101.06		100.40	100.69	100.34	101.28	100,30	102.43		96.39
I	NA20	0.49	0.59	0.48	i	0.01	0.01	0.00	0.00	0.00	0.01	0.00		0.00	0.00	0.00	0.00	0.00	0.07		0.00
	CAO	8.92	9.04	8.87		0.25	0.20	0.15	0.35	0.27	0.21	0.22		2.62	2.24	2.01	1,69	1.58	2.98		0.08
cooling	MGO	7.66	7.62	7.59	1	37.72	37.95	32.76	31.70	31.82	37.39	37.39		23.40	22.29	23.56	25.10	25.07	26.56		6.48
Ö	MNO	0.53	0.53	0.53		0.43	0.41	0.48	0.50	0.56	0.44	0.45		0.45	0.46	0.45	0.36	0.45	0.43		0.44
	FEO	16.79	16.62	16.75	;	22.87	24.13	28.71	30.27	30.26	24.17	23.59		17.91	20.11	19.41	16.76	16.69	14.40		25.33
IW-0.5	CR203	0.42	14.0	0.42	į	0.26	0.29	0.34	0.31	0.32	0.26	0.35		0.76	0.72	0.60	0.58	0.89	0.77		54.25
	TIO2 AL203 CR203	12.07	12.64	12.27	4	0.04	0.04	0.04	0.04	0.04	90.0	90.0		0.53	0.59	09.0	0.74	0.91	96.0		8.60
ပ	T102	0.54	0.55	0.55		0.02	0.02	0.02	0.03	0.00	0.02	0.00									0.93
1180°C	SIO2	glass 51.28	51.26	51.36	olivine	39.11	38.68	37.55	37.16	37.25	38.17	39.00	pigeonite	54.66	54.23	53.66	55.97	54.64	56.16	spinel	0.28

	FS																				21.30	23.95	24.07	23.80	23.60	24.74
	WO																				4.30	4.19	3.98	4.54	3.98	3.58
	FFM	54.32	55 02	54 27	55.25	54.37	54.96		27.20	27.89	27.50	27.10	28.03	28.08	27.58	27.45	27.33	28.23	27.71		22.26	24.99	25.07	24.93	24.57	25.66
	FEMIN	40.73	40.71	40.18	36 22	38.64	46.19		58.69	54.50	52.57	38.57	50.84	55.72	61.19	61.24	52.61	49.69	65.75		22.30	51.57	34.62	43.81	29.42	36.19
	Na	278	238	257	325	299	277		0		0	-	0			-			0		6	-	ო	7	ო	0
	S	1495	1521	1484	1454	1496	1482		7	80	7	œ	9	9	9	7	Q	œ	89		89	86	79	8	78	70
	Mg	1513	1434	1492	1457	1421	1471		1439	1417	1427	1448	1429	1410	1432	1423	1429	1391	1434		1516	1458	1409	1416	1405	1396
	Μu	46	44	45	51	45	40		6	10	10	14	Ξ	10	თ	6	10	-	89		20	თ	14	7	16	13
	Fe	1853	1809	1825	1863	1746	1844		541	552	545	543	561	554	549	545	541	552	553		440	489	476	474	463	486
	Ö	37	52	45	44	20	44		80	æ	7	9	10	7	_	80	-	80	9		23	2	18	15	7	\$
	¥	2392	2409	2395	2411	2412	2415		-	-	-	-	-	-	•	-	-	-	7		52	28	58	28	34	32
	F	20	9	94	20	2	75		0	0	0	-		0	0	0	0	0	0		4	7	ო	-	m	
	S	7587	7631	7619	7595	7650	7592		982	666	666	987	885	1003	966	1002	266	1011	385		1926	1940	1974	1970	1975	1976
H60	TOTAL	97.70	97.02	29.67	98.44	97.88	98.86		99.24	99.46	99.13	100.13	99.61	98.90	99.71	98.34	99.86	98.60	98.47		98.36	97.74	98.38	99.45	97.50	98.34
Ī	NA2O	96.0	0.82	0.89	1.13	1.04	0.97		0.00	0.01	0.00	0.01	0.00	0.01	0.01	0.01	0.01	0.01	0.00		0.13	0.01	0.04	0.03	0.0	0.00
	CAO	9.35	9.47	9.30	9.16	9.42	9.39		0.24	0.28	0.25	0.29	0.22	0.22	0.21	0.25	0.22	0.28	0.28		2.24	2.14	1.99	2.31	1.96	1.78
heating	MGO	6.80	6.42	6.72	09.9	6.43	6.70		36.35	35.78	35.96	36.78	36.06	35.42	36.28	35.65	36.30	34.90	35,85		27.52	26.21	25.62	26.03	25.37	25.37
ž	MNO	0.36	0.35	0.36	0.41	0.36	0.32		0.41	0.45	0.46	0.63	0.49	0.44	0.40	0.39	0.46	0.49	0.37		0.63	0.30	0.44	0.35	0.50	0.43
	FEO	14.85	14.43	14.65	15.04	14.09	14.97		24.37	24.84	24.49	24.61	25.23	24.83	24.79	24.19	24.51	24.66	24.64		14.23	15.67	15.43	15.53	14.90	15.76
W-1.0	CR203	0.31	0.44	0.38	0.38	0.48	0.38		0.37	0.39	0.35	0.29	0.23	0.35	0.35	0.39	0.54	0.38	0.28		08.0	69.0	0.61	0.53	0.70	0.61
	TIO2 AL203	13.60	13.64	13.64	13.81	13.81	13.91		0.02	0.02	0.02	0.02	0.02	0.02	0.04	0.04	0.04	0.04	90.0		0.57	0.64	0.64	0.64	0.77	0.81
ပ	T102	0.62	0.53	0.57	0.63	0.63	0.68		0.02	0.02	0.00	0.05	0.03	0.02	0.00	0.00	0.02	0.02	0.00					0.05		
1180°C	SIO2	glass 50.85	50.92	51.15	51.28	51.62	51.54	olivine	37.46	37.63	37,55	37.40	37.33	37.59	37.63	37.42	37.76	37.82	36.99	pigeonite	52.11	52.01	53.51	53.98	53.16	53.55

	FS								9	20.04	10.04	15.10
	WO								5	3 5	2 6	5 27
	FFM	36.48	36.58	35.85	37.05 37.05	14.81	15.36	14.95	16.67	10.01	2.0	15.94
	FEMN	19.20	21.60	16.90	16.30	28.22	28.29	27.74	35 36	10.00	19.02	21.14
	S	88	108	86	801	c	0	0	·	4 +	- (*)
	S	1832	1794	1//6	1831	Œ	9	9	102	2 6	117	103
	Μ̈́	1811	1816	1843	1704	1715	1675	1703	1591	1568	1530	1537
	Mu	26	20	3 8	64	Ξ	=	Ξ	6.	2 4	<u></u>	4
	ą.	1072	1076	1065	1040	300	306	304	321	318	200	294
	Ö	51	44	\$ ¥	33 1	80	0	00	24	5	3:	18
	Ā	2967	2939	2841	2859	-	-	-	23	27	6	32
	F	75	76	4 6	77	7	0	0	4	2	2	2
	S)	7253	7292	7408	7408	976	993	982	1948	1968	1971	1981
46	TOTAL	97.26	99.23	98.36	99.91	100.27	100.45	100.00	97.83	97.33	99.61	99.00
I	NA20	0.31	0.39	0.30	0.39	0.00	0.01	0.00	0.03	0.02	0.0	0.02
~	CAO	11.67	11.67	11.21	12.02	0.23	0.23	0.22	2.61	2.10	3.07	2.68
eating	MGO	8.29	8.49	8.31	8.04	45.84	44.88	45.43	29.29	28.85	28.98	28.87
Ē	MNO	0.45	0.41	0.75	0.53	0.50	0.51	U.51	0.41	0.52	0.54	0.46
	FEO	8.75	76.8 8.90	8.68	8.75	14.29	14.61	14.33	10.53	10.44	10.05	9.85
IW-1.5	CR203	0.44	0.39	0.39	0.29	0.40	0.47	0.42	0.82	0.72	0.77	0.82
	TIO2 AL203 CR203	17.18	16.72	16.70	17.06	0.04	0.05	90.0	0.53	0.62	0.75	0.82
180-0	TIO2	0.68	0.78	0.70	0.72	0.08	0.02	20.0	0.14	0.09	0.07	0.08
200	- 1	glass 49.50	51.54	51.32	52.11	olivine 38.89	39.67	50.00	pigeonite 53.47	53.98	55.34	55.47

	FS																24 09	23.96	25.00	24.92	24.91					
	WO																4 49	3.12	404	3.29	3.20					
	FFM	53.40	54.00	54.92	54.57	58.76	58.59	27 17	27.62	28.16	28.02	27.65	27.66	27.13	27.69	27.10	25.22	24.73	26.05	25.77	25.74		63.54	61.98	62.43	
	FEMN	28.31	31.73	28.53	33.05	30.48	28.29	44 41	38.73	37.76	40.34	36.64	48.85	43.92	39.61	47.47	24 12	20.35	32.14	30.78	32.40		50.29	51.39	50.27	
	Na	122	124	145	136	128	153	c	0	0	0	0	7	7	0	-	4	-	7	e	-		0	0 0	0	
	S	406	446	412	476	1535	522	œ	9	80	9	ო	7	œ	7	9	87	. 6	78	94	62		7	x 0	2 1	
	Mg					1370		1435	1407	1408	1399	1407	1410	1416	1407	1419	1365	1413	1355	1390	1379		364	381	374	
	Ψu					. 29		5	4	15	4	15	F	12	14	=	Œ.	23	15	16	15		5	2 5	<u> 6</u>	
	Fe	2076	2075	2053	2047	2048	2027	540	542	558	550	543	543	532	544	532	467	472	483	488	483		658	641 668	643	
	Ö	4	45	7	4	18	10	o	10	œ	9	œ	7	თ	6	9	17	33	17	18	16		1422	1351	1328	
	A	2224	2249	2258	2264	2429	2430	0	0	-	-	*	7	7	-	7	22	39	28	20	22		496	559 485	616	
	F	•				89		o	0	0	0	0	-	0	0	0	-	8	-	7	-		4 :	4 6	5 5	
	S	7558	7539	7625	7560	7556	7558	266	1008	666	1007	1009	1006	1008	1006	1010	1999	1967	1999	1990	2000		58	es a	ာတာ	
=	TOTAL	98.46	99.02	98.28	98.42	98.05	98.80	99.18	00.41	100.14	00.05	99.30	00.27	99.29	99.13	99.29	99.82	99.42	00.13	00.80	28.67		98.83	97.98	97.48	
Ή	NA20	0.42	0.43	0.50	0.47	0.44	0.53			0.00								0.01						900		
	CAO	8.79	60.6	8.84	9.23	9.57	9.57	0.22	0.22	0.27	0.22	0.10	0.24	0.27	0.25	0.22	2.24	1.57	2.01	1.67	1.58		0.20	0.27	0.13	
heating	MGO	7.81	7.69	7.26	7.38	6.14	6.15	36.21	35.96	35.75	35.57	35.57	36.01	35.88	35.50	35.99	25.24	25.97	25.12	25.90	25.22		7.28	7.31	7.48	
he	MNO	0.58	0.52	0.57	0.49	0.53	0.57			99.0								0.75					0.46	44.0	0.45	
	FEO	16.63								25.24								15.46					23.43	22.90	22.91	
IW-0.5	CR203	0.37								0.38											0.54			54.03		
=	TIO2 AL 203 C	12.64	12.85	12.85	12.87	13.77	13.89	0.00	00.00	0.02	0.02	0.04	90.0	90.0	0.04	0.05	0.51	0.91	0.66	0.47	0.51		12.55	12.17	15.57	
ပ	1102	0.58	0.58	0.58	0.58	0.60	0.68	0.00	00.0	0.00	0.02	0.02	0.05	0.02	000	0.01		70.0					0.57	0.57	09.0	
1200°C	SIO2	glass 50.64	50.77	51.15	50.66	50.49	26.06	olivine 37.50	38.42	37.82	38.19	38.04	38.32	38.06	37.85	38.17	pigeonite 55,11	53.89	55.22	55.28	54.53	spinel	0.77	0.98	0.28	

	FS		24.30 25.16	
	WO		4.61 5.42	
	FFM	55.79 55.79 55.79 55.79 55.74 55.74 55.74 55.74 55.74 55.74 55.74 56.74	25.48 26.60	63,35
	FEMIN	38.70 39.870 39.870 39.15 39.15 39.77 39.77 48.29 48.29 48.29 49.31 49.3	31.85 30.84	54.30
	Na	338 339 339 337 337 336 336 336 336 336 000 000 000 000 000	e 4.	0
	Ca	1335 1366 1366 1341 1367 1369 1376 1376 7 7 7 7 7 7	901	4
	Mg	1592 1599 1658 1658 1651 1621 1635 1548 1578 1578 1578 1578 1578 1578 1578 157	1390	385
	Min	61:64:64:64:64:64:64:64:64:64:64:64:64:64:	15	5
	Fe	1910 1932 1932 1806 1919 1919 1919 1833 577 577 570 560 560 562 563 565 565 565 565 565 565 565	481	687
	ŏ	04004086088 04004086088 040004086088	16	1620
	Ι	2208 2208 2208 2208 2201 2201 2201 2210 2210	18 34	286
	i=	8,8,6,6,6,6,6,6,6,6,6,6,6,6,6,6,6,6,6,6	9.19	5
	S	7752 7771 7711 7693 7654 7654 7652 7695 7695 1006 1006 1000 1000 1000 1000	1983 1980	10
H40	TOTAL	99 90 99 90 99 90 99 50 99 81 99 14 100 17 99 17	102.00	98.45
Ì	NA20	2.1.1.2.1.1.2.2.1.1.2.2.1.1.2.2.1.1.2.2.1.1.2.2.1.1.2.2.1.2.2.1.2	0.05	0.00
	CAO	8 8 8 3 2 2 4 2 4 2	2.39	0.11
cooling	MGO	7.33 7.56 7.56 7.56 7.58 7.58 7.58 7.18 7.10 7.21 7.21 7.21 7.21 7.21 7.21 7.21 7.21	26.18 24.84	7.43
8	MNO	0.00 0.00 0.00 0.00 0.00 0.00 0.00 0.0	0.50	0.43
	FEO	15 68 15 86 175 18 175 18 175 18 175 17 175 17 175 17 175 18 175 17 175 18 175 17 175 18 175	16.13 16.24	23.65
IW-0.5	CR203	0.026 0.026 0.026 0.026 0.036 0.037	0.58	58.96
=	AL203 (721	0.43	6.99
ပ	TI02 A	0.55 0.55 0.55 0.55 0.55 0.05 0.00 0.00	0.07	0.58
1200°C	SIO2	glass 53.21 52.95 52.95 52.95 52.31 52.39 52.39 52.46 52.56	pigeonite 55.67 54.83	spinel 0.30

	FS																				!	13.57	10.10	10.00	14.51				
	WO																				,	3.60	0.0	0.0	2.56				
	FFM	34.42	34.05	33.56	38 22	37.84	40.21	39.73	34.71		16.41	14.26	14.05	14.16	13,99	14.04	14.24	14.40	13.89	14.58	;	14.08	70.07	2.00	14.89		34.84	36.47	
	FEMN	16.42	22.17	19.28	24.29	23.23	16.77	17.90	15.29		25.20	25.86	21.16	23.77	23.35	23.82	25.90	25.60	20.17	26.03	;	16.69	13.46	13,40	18.70		28.38	28.31	
	Ra	61	44	49	99	73	102	102	71		0	0	0	0	0	0		0	0	0	•	m +	- ,	-	7		o -	7	
	S	1707	1723	1685	1685	1536	1563	1572	1623		9	2	ß	9	9	9	9	7	9	9	i	28	3 5	4	20		9 1	16	
	Mg	1971	2008	1935	1605	1483	1459	1478	1980		1658	1706	1710	1694	1708	1704	1692	1657	1682	1659		1602	1570	8/0	1604		655	616	
	Ā	99	48	52	42	9	19	22	71		13	=	13	12	12	12	=	=	14	=	:	9 0	2 0	<u> </u>	5		£ £	13	
	Fe	1069	1062	1004	1019	927	1022	1012	1091		328	286	282	281	280	280	283	281	273	285		265	- a	200	283		374	361	
	Ö	65	29	26	47	38	20	49	90		6	Ξ	F	6	6	1	6	F	o	7		5 28	9 6	70	23		936 986	973	
	F	2793	2742	2718	2327	2160	2164	2164	2700			7	-	7	-	7	-	7	7	-		89 8	7 4	0	30		976 921	893	
	F	11	7	65	51	49	48	46	71		0	0		0	0	0	0	0	0	0	•	nc	v =	7	7	:	= 2	10	
	S	7358	7396	7504	9767	8291	8214	8209	7460		066	986	982	995	066	066	966	1013	1004	1010		19/0	1050	000	1982	!	55 55	88	
H32	TOTAL	98.86	100.44	101.12	99.18	100.05	100.88	100.36	101.07		100.31	99.92	100.42	100.20	100.87	100.86	100.88	100.23	101.30	100.48		101.37	101.20	07.70	100.37	;	95.34	96.52	
I	NA20	0.22	0.16	0.18	0.24	0.27	0.38	0.38	0.26		00.00	0.00	0.00	0.00	0.00	0.00	0.03	0.00	0.00	0.01		90.0	0.00	0.0	0.03	;	0.00	0.04	
_	CAO	11.19	11.38	11.26	11.12	10.35	10.54	10.55	10.79		0.21	0.20	0.20	0.22	0.24	0.22	0.22	0.25	0.22	0.23	,	1.85	7 00	00	1.33		0.17	0.48	
heating	MGO	9.29	9.53	9.29	7.61	7.18	7.07	7.13	9.46		44.14	45.63	45.98	45.53	46.18	46.08	45.78	44.67	45.81	44.76	6	30.96	30.12	7	30.68		14.79	13.45	
ž	ONM	0.54	0.40	0.44	0.35	0.34	0.52	0.48	0.60		0.61	0.52	0.63	0.56	0.57	0.56	0.52	0.52	0.65	0.52		0.54	9.0	3	0.51	1	0.50	0.49	
	FEO						8.83				15.57											9.13					14.37 15.48		
IW-1.5		0.58	0.53	0.51	0.42	0.35	0.46	0.45	0.54		0.44	0.53	0.57	0.47	0.48	0.51	0.45	0.54	0.48	0.55	Š	50.0	4 4 4	2 !	0.82	;	37.99 40.68	40.06	
*****	AL203 CR203	16.65	16.46	16.51	13.96	13,23	13.27	13.20	16.32		0.04	90.0	0.04	90.0	0.02	90.0	0.04	90.0	90.0	0.04		0.94	, t	9 1	0.72		26.57	24.68	
ပ	TI02	0.72	0.67	0.62	0.48	0.47	0.46	0.44	0.67		0.02	0.02	0.03	0.00	0.00	0.02	0.02	0.02	0.02	0.02		21.0				2	0.48	0.42	
1200°C	SIO2	glass 51.69	52.33	53.72	56.39	59.86	59.36	59.03	53.15	olivine	39.28	39.34	39.47	39.88	39.90	39.90	40.18	40.69	40.78	40.64	pigeonite	56.74	56.33	0 1	56.54	spinel	0.47	2.85	

	FS										19.92	19.57	20.15	20.31	20.21	20.33	20.33	20.44	20.03	20.36	19.66
	WO										0.97	96.0	101	96.0	1.01	1.08	1.01	1.04	1.06	1.15	1.09
	FFM	51.02	52.03	54.45	23.20	23.58	23.66	23.75	23.64		20.12	19.76	20.35	20.51	20.42	20.56	20.53	20.66	20.24	20.60	19.87
	FEMN	30.75	28.65	27.48	36.65	36.21	39.18	36.50	40.41		29.40	28.84	28.40	29.27	27.42	29.05	27.71	29.23	26.78	27.04	26.58
	Na	35	46	4	c	0	0	0	0		•	-	7	-	7	0	•	-		0	0
	Ca	147	1166	289	ď	o un	တ	5	4		19	19	20	19	20	71	50	20	71	22	21
	Mg		2279 1		7492	1485	1485	1487	1487		1529	1537	1512	1529	1523	1506	1528	1502	1537	1517	1530
	Ma		8 8		5		12	13	<u></u>		13	13	4	4	14		•		15	•	•
	Fe	2481	2569	2543	455	462	464	467	464		388	382	390	398	394	393	398	395	394	397	383
	Ö	9	93	96	6.	4	14	12	4		21	24	71	24	24	52	52	24	52	27	27
	₹	415	1404	436	-		-	-	7		13	14	14	17	17	18	19	19	20	21	73
	E		43		0	0	0	0	0		-	-	٠	-	-	-	0	-	-	•-	-
	Si	7807	1777	7812	1007	1007	1005	1004	1005		1998	1995	2004	1989	1992	2000	1986	2002	1982	1987	1989
Ņ	TOTAL	98.45	99.90	96.96	00.80	101.55	00.19	01.37	01.42		00.44	00.22	01.01	90.00	00.85	96.00	26.00	09.00	100.76	00.88	00.52
H42	NA20 T		0.16 1			0.00										_			0.01	_	_
	CAO	7.08	7.30	8.04	0.21	0.17	0.20	0.17	0.15		0.50	0.49	0.52	0.49	0.52	0.55	0.52	0.53	0.55	0.59	0.56
neating	MGO	10.21	10.25 9.09	9.12	38.97	39.00	38.48	38.95	39.00		28.93	29.03	28.78	28.75	28.87	28.45	28.97	28.44	29.10	28.74	28.95
ř	MNO		0.74			0.59													0.49		
	FEO		20.60								13.10										
IW-0.5	CR203 F				0.66						0.76 1										
≥	AL203 CR				0.02						0.32 (
	2. AL2																				
1300°C	TIO2				00.00						0.03										
130	SIO2	glass 51.64	52.11	52.21	olivine 39.21	39.43	38.83	39.19	39.30	pigeor	56.35	56.18	56.86	55.75	56.31	56.33	56.16	56.52	55.94	56.12	56.12

	FS											17.72	17.70	18.11	13.68	16.46	14.58	12.81
	WO											1.11	1.16	1.19	1.36	1.38	1.33	1.46
	FFM	33.11	34.10	32.84	33.16	32.46		14.97	15.39	14.58		17.92	17.90	18.33	13.87	16.69	14.78	13.00
	FEMN	17.59	18.03	17.87	16.95	16.98		24,46	23.53	22.43		24.96	25.18	26.53	16.10	23.81	18.61	15.95
	Sa	#=	14	=	7	14		0	0	0		0	0		-		0	-
	Ca	1266	1267	1341	1306	1289		4	9	9		22	23	23	27	27	56	29
	Mg	2988	2978	3050	3008	3041		1673	1663	1675		1593	1573	1554	1638	1585	1623	1681
	Ma	87	88	86	91	89		12	13	13		14	14	13	11	5	15	16
	Fe	1522	1586	1533	1538	1504		297	305	288		351	346	352	266	320	284	253
	Ö	=======================================	108	109	105	114		12	14	14		25	23	23	52	27	56	52
	A	1653	1679	1699	1703	1797		-	-	7		15	15	17	22	23	52	28
	F	49	49	49	46	52		0	0	0		-	-	7	-	-	-	7
	iS	7694	7647	7588	7624	7550		266	995	266		1980	1993	1996	1990	1988	1986	1969
H44	TOTAL	100.20	100.28	100.00	99.83	98.05		101.64	100.27	100.82		98.45	100.41	101.85	100.42	100.10	101.34	98.93
I	NA20	0.04	0.05	0.04	0.04	0.05		0.00	0.00	00.0		00.00	0.00	0.01	0.01	0.01	0.00	0.01
	CAO	8.21	8.20	8.65	8.42	8.17		0.17	0.21	0.24		0.57	09.0	0.62	0.71	0.71	0.70	0.76
neating	MGO	13.93	13.85	14.14	13.94	13.85		45.47	44.49	45.20		29.63	29.91	29.96	31.50	30.13	31.40	31.84
Ĕ	ONM	0.71	0.72	0.70	0.74	0.71		0.58	0.61	0.61		0.46	0.46	0.45	0.56	0.45	0.52	0.53
	FEO	12.65	13.15	12.67	12.70	12.21		14.37	14.54	13.86		11.63	11.73	12.09	9.13	10.85	9.80	8.56
IW-1.5	CR203	0.98	0.95	0.95	0.92	0.98		0.64	0.69	0.72		0.86	0.82	0.85	0.89	0.98	0.94	0.88
	TIO2 AL203 CR203	9.75	9.88	96'6	9.98	10.35		0.02	0.04	90.0		0.36	0.36	0.42	0.53	0.55	0.62	0.68
ပ	1	0.45	0.45	0.45	0.42	0.47		00.00	0.00	0.02								
1300°C	SIO2	glass 53.48	53.03	52.44	52.67	51.26	edivine	40,39	39.69	40.11	pigeonite	54.92	56.50	57.38	90'29	56.37	57.31	55.60

	FS																	
	ow.																	
	FFM		34 38	33.12	33.68	33.99	33.70	33.06	33.82	35.86		16.58	16.32	15.85	16.38	16.70	16.29	
	FEMN		30.22	32.73	28 76	28.76	29 14	25.69	28.68	27.81		38.56	39.42	39.79	38.26	37.93	42.51	
	Sa		=	4	14	-	-	4	=	4		O	0	0	-	0	0	
	ථි		570	564	578	580	565	571	553	636		2	0	က	2	7	7	
	Mg		4161	4245	4338	4293	4266	4285	4255	4046		1645	1645	1653	1668	1633	1652	
	Æ		73	65	78	78	76	84	77	83		o	α0	œ	6	o	80	
	F.		2219	2135	2243	2251	2207	2158	2214	2309		329	322	313	328	329	323	
	Ö		106	66	103	100	110	66	95	111		10	=	-	=	10	11	
	¥		687	683	969	702	705	719	729	729		-	-	-	-		-	
	F		70	2	20	22	55	56	20	20		0	-	0	0	0	0	
	Ø		7871	7886	7760	7773	7806	7808	7810	7809		1000	1002	1002	987	1006	666	
8	TOTAL		100.01	100.73	100.89	100.50	100.30	99.18	100.00	101.00		100,26	101.64	100.16	99.70	100.91	101.15	
I	NA20		0.04	0.05	0.05	0.04	0.04	0.05	0.04	0.05		0.00	0.00	0.00	0.03	0.00	0.00	
	CAO		3.62	3.62	3.69	3.69	3.60	3.60	3.51	4.06		0.08	0.01	0.10	0.08	0.08	0.08	
eating	MGO		19.00	19.60	19.91	19.63	19.52	19.42	19.42	18.55		43.86	44.55	44.19	44.16	43.84	44.48	
Ĕ	ONM		0.59	0.53	0.63	0.63	0.61	0.67	0.62	0.67		0.40	0.39	0.37	0.40	0.41	0.36	
	FEO		18.06	17.57	18.35	18.35	18.00	17.43	18.01	18.87		15.62	15.57	14.91	15.50	15.75	15.50	
W-0.5	CR203		0.91	0.86	0.89	0.86	0.95	0.85	0.82	96.0		0.50	0.57	0.56	0.53	0.50	0.54	
	TIO2 AL203		3.97	3.99	4.04	4.06	4.08	4.12	4.21	4.23		0.02	0.02	0.02	0.02	0.02	0.02	
ပ္	T102		0.18	0.18	0.18	0.20	0.20	0.23	0.18	0.18		00.0	0.05	0.02	0.02	0.00	0.00	
1400°C	SIO2	glass	53.59	54.28	53.10	52.99	53.25	52.76	53.14	53.38	arinito	39.73	40.48	39.94	38.96	40.26	40.11	

	FS											96	3.60	5.72	3.29	
	o N											0.32	0.47	0.49	0.49	
	FFM	7.53	7.14	8.27	7 46	6.51	8.54		3.18	3.16	3.02	9.97	3.59	5.72	3.29	
	FEMN	4 51	4.62	4.62	4 56	4.67	4.91		5.96	7.14	6.58	20.05	6.10	10.33	5.55	
	Na	=	S.	m	ന	æ	2		0	-	0	0	0	0	0	
	Ca	1178	1139	1200	1150	1159	1211		7	4	4	4	9	9	9	
	М	4616	4629	4453	4568	4618	4445		1944	1957	1986	1172	1280	1230	1272	
	W	85	78	83	82	20	86		-	6	თ	7	80	7	80	
	Fe	383	362	409	375	326	423		94	94	62	130	48	75	43	
	Ö	106	103	107	110	103	98		11	Ţ	13	7	=	12	12	
	¥	1398	1402	1417	1435	1437	1470		*- -	-		4	7	8	89	
	F	36	39	39	37	24	40		0		-	-	-		-	
	S	7699	7727	7741	7716	7733	7700		981	972	958	1331	1314	1324	1320	
150	TOTAL	97.87	98.49	97.83	98.32	98.53	98.03		97.90	96.93	96.93	98.98	97.55	98.24	97.97	
_	NAZO	0.04	0.02	0.01	0.01	0.03	0.02		0.01	0.03	0.00	0.00	0.01	0.01	0.00	
_	CAO	7.78	7.59	7.92	7.64	7.74	8.00		0.07	0.15	0.16	0.17	0.25	0.26	0.26	
neating	MGO	21.91	22.17	21.12	21.82	22.16	21.11		53,39	53.16	53.87	33.77	36.96	35.59	36.93	
Ē	MNO	0.71	99.0	0.74	69.0	0.59	0.72		0.52	0.43	0.45	0.33	0.40	0.37	0.40	
	FEO	3.24	3.09	3.46	3.19	2.79	3.58		3.14	3.11	3.00	6.70	2.47	3.87	2.25	
IW-1.5	CR203	0.95	0.93	96.0	0.99	0.93	0.88		0.58	0.55	0.64	0.59	0.61	0.67	0.65	
	TIO2 AL203 CR203	8.39	8.49	8.50	8.67	8.72	8.83		0.03	0.04	0.05	0.16	0.26	0.29	0.29	
1400°C	T102	0.37	0.37	0.37	0.35	0.23	0.38		0.01	0.08	0.04					
1400	SIO2	glass 54.48	55.17	54.74	54.95	55.32	54.52	olivine	40.16	39.39	38.73	pigeonite 57.18	56.56	57.13	57,16	

		Na Ni FEMN FFM WO FS	13 82.82 12 98.33	22 83.53	131 10 92.60 77.76 133 16 94.79 77.42		102.90	103.70	109.86	94.11	3 106.38	6 88.68	4 106.22	0 3 123.68 39.38	2 89.25	109.10	87.82	118.48		114.95	4 3 152.04 60.66		0 10732
		Ca	2197	2163	2304 2389	;	4	14	16	16	16	16	16	14	16	50	15	15		4	15	,	0
		Mg			779	1	1217	1202	1212	1205	1206	1203	1211	1210	1225	1190	1198	1197		358	425	•	0
		Ψu	3 4	33	34		x 0	7	7	œ	7	თ	7	9	ග	7	თ	7		9	4	ŧ	S
		Fe	2778	2783	2832 2627	i	788	762	773	785	786	780	786	290	992	771	99/	794		736	663		13171
		Ö	23	15	22 15	•	m	4		4	4	က	4	9		4	2	æ		884	525		15
		₹	2638	2741	2822 2844		-		7	7	ന	ო	ന	ო	4	o	ç	12		1007	1277		0
		F	89	86	111		0	~	0	0	0	0	0	~	0	0	0	0			12		œ
		Ñ	6898 6750	6865	6764 6812		981	1000	992	985	985	988	983	981	987	994	885	977		0	80	1	27
		TOTAL	97.30	97.94	98.64 98.71		98.14	98.42	99.62	98.52	98.45	98.56	98.28	99.22	98.00	99.19	99.13	98.17		95,62	97.38	!	140.47
	NM55	ON N	0.10	0.17	0.08		0.13	0.22	0.11	0.23	0.15	0.27	0.17	0.13	0.11	0.15	0.22	0.11		0.18	0.14		64.28
	2	NA20	0.43	0.47	0.43 0.43		0.01	0.01	0.00	0.03	0.01	0.00	0.00	0.00	0.01	0.01	0.03	0.01		00.0	0.07		0.00
		CAO	12.94	12.83	13.70	;	0.46	0.46	0.53	0.53	0.53	0.52	0.53	0.48	0.52	0.66	0.52	0.49		0.10	0.46		0.00
	heatin	MGO	4.01 3.88	3.70	3.33 3.18	6	28.92	28.83	29.35	28.74	28.74	28.74	28.80	29.05	29.20	28.70	28.88	28.40		7.31	9.43		00.00
	_	MNO	0.25	0.25	0.23		0.32	0.31	0.30	0.35	0.31	0.37	0.31	0.27	0.36	0.30	0.37	0.28		0.23	0.17		0.03
		FEO	20.97	21.15	21.57 20.16		33,35	32.56	33.38	33.36	33.40	33.23	33,35	33.82	32.54	33,15	32.91	33.60		26.78	26.18		75.89
U C	W-0.5	CR203	0.18	0.12	0.18		0.13	0.19	90.0	0.16	0.20	0.15	0.16	0.26	90.0	0.19	0.22	0.35		34.00	21.95		60:0
ırchis	_	AL 203	14.13	14.78	15.25 15.49	į	0.0	0.04	90.0	90.0	60.0	0.09	0.09	0.09	0.11	0.26	0.32	0.36		25.98	35.81		0.00
Natural Murchison	ပ	T102	0.75	0.83	0.78		0.02	0.03	0.00	0.02	0.02	0.00	0.02	0.03	00.0	0.02	00.0	0.00		0.78	0.52		0.05
Natu	1180°C	SIO2	glass 43.54 42.30	43.64	43.09	olivine	34.76	35.77	35.83	35.04	35.00	35.19	34.85	35.09	35.09	35.75	35.66	34.57	lacion	900	2.65	metal	0.13

(Exp type) (Exp #)

C. Series III Experiments (Temp) (fO_2)

	FS																								28.65
	WO																								3.75
	FFM		65.93	65.67	65.78	66.46	65.84		32 48	32.46	32.38	32.58	32.58	32.44	32.76	33.15	32.75	32.39	32.89	32.81	32.76	32.38	32.21	33.52	29.76
	FEMN		81,19	65.62	73.97	74.92	75.51		100 89	79.21	100 89	80.57	20.77	70.79	113.11	111.65	91.95	77.95	125,35	80.52	89,36	72.28	82.71	84.07	57.45
	ž		-	0	7	2	2		4	4	ۍ.	ന	2	4	4	•	7	7	က	က	က	4	-	2	2
	Sa		207	136	165	174	201		0	٥ ا			7	0	-	-			0	7		0	-	0	7
	ပီ		1808	1877	1833	1850	1818		σ	0	0	0	o	7	6	6	6	O	9	6	12	6	10	10	75
	Mg		1165	1120	1158	1141	1136		1334	1312	1328	1309	1318	1314	1307	1317	1322	1316	1314	1335	1318	1321	1325	1306	1350
	Mn		58	34	31	31	30		œ	00	9	æ	7	6	9	9	7	œ	S	80	7	6	œ	œ	10
	E.		2309	2207	2287	2322	2247		645	635	639	636	640	635	640	656	647	634	646	929	646	637	633	662	576
	ŏ		4	42	43	46	40		9	150	7	7	2	9	6	2	7	0 0	7	5	S	80	9	10	17
	A		2367	2428	2409	2378	2427		2	0	2	2	7	7	7	7	n	က	e	ന	4	9	7	7	36
	F		72	72	71	7.1	72		-	0	0	0	0	-	0	0	0	0	0	0	0	0	-	-	7
	S		7409	7422	7393	7393	7409		994	1010	666	1010	1006	1009	1009	1000	666	1008	1002	988	1000	666	866	991	1949
	TOTAL		97.97	97.47	98.20	98.13	97.59		06 66	99.23	100.64	100.66	99.91	101.27	100.48	100.92	100.39	100.49	101.56	101.05	100.88	100.74	100.94	101.74	99.62
VM59	ON N		60.0	0.00	0.02	0.04	0.04		0.19	0.17	0.25	0.15	60.0	0.20	0.18	0.04	60.0	0.08	0.14	0.14	0.13	0.20	90.0	0.24	0.06
Z	NA20		0.70	0.46	0.56	0.59	0.68		0.03	0.03	0.01	0.01	0.03	00.0	0.01	0.01	0.01	0.01	0.00	0.03	0.01	0.00	0.01	0.00	0.03
	CAO		11.06	11.47	11.25	11.32	11.11		0.32	0.32	0.32	0.32	0.32	0.24	0.31	0.32	0.32	0.31	0.36	0.32	0.43	0.31	0.35	0.35	1.90
ating	MGO		5.12	4.92	5.11	5.02	4.99		33.21	32.57	33,36	32.96	32.91	33.28	32.83	33.11	33.08	33.08	33.29	33.53	33.16	33.23	33.43	32.96	24.47
Pe	MNO		0.22	0.26	0.24	0.24	0.23		0.28	0.35	0.28	0.35	0.31	0.40	0.25	0.26	0.31	0.36	0.23	0.36	0.32	0.39	0.34	0.35	0.32
	FEO		18.09	17.28	17.98	18.21	17.59		28.61	28.08	28.61	28.56	28.50	28.68	28.64	29.40	28.87	28.42	29.20	29.36	28.96	28.55	28.48	29.80	18.62
IW-1.0	CR203		0.34	0.35	0.36	0.38	0.33		0.29	0.25	0.35	0.32	0.23	0.28	0.44	0.22	0.34	0.37	0.34	0.23	0.22	0.39	0.48	0.47	0.57
	AL203		13.16	13.49	13.44	13.23	13.48		90.0	90.0	90.0	90.0	90.0	90.0	90.0	90.0	0.08	0.08	0.11	0.11	0.13	0.19	0.21	0.23	0.89
ပ	TI02	:	0.63	0.63	0.62	0.62	0.63		0.03	000	0.00	0.02	0.00	0.03	00.0	00.0	0.00	0.00	0.02	0.00	0.02	0.00	0.03	0.03	0.07
1180°C	SIO2	glass	48.56	48.61	48.62	48.48	48.51	alivina	36.88	37.40	37.40	37.91	37.46	38.10	37.76	37.50	37.29	37.78	37.87	36.97	37.50	37.48	37.55	37.31	pigeonite 52.69
				-	-	-										-		-	-			-		-	

	FS																				21.00	20.27	20.33	19.51	21.43	19.84	20.07				
	٥																				4.50	4.13	4.34	3.89	4.41	4.30	4.48				
	FFM	47.89	46.72	46.75	49.16	46.84	49.65	45.29	49.78		21.63	21.57	21.71	21.71	22.31	21.38	21.25	21.05	21.40		21.99	21.14	21.25	20.30	22.41	20.73	21.01				
	FEMN	41.93	37.55	39.85	47,59	39.30	37.93	43.08	38.97		49.34	70.56	71.69	56.39	46.22	55.57	60.01	60.29	55.07		48.10	28.46	35.22	35.11	38.79	36,39	37.81				
	ź	13	5	7	F	თ	7	-	7		-	-	-		2	7	7	-	7		၈	7	S	0		-	0		3071	2066	
	Ra	287	256	243	217	207	263	227	262		0	0	0		-	0	0	0	0		2	ო	7	•	က	7	7		0 0	00	
	S	1615	1631	1584	1557	1534	1663	1501	1627		9	7	7	œ	œ	7	7	7	7		87	82	82	11	86	84	88		00	00	
	Mg	1552	1559	1536	1484	1617	1434	1707	1414		1553	1567	1565	1556	1512	1548	1558	1559	1538		1429	1487	1456	1505	1435	1470	1463		0 ;	_ 0	
	Ψ	35	37	32	31	37	38	33	37		6	9	ထ	œ	6	80	7	7	œ		89	14	F	7	7	=	10		۲.	2 12	
	Fe	1458	1399	1379	1465	1458	1452	1441	1439		431	432	436	434	437	423	422	418	421		405	402	396	386	418	387	392		20857	21917	
	ঠ	4	44	48	51	25	47	4	44		œ	5	9	9	œ	80	თ	6	F		17	16	17	23	73	7,	22		58		
	₹	2433	2551	2474	2474	2486	2510	2563	2556		-		7	7		-	 -	7	4		33	4	48	35	38	45	47		0 0	0	
	F	61	72	64	63	99	62	64	69		0	-	0	₹	0	0	0	0	-		ĸ	•	7	7	e	ო	6		0 (00	
	S	7676	7603	7713	7715	7651	7656	7586	7654		993	984	983	986	1009	666	994	966	1001		1992	1962	1972	1965	1977	1970	1969		5 5	<u>5</u> 4	
	TOTAL	98.00	97.81	97.91	98.55	98.20	98.57	97.95	98.74		100.04	99.91	100.83	99.60	99.35	100.18	99.15	100.56	99.77		100.98	99.04	100.04	99.31	99.04	99.77	99.61		132.54	133.39	
NM57	oN N	0.11	0.04	90.0	0.09	0.08	90.0	0.01	90.0		0.04	0.03	0.04	0.05	0.08	0.09	0.08	0.03	0.11		0.11	0.08	0.19	00.0	0.03	0.04	0.00		17.56	11.90	
_	NAZO	1.01	0.90	0.86	0.77	0.73	0.93	0.80	0.93		0.00	0.00	0.01	0.01	0.01	0.01	0.00	0.00	0.00		0.07	0.04	0.03	0.01	0.04	0.03	0.03		0.00	000	
_	CAO	10.28	10.38	10.13	10.00	9.81	10.65	9.58	10.45		0.22	0.27	0.24	0.29	0.28	0.25	0.24	0.25	0.24		2.28	2.11	2.20	1.99	2.21	2.18	2.27		0.00	900	
eating	MGO	7.10	7.13	7.06	6.85	7.43	6.60	7.83	6.53		40.41	40.66	40.96	40.24	39.08	40.44	40.26	40.92	40.03		26.99	27.49	27.23	27.99	26.50	27.47	27.26		0.00	0.0	
٤	MNO	0.28	0.30	0.28	0.25	0.30	0.31	0.27	0.30		0.40	0.28	0.28	0.35	0.43	0.35	0.32	0.32	0.35		0.28	0.46	0.37	0.36	0.35	0.35	0.34		0.04	0.03	
	FEO	11.89	11.41	11.30	12.05	11.94	11.91	11.78	11.84		19.99	20.01	20.33	19.99	20.13	19.70	19.45	19.54	19.52		13.64	13.26	13.20	12.80	13.75	12.90	13.02		114.73	119.26	
IW-1.5	CR203	0.35	0.38	0.42	0.44	0.45	0.41	0.35	0.38		0.41	0.51	0.51	0.50	0.41	0.41	0.45	0.44	0.53		09.0	0.56	0.61	0.79	0.72	0.75	0.79		0.15	0.01	
	TIO2 AL203	14.08	14.76	14.38	14.44	14.45	14.61	14.87	14.93		0.04	0.04	90.0	0.08	0.04	0.04	0.04	90.0	0.13		0.79	0.93	1.13	0.83	0.89	1.06	1.10		0.00	0.00	
ပ္	TIO2	0.55	0.65	0.58	0.58	09.0	0.57	0.58	0.63		0.02	0.03	0.02	0.05	0.00	0.00	0.02	0.02	0.03								0.10		0.00	0.00	
1180°C	SIO2	glass 52.35	51.86	52.84	53.08	52.41	52.52	51.88	52.69	olivine	38.51	38.08	38.38	38.04	38.89	38.89	38.29	38.98	38.83	nineonit	56.09	54.06	55.00	54.47	54.43	54.87	54.70	metal	0.06	0.02	

1400°C	ပ္		IW-0.5		_	eating		Z	NM63															
SIO2	T102	AL203	CR203	FEO	MNO	MGO	CAO	NA20	<u>Q</u>	TOTAL	S	ï	Ā	ర	e e	Mn	Μ̈́	ပ္ခ	s E	Z	FEMN	FFM	OM M	FS
glass	0	i.	5	9	C		,	Č	0	00		ţ	6		ļ			9	;	4		:		
52.20	0.33	6,35	0.73	13.79	0.85	18.64	4.72	0.05	0.00	97.66	1121	3/	1108	•	707			/49	4	0	16.02	28.80		
52.37	0.35	6.65	0.69	14.27	0.74	18.34	5.7	0.05	0.00	99.17	7668	36	1148	•	748			969	4	0	19.04	29.91		
52.48	0.38	6.80	0.80	14.37	0.79	17.21	5.85	0.05	0.00	98.73	7719	42	1179	93	1768	98	3774	922	14	0	17.96	31.34		
52.10	0.37	6.84	0.83	15.49	96.0	15,31	6.56	0.04	0.00	98.50	7743	41	1198		925		-	045	12	0	15.93	35.40		
olivine																								
40.07	0.00	0.01	0.44	11.52	0.53	46.90	0.14	0.0	0.16	99.78	966	0	0	თ	239		738	4	0	က	21.46	12.04		
40.09	0.00	0.02	0.48	11.46	0.54	47.47	0.14	0.00	0.03	100.23	991	0		6	237		750	4	0	-	20.95	11.86		
39.88	0.01	0.02	0.41	11.60	0.41	47.45	0.11	0.00	0.00	99.88	066	0		œ	241		755	က	0	0	27.94	12.01		
39.95	0.01	0.03	0.38	11.48	0.41	46.82	0.12	0.00	0.00	99.20	266	0		7	240		742	e	0	0	27.65	12.04		
40.11	0.00	0.04	0.35	11.54	0.56	47.39	0.14	0.03	0.10	100.26	992	0		7	239		747	4	-	7	20.35	11.95		
40.46	0.00	0.04	0.37	11.46	0.40	47.14	0.14	0.03	00.0	100.04	1000	0		7	237		737	4		0	28.29	11.95		
40.46	0.00	0.04	0.51	11.64	0.45	46.88	0.13	0.01	0.11	100.23	1000	0		10	241		727	es	0	7	25.54	12.17		
40.07	0.03	0.04	0.44	11.31	0.45	47.34	0.14	0.01	0.00	99.83	993	-		6	234		750	4	0	0	24.82	11.76		
40.13	00.0	0.05	0.47	11.46	0.49	47.52	0.13	0.01	0.05	100,30	991	0		0	237		750	e	0	•	23.09	11.86		
40.22	0.00	90.0	0.47	11.41	0.50	47.17	0.10	0.00	0.09	100,02	966	0	7	6	236		741	e	0	7	22.53	11.89		
39.81	0.00	90.0	0.56	11.48	0.44	47.37	0.15	0.03	90.0	96.66	988	0	7	Ξ	238		752	4			25.76	11.91		
40.19	0.03	90.0	0.54	11.39	0.48	46.76	0.12	0.00	0.04	99.65	666	-	7	11	237		732	en	0	-	23.43	11.96		
40.24	0.05	0.08	0.50	11.26	0.43	46.91	0.14	0.03	0.00	99.64	666	-	7	10	234		736	4		0	25.86	11.81		
39.86	0.00	0.08	0.41	11.28	0.45	47.44	0.15	0.00	0.09	96.76	066	0	7	80	234	9	1756	4	0	7	24.75	11.71		
40 11	0 0	0.36	0.42	11 50	0.48	46.21	0.29	000	00.0	99.39	566	0	1	80	240		716	æ	С	О	23.66	12 19		

70% Allegan-30% Murchison

	FS																			28.04	28 12	28.85	27.34					
	WO																						3.47					
	FFM	62.05	64.58	65.10		32.60	32.66	33,53	32.92	22.53	34.24	22.64	33.05	34.50	33.53	32.70	33.31	33.13	32.62	28 98	20 24	29.91	28.33		72.48	72.29		
	FEMN			69.35									71.58										49.51		94.31	86.16		
	Ē	r :	- 4	0 0	2	4	،	7	ۍ ،	dr u	n u	> <	1 4	. 2	7	m	œ	9	2	4			4		2	ო	41.50	14244
	Na	346	406	433	3	-	 -	0	Ψ.	- •	- c	o c	· ·	. 0	က	0	0	0	-	0	0	0	7		0	0		. 4.
	C	1346	1395	1390		7	∞ .	φ.	co (1 0	~ α	۸ د	~ oc	ω	9	80	80	7	0	8	26	69	68		•	7	c	00
	Mg	1318	1256	1164		1324	1328	1302	1305	1216	1268	1200	1310	1280	1300	1313	1311	1299	1316	1331	1333	1308	1353		298	296	c	4
	M	32	3 2	33	5 '	∞ :	9	œ	ao a	0 0	۸ ٥	۰ ۵	n o	9	Ø	9	6	9	9	5	ç	5	=======================================		6	6	u	9 0
	Đ.	2208	2351	2231	3	644	647	99	644	000	400	89	651	678	629	641	629	647	640	547	555	562	539		807	795	1000	9714
	ö	37	8 8	38	3	တ	မှ မ	on -	ဖ	10	- 1	- 1	- «	2	7	9	7	œ	2	17	<u>0</u>	19	7.		1565	1568	4	0
	Ā	1863	1887	1928		0 -	-	-		- •					က	m	က	es	9	7	e	32	38		300	314	c	00
	F	54	27	59	3	,- .	Ψ.		0 +	- c	> c	0 0	o c	·	-	0	-	0	-	c			5		37	33	c	0.4
	S	7979	7884	7949		1001	666	1002	1010	200	989 401 8	100	500	1006	1003	1007	994	1009	1002	1991	1972	1979	1966		2	ю	*	<u>,</u> 00
	TOTAL	96.34	98.40	96.90		100.06	100.24	99.71	100.36	50.00	100.25	00 77	100 27	100.26	100.33	99.66	100.21	100.19	99.86	60	98.64	99.84	99.73		92.75	95.31	, ,	153.04
AM54	ON N	0.06	0.03	00.0	3	0.20	0.05	0.09	0.22	0 0	77.0	120	0.0	0.22	0.11	0.15	0.38	0.29	0.23	0.15	0.04	0.00	0.13		0.15	0.11	90	92.33
∢	NA20	1.17	1.39	1.47	3	0.0	0.0	000	0.0	5 6	5 6	000	8 6	000	0.05	00.0	0.00	0.00	0.01	000	000	000	0.03		00.0	0.00	8	0.0
	CAO	8.24	8.65	8.54 7.86	3	0.25	0.27	0.27	0.28	0.00	0.24	200	0.29	0.27	0.35	0.28	0.28	0.24	0.31	6	6	1.75	174		0.03	0.04	5	000
eating	MGO	5.80	5.60	5.14 20.00		33.03	33.20	32.28	32.70	32.30	31.65	3 6	32.73	31.84	32.45	32.71	32.60	32.47	32.83	24 19	24.04	23.86	24.71	: !	5.34	5.46	ć	0.05
þe	MNO	0.25	0.26	0.25		0.37	0.28	0.35	0.34	0.0	20.0	300	0.39	0.28	0.27	0.28	0.41	0.28	0.27	23		0.0	0.35		0.27	0.30	ć	0.03
	FEO	17.32	18.68	17.56		28.66	28.84	29.20	28.77	20.02	20.80	20.04 20.08	29.00	30.05	29.32	28.47	29.23	28.82	28.47	17 72	17.84	18 29	17.55		25.79	26.18	0	60.57
IW-0.5	CR203	0.31	0.26	0.32		0.26	0.29	0.42	0.28	200	2,0	500	0.02	0.22	0.32	0.28	0.31	0.37	0.25	750	800	99	0.72	!	52.92	54.60	ć	0.03
	AL203	10.37	10.64	10.77		0.00	0.02	0.02	0.05	70.0	0.07	0.02	0.0	0.04	0.08	0.08	0.09	0.11	0.19	0.40	27.0	2.0	0.87		6.80	7.33	6	000
ပ့	T102	0.47	0.50	0.52		0.03	0.03	0.03	0.00	0.0	20.00	3 6	0.0	0.03	0.03	0.02	0.03	0.02	0.05				0.0		1.32	1.20	6	0.03
1180°C	SIO2	glass 52.35	52.39	52.33 52.56	olivine	37.25	37.25	37.05	37.74	00.70	37.87	27.10	37.35	37.31	37.35	37.42	36.88	37.59	37.25	pigeonite	53.03	53.63	53.55	spine	0.13	60.0	metal	0.09

	FS																27.80	27.59	27.97	27.89	28.24	29.68
	WO																3,60	3.38	3.61	3.63	3.33	3.31
	FFM	65.94	64.17	67.68	60.77		33.07	33.31	33.24	33.72	32.89	33.07	33.29	33.01	33.17		28.84	28.55	29.05	28.95	29.21	30.69
	FEMIN	59.75	63.37	53.59	62.59		65.44	84.42	77.76	81.79	84.62	77.20	79.56	68.22	70.63		48.63	50.44	79.66	65.50	49.72	51.13
	ž	17	18	~ *	0		-	0	0	,- -	0	0	7	0	က		0	0	က	m	7	- -
	Na	295	331	328	292		0	7	7	0	0	7	7		-		4	က	7	-	6	7
	S	1464	1419	1490	1448		6	6	80	7	6	æ	9	9	თ		71	67	7	72	65	49
	Mg	1193	1232	1038	1424		1323	1308	1313	1302	1322	1321	1311	1300	1306		1347	1355	1335	1351	1334	1340
	Man	40	36	42	8, 8		10	80	80	80	œ	6	80	6	o		11	-	7	80	7	12
	Fe	2388	2271	2261	2261		629	657	658	999	652	657	658	645	653		550	546	548	554	555	598
	Ö	49	56	35	94		œ	10	0	Ę	7	10	ю	20	7		24	22	77	24	22	24
	¥	2138	2130	2192	2045		-	-	-	2	7	7	e	4	2		33	33	34	36	38	44
	F	22	65	8 63	25		0	0	-		0		0	0	-		7	-	-	7	7	က
heating AM58	S	7677	7747	7767	7719		992	1000	266	266	988	885	266	666	1000		1964	1967	1975	1958	1969	1937
	TOTAL	97.73	77.76	97.85	97.10		96.66	99.73	89,89	101.12	100.66	99.37	99.43	98.88	101.23		98.84	100.39	99.12	99.34	100.19	99.11
	ON	0.14	0.15	0.06	0.00		0.03	0.00	0.00	90.0	0.00	00.0	0.08	0.01	0.14		0.00	0.01	0.09	0.11	90.0	0.04
	NA20	1.00	1.13	1.12	8		0.00	0.03	0.03	0.00	0.00	0.03	0.04	0.01	0.01		0.05	0.04	0.03	0.01	0.04	0.03
	CAO	8.98	8.76	9.21	8.87		0.31	0.31	0.27	0.25	0.31	0.28	0.35	0.34	0.32		1.79	1.71	1.79	1.82	1.67	1.67
	MGO	5.26	5.47	4.61	6.27		32.85	32.45	32.62	32.68	33.16	32.63	32.40	32.02	32.91		24.34	24.90	24.23	24.51	24.44	24.06
	ONM	0.31	0.28	0.33	0.28		0.44	0.34	0.37	0.36	0.34	0.37	0.36	0.41	0.41		0.36	0.35	0.22	0.27	0.36	0.37
	FEO	18.76	17.97	17.91	17.75		29.16	29.07	29.14	29.82	29.14	28.93	29.01	28.33	29.33		17.73	17.88	17.75	17.91	18.13	19.16
IW-1.0	CR203	0.41	0.22	0.29	0.38		0.38	0.48	0.44	0.51	0.34	0.48	0.37	0.94	0.34		0.83	0.77	0.73	0.83	0.75	0.80
_	AL203 CR203	11.92	11.96	12.32	11.39		0.04	0.04	0.04	90.0	90.0	90.0	0.09	0.13	0.15		0.76	0.77	0.79	0.83	0.87	1.00
ပ္	T102	0.50	0.57	0.55	0.48		0.02	00.0	0.03	0.05	00.0	0.05	0.0	00.0	0.03							0.12
1180°C	SIO2	glass 50.45	51.26	51.45	50.68	ofivine	36.73	37.01	36.95	37.33	37.31	36.54	36.73	36.69	37,59	1	52.91	53.91	53.46	52.97	53.80	51.86

	FS			19.51 20.75 20.63 19.13 19.31	
	0M			2 9 9 9 9 9 9 9 9 9 9 9 9 9 9 9 9 9 9 9	
	FFM	39.55 42.24 41.55 40.27 40.54 41.19	17.21 16.99 17.60 17.51 17.56 17.46 17.08	20.13 21.40 21.34 19.74 20.10	45.66 54.41
	FEMN	37.90 34.10 34.95 29.56 28.14 30.21	37.11 36.92 40.88 41.49 43.98 39.37 46.70 37.90	37.29 39.81 36.56 35.43 29.18	39.40 44.23
	ž	V080+0	-40-0-00	00000	2 1 6066 3683
	Sa Sa	303 300 226 247 279	0000000	n ∞ + n 01 +	1 4 60
	S.	1487 1582 1524 1508 1525 1493	∠ 9 9 ∠ 9 9 9 9	60 61 64 60 52	m 0 0 0
	Mg	1579 1 1579 1 1651 1 1694 1 1678 1	1630 1635 1624 1623 1616 1628	1503 1510 1458 1507 1481	529 487 22 0
	Wu	36 4 4 4 8 8 8 8 8 9 9 9 9 9 9 9 9 9 9 9 9	o o o o o o o o o o o o o o o o o o o	551155	54 25
	Fe	1180 1180 11198 1172 1176	341 337 349 348 347 337 357	381 414 399 373 376 341	454 598 17894 20183
	ŏ	4 4 4 4 8 8 8 8 8 8 8 8 8 8 8 8 8 8 8 8	∞ no no no no no t−	25 25 25 23 25 23	1139 1553 5 1 5 1
	A	2517 2608 2675 2700 2728 2729	-0000004	23 28 35 54 51	811 318 0
	E	68 63 65 63	000-0-00	Иипппп	43 3 0
	ത	7661 7695 7626 7612 7612 7587	1000 999 999 999 1004 1003	1985 1966 1993 1975 1970	23 3 15 15
	TOTAL	97.98 97.18 98.09 97.02 97.97	100.37 101.01 100.63 99.88 100.42 100.64 100.83	100.53 99.59 100.04 100.45 101.60	97.58 94.65 136.83
AM56	ON N	0.00 0.03 0.01 0.00	0.03 0.17 0.04 0.04 0.01 0.00	0.00 0.08 0.05 0.00 0.04	0.08 0.05 35.62 18.34
∢	NA20	1.06 0.96 0.88 1.00	00000000	0.00 0.00 0.00 0.00 0.00	0.00
_	CAO	9.58 10.12 9.84 9.65 9.65	0.25 0.25 0.25 0.25 0.22 0.22 0.22 0.22	1.58 1.68 1.68 1.57 2.03 1.39	0.00 0.00 0.00
eating	MGO	8.41 7.26 7.66 7.79 7.78 7.78	43.38 43.26 43.26 43.09 43.14 43.59 43.59	28.39 28.06 27.36 28.45 28.09 29.41	9.22 9.22 0.07
ž	MNO	0.28 0.28 0.32 0.34	0.43 0.39 0.37 0.34 0.34	0.34 0.34 0.35 0.35	0.43 0.45 0.01 0.10
	FEO	9.98 9.67 9.91 9.58 9.69	16.16 16.08 16.39 16.39 16.83 16.08	12.84 13.71 13.33 12.56 12.71 11.68	17.16 20.16 101.07 96.68
IW-1.5	CR203	0.37 0.34 0.37 0.38 0.37	0.39 0.28 0.44 0.41 0.38	0.82 0.56 0.79 0.88 0.89	45.51 55.41 0.03 0.25
	- 1	14.74 15.17 15.70 15.70 16.00	0.04 0.06 0.06 0.08 0.08 0.03	0.55 0.62 0.66 0.83 1.08	21.75 7.60 0.00 0.02
ပ	TI02 AL 203	0.62 0.58 0.60 0.60 0.58	0.000	0.07 0.07 0.10 0.12 0.12	0.62 1.62 0.02 0.00
1180°C	SIO2	glass 52.88 52.76 52.76 52.18 52.46 53.06	olivine 39.69 39.90 39.66 39.19 39.94 40.03	pigeonite 55.90 54.47 55.75 55.60 55.71 56.50	spinel 0.73 0.09 metal 0.00

	FS																	
	۸o																	
	Z.	52 63	47.53	33.35	8.89		8 84	8 64	8.39	8.72	17.85	8.21	8.47	8.75	8.67	8.45	8.38	8.86
	FEMN		27.91															36.22
	u.						7	7	4,	7	7	7	(,	(-)	ų)	7	7	(,)
	Ž	12	0	4	0		4	(4)	4	4	4	•	2	5	0	4	7	
	Na	15	12	12	24		0	0	0	0	0	0	0	0	0	0	0	0
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	M	2368	2686	2309	2447		1616	1620	1603	1610	1614	1640	1610	1621	1618	1608	1615	1619
	ž	85	06	82	74		œ	6	7	6	00	œ	10	10	9	Ø	6	10
	ñ. e	2726	2514	2735	2411		377	373	363	373	352	367	367	376	373	366	365	379
	ర	102	96	97	8		12	10	7	2	თ	o	6	10	10	O	œ	Ξ
	₹	1049	1160	1250	1313		-	-	-	-	-	-	-	-	-	ď	7	7
	F	36	58	38	45		0		0	0	0	0	0	0	0	0	0	 -
	S	8051	7965	7927	7991		987	988	1004	885	1002	983	966	984	991	866	992	983
	TOTAL	99.17	97.47	97.22	97.44		98.80	99.03	99.10	99.70	101.11	99.13	99.58	100.06	100.58	100.07	100.19	99.43
M62	OIN	0.10	0.00	0.03	0.00		0.19	0.14	0.20	0.19	0.19	0.05	60.0	0.25	0.00	0.18	0.32	90.0
₹	NA2O	0.05	0.04	0.04	0.07		0.00	00.0	0.00	0.00	0.01	0.01	0.00	0.01	0.00	0.00	0.00	0.00
	CAO	5.58	5.09	5.55	5.53		0.10	0.13	0.10	0.13	0.13	0.11	0.10	0,10	0.13	0.08	0.13	0.17
ating	MGO	10.55	11.86	0.08	98.0		11.93	12.18	11.97	12.23	13.16	12.80	12.28	12.56	12.85	12.43	12.60	12.23
þe	ONM		0.70				0.37	•	•	•	•	•	•	•	•	•	•	•
			19.79 0				17.44 0											
.5	3 FEO								•									
IW-0.5	CR20	0.86	0.82	0.8	0.6		0.58	0.4	0.3	0.5	0.4	0.47	0.44	0.5(0.5(0.44	0.43	0.56
	TIO2 AL203 CR203	5.91	6.48	6.90	7.37		0.02	0.02	0.02	0.02	0.04	0.04	0.04	0.04	0.04	90.0	0.06	0.0
O	T102	0.32	0.25	0.33	0.40		0.00	0.03	00.0	0.02	0.00	0.02	0.02	0.00	0.00	0.02	0.02	0.03
1400°C	SIO2	glass 53.48	52.44	51.58	52.88	olivine	38.17	38.34	39.21	38.77	39.94	38.25	39.00	38.53	39.15	39.26	39.04	38.23

Cation mineral formulas based on: 24

24000 oxygens for glass
4000 oxygens for olivine (ideal olivine = 3000 cations per 4000 oxygens)
6000 oxygens for pigeonite (ideal pyroxene= 4000 cations per 6000 oxygens)
4000 oxygens for spinel (ideal spinel = 3000 cations per 4000 oxygens)
24000 oxygens for metal

FEMN = (FeO/71.846) / (MnO/70.938) FFM = (FeO/71.846) / [(FeO/71.846)+(MnO/70.938)+(MgO/40.304)] WO = (CaO/56.08) / [(CaO/56.08)+(FeO/71.846)+(MgO/40.304)] FS = (FeO/71.846) / [(CaO/56.08)+(FeO/71.846)+(MgO/40.304)]